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# Activity *versus* selectivity in photocatalysis:

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## Morphological or electronic properties tipping the scale

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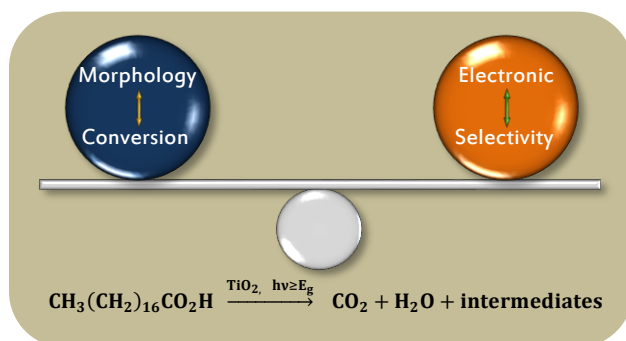
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### Graphical abstract



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14 **ABSTRACT**

15 In this paper a structure-activity and -selectivity relation is established for three commercial  
16 TiO<sub>2</sub> sources (P25, P90, PC500). Morphological and electronic parameters of the photocatalysts  
17 are determined using widely applicable and inexpensive characterization procedures. More  
18 specifically, the electronic properties are rigorously characterized using an electron titration  
19 method yielding quantitative information on the amount of defect sites present in the catalyst.  
20 Surface photovoltage measurements on the other hand provide complementary information on  
21 the charge carrier recombination process. As model reaction, the degradation of a solid layer of  
22 stearic acid is studied using an in-situ FTIR reaction cell that enables to investigate the catalyst  
23 surface and possible formation of reaction intermediates while the reactions are ongoing. We  
24 show that the order of photocatalytic conversion is: PC500 > P90 > P25, matching the order of  
25 favorable morphological properties. In terms of selectivity to CO<sub>2</sub> formation (complete  
26 mineralization), however, this trend is reversed: P25 > P90 > PC500, now matching the order of  
27 advantageous electronic properties, i.e. low charge carrier recombination and high charge  
28 carrier generation. With this we intend to provide new mechanistic insights using a wide variety  
29 of physical, (wet) chemical and operando analysis methods that aid the development of  
30 performant (self-cleaning) photocatalytic materials.

31 **KEYWORDS**

32 Photocatalysis, titanium dioxide (TiO<sub>2</sub>), structure-activity relation, stearic acid, in-situ FTIR,  
33 selectivity, electron titration, surface photovoltage, self-cleaning

## 34 1. INTRODUCTION

35 Photocatalysis is a well-established technology for pollution abatement in both the gas and  
36 aqueous phase.[1–4] More recently, photocatalytic self-cleaning materials have become an  
37 important application field as well.[5] In the selection of the most suitable catalyst, not only  
38 high activity but also pricing and safety play an important role. In this regard titanium dioxide  
39 ( $\text{TiO}_2$ ) is an ideal candidate since it is relatively inexpensive, non-toxic, (photo-)stable and easily  
40 activated under UV light illumination.[6] In addition, it is already used in a number of everyday  
41 applications like the food industry, sunscreen, paint and cosmetics, that could facilitate rapid  
42 transition of self-cleaning  $\text{TiO}_2$  applications from the laboratory scale to commercial  
43 applications.[5] One of the major drawbacks of  $\text{TiO}_2$  based photocatalysis is its low efficiency,  
44 which is mainly due to charge carrier recombination. This problem can be circumvented by  
45 improving the photo-electronic properties or by modifying the photocatalyst on the  
46 morphological level.[7] The former involves the formation of composite semiconductors,  
47 doping or modification with (noble) metal nanoparticles.[8] The latter is mainly based on  
48 maximizing the exposure of the catalyst surface to pollutants by increasing the surface area  
49 using different (nanosized) geometries.[9] An additional advantage is that charge carriers are  
50 formed closer to the surface thus reducing charge carrier recombination.[7,10] It is evident that  
51 different factors play an important role and understanding the interplay between these  
52 phenomena is essential for further optimizing the photocatalytic efficiency. One should also  
53 keep in mind the final application. It is shown that the material properties that dominate the  
54 overall photocatalytic activity in aqueous phase differ from those in gas phase.[11–13] On the

55 other hand, very limited information is available on the driving properties for self-cleaning  
56 applications taking place at the solid (catalyst) - solid (pollutant) - gas (ambient) interface.

57 In this study both morphological and electronic properties of three commercially available  
58 photocatalysts are rigorously characterized in order to adequately evaluate the photocatalytic  
59 self-cleaning activity and thus establish a full structure-activity relation. A valuable correlation  
60 between photocatalytic activity and structural and physical properties of commercial  $\text{TiO}_2$  has  
61 been established by Ohtani's group using statistical multivariable analysis.[13] They used five  
62 different representative reactions and their relative rates were linked to six different material  
63 properties. Since all reactions were conducted in aqueous or gas phase, no conclusions can be  
64 made on the driving factors of self-cleaning photocatalytic materials. The electronic properties  
65 of photocatalysts can be studied using different (fundamental) spectroscopic techniques like  
66 electron paramagnetic spectroscopy (EPR),[14–16] time resolved microwave conductivity  
67 (TRMC) measurements,[17] photoacoustic spectroscopy (PAS)[18,19] and other ultrafast  
68 spectroscopic techniques.[20–22] While these techniques yield insightful information, they  
69 require dedicated expensive equipment and often tedious and complex sample preparation  
70 steps. In this work it is our intent to use simple, widely applicable and inexpensive  
71 characterization procedures that yield all the information required. More in particular, we used  
72 a facile and fast characterization technique based on an adapted electron titration method  
73 using a thiazine dye (thionine) yielding information on the electronic properties of powder  $\text{TiO}_2$   
74 samples. This electron titration method is based on the generation and storage (trapping) of  
75 electrons of irradiated  $\text{TiO}_2$  colloids in de-aerated conditions in the presence of a hole  
76 scavenger. UV excitation of  $\text{TiO}_2$  in de-aerated ethanol renders the suspension blue, which is

77 attributed to trapping of electrons at  $Ti^{4+}$  sites. This electron excess can be titrated using a  
78 suitable acceptor molecule like thionine. Because of the charging effect and the possibility to  
79 titrate the excess charge, it is possible to quantify the number of electrons stored on the  $TiO_2$   
80 particles.[23–30] The abovementioned electron titration method is thus used to evaluate the  
81 electronic properties of different commercially available  $TiO_2$  photocatalysts. These results are  
82 complemented by direct surface photovoltage (SPV) measurements and morphological  
83 characterization data. Further linking the electronic and morphological characteristics of the  
84 samples to their ability to degrade a solid layer of stearic acid (SA), enables us to interpret the  
85 photocatalytic self-cleaning behavior of the investigated catalysts.

86

## 87 **2. EXPERIMENTAL**

### 88 **2.1 Chemical reagents**

89 All products were used as received from the manufacturers without any modification or  
90 purification unless stated otherwise. Three commercially available  $TiO_2$  sources were used:  
91 Aeroxide P25 (Evonik), Aeroxide P90 (Evonik) and PC500 (CristalACTiV). All other chemicals  
92 were purchased from Emplura or Sigma-Aldrich.

### 93 **2.2 Photocatalyst characterization**

94 X-ray powder diffraction (XRD) measurements were conducted using a STOE StadiP apparatus  
95 with  $CuK\alpha$  radiation and an image plate detector. The Brunauer-Emmett-Teller (BET) specific  
96 surface area was determined via  $N_2$  adsorption with a Micromeritics Tristar Surface Area and  
97 Porosity Analyzer. The samples were first degassed overnight at 573 K. A filler rod and  
98 isothermal jacket were used during the measurements.

99 The amount of electrons stored on TiO<sub>2</sub> was determined quantitatively using thionine acetate  
100 salt (Sigma-Aldrich) as electron acceptor. 0.625 mg TiO<sub>2</sub> powder was suspended in 5 mL  
101 absolute ethanol (Emplura, 99.5%) in an airtight reaction vessel and de-aerated with N<sub>2</sub>. The  
102 weight before and after the de-aeration step was measured to account for volume losses. The  
103 reaction vessel was subsequently illuminated with a low-intensity UVA lamp ( $\lambda_{\text{max}} = 352 \text{ nm}$ ,  $5.3$   
104  $\pm 0.3 \text{ mW cm}^{-2}$  at sample distance) for one hour while stirring. The holes are readily scavenged  
105 by the organic solvent and the suspension turns grayish-blue, which indicates the formation of  
106 electron-rich TiO<sub>2</sub>. Afterwards, 1 mL of a de-aerated aqueous solution of thionine (0.1 mM) was  
107 added to the reduced TiO<sub>2</sub> suspension causing the stored electrons to reduce the thionine dye  
108 (blue) to its colorless *leuco*-thionine form. The resulting suspension was centrifuged for 20  
109 minutes at 6000 rpm to eliminate the photocatalyst, after which the supernatant was loaded in  
110 a sealable quartz cell (optical path length 10 mm) and the UV-VIS absorption at 602.5 nm was  
111 measured using a Shimadzu UV-VIS 2501PC double beam spectrophotometer. To exclude the  
112 adverse effect of oxygen, all experiments were entirely conducted under N<sub>2</sub> atmosphere (two-  
113 hand Atmosbag with zipper lock, Sigma-Aldrich). Blanc samples were measured to correct for  
114 dye adsorption on the TiO<sub>2</sub> catalyst. These samples underwent the same experimental  
115 procedure but were shielded from UV light illumination using aluminum foil.

116 Surface photovoltage measurements were conducted as described by Verbruggen et al.[12]  
117 using a custom made apparatus where the catalyst powder is sandwiched between two ITO  
118 electrodes (Sigma-Aldrich,  $d = 1.2 \text{ mm}$ , resistivity:  $8\text{-}12 \text{ } \Omega \text{ cm}^{-2}$ ), connected to an amplifier ( $1 \times$   
119  $10^6$  voltage amplification). No external bias was applied. For all samples,  $4.0 \pm 0.1 \text{ mg TiO}_2$

120 powder was used in such a way that a controlled area of 5 mm by 5 mm was illuminated with  
121 UVA light ( $\lambda_{\max} = 352 \text{ nm}$ ,  $1.8 \text{ mW cm}^{-2}$  at sample distance).

122

### 123 **2.3 Photocatalytic self-cleaning activity measurement**

124 The photocatalytic self-cleaning test was conducted by means of an in-situ stearic acid (SA)  
125 degradation experiment, based on the method proposed by Paz et al.[31] Silicon wafers (1.5 cm  
126 x 1.5 cm) were ultrasonically cleaned in ethanol and dried with compressed air. Afterwards, a 2  
127 wt%  $\text{TiO}_2$  suspension in absolute ethanol was prepared of which 25  $\mu\text{L}$  was drop casted on the  
128 silicon wafer. The coated wafers were subsequently dried overnight at 363 K. A layer of stearic  
129 acid (Sigma-Aldrich, 98.5%) was applied by spin coating 50  $\mu\text{L}$  of a 0.25 wt% stearic acid solution  
130 in chloroform (Sigma-Aldrich, 99.8%) for 1 minute at 1000 rpm after which the samples were  
131 dried for 15 minutes at 363 K. The photocatalytic degradation of stearic acid was monitored by  
132 means of FTIR spectroscopy using a patented in-situ reaction cell developed in our research  
133 group.[32] This cell enables the in-situ observation of volatile and non-volatile reagents,  
134 intermediates and products on the surface while the reactions are ongoing. The samples were  
135 placed in the center of the reactor at an angle of  $9^\circ$  with the IR beam to minimize internal  
136 reflection effects. During the experiments, IR spectra were recorded from  $4000 - 400 \text{ cm}^{-1}$  at a  
137 resolution of  $1 \text{ cm}^{-1}$  by a Thermo Nicolet 6700 spectrometer (Thermo Fisher Scientific). The  
138 reactor was first flushed with air after which it was sealed airtight and the sample was allowed  
139 to equilibrate for one hour. An average of eight spectra was logged every three minutes using  
140 an automated protocol (MacrosBasic, Thermo Fisher) and the built-in UV LEDs ( $\lambda_{\max} = 377 \text{ nm}$ ,  
141  $1.8 \text{ mW cm}^{-2}$  at sample distance) were switched on. The disappearance of stearic acid was



142 followed by monitoring the integrated area in the wavenumber range 3000 - 2800  $\text{cm}^{-1}$ ,  
 143 corresponding to the asymmetric  $\nu_{\text{as}}(\text{CH}_3)$  in-plane C-H stretch at 2958  $\text{cm}^{-1}$ , an asymmetric  
 144  $\nu_{\text{as}}(\text{CH}_2)$  in-plane C-H stretch at 2923  $\text{cm}^{-1}$  and a symmetric  $\nu_{\text{s}}(\text{CH}_2)$  in-plane C-H stretch at 2853  
 145  $\text{cm}^{-1}$ . [33] The concomitant formation of  $\text{CO}_2$  and CO was monitored following the peak height  
 146 of the asymmetric stretch  $\nu_{\text{as}}(\text{CO}_2)$  at 2360  $\text{cm}^{-1}$  and the stretch  $\nu(\text{CO})$  at 2179  $\text{cm}^{-1}$  respectively.

147

### 148 3. RESULTS AND DISCUSSION

#### 149 3.1 Morphological parameter characterization

150 An overview of the main important morphological material properties is given in Table 1. The  
 151 surface area was determined by means of  $\text{N}_2$  adsorption/desorption measurements by applying  
 152 the BET method. XRD measurements were used to determine the crystallinity/crystal phases as  
 153 well as the primary crystallite size.

154 **Table 1.** Overview of important morphological properties of the three commercial  $\text{TiO}_2$  sources.

$\text{TiO}_2$ source	Surface area ( $\text{m}^2 \text{g}^{-1}$ )	Primary crystallite size <sup>a</sup> (nm)	Crystallinity
P25	59	18.8	85.6 wt% anatase 14.4 wt% rutile <sup>b</sup>
P90	125	10.9	89.3 wt% anatase 10.7 wt% rutile <sup>b</sup>
PC500	352	6.5	85 wt% anatase 15 wt% amorphous <sup>c</sup>

155 <sup>a</sup> Primary crystallite size of the dominant crystal phase (anatase), determined using the Scherrer equation.

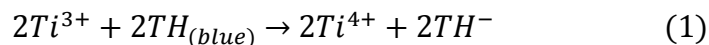
156 <sup>b</sup> Determined with the formula of Spurr and Myers using the intensity of the strongest anatase and rutile peaks,  
 157 being 25.28° and 27.4° 2 $\theta$ . [34]

158 <sup>c</sup> As obtained from the manufacturer data.

## 159 **3.2 Electronic parameter characterization**

### 160 *Determination of defect states using electron titration method*

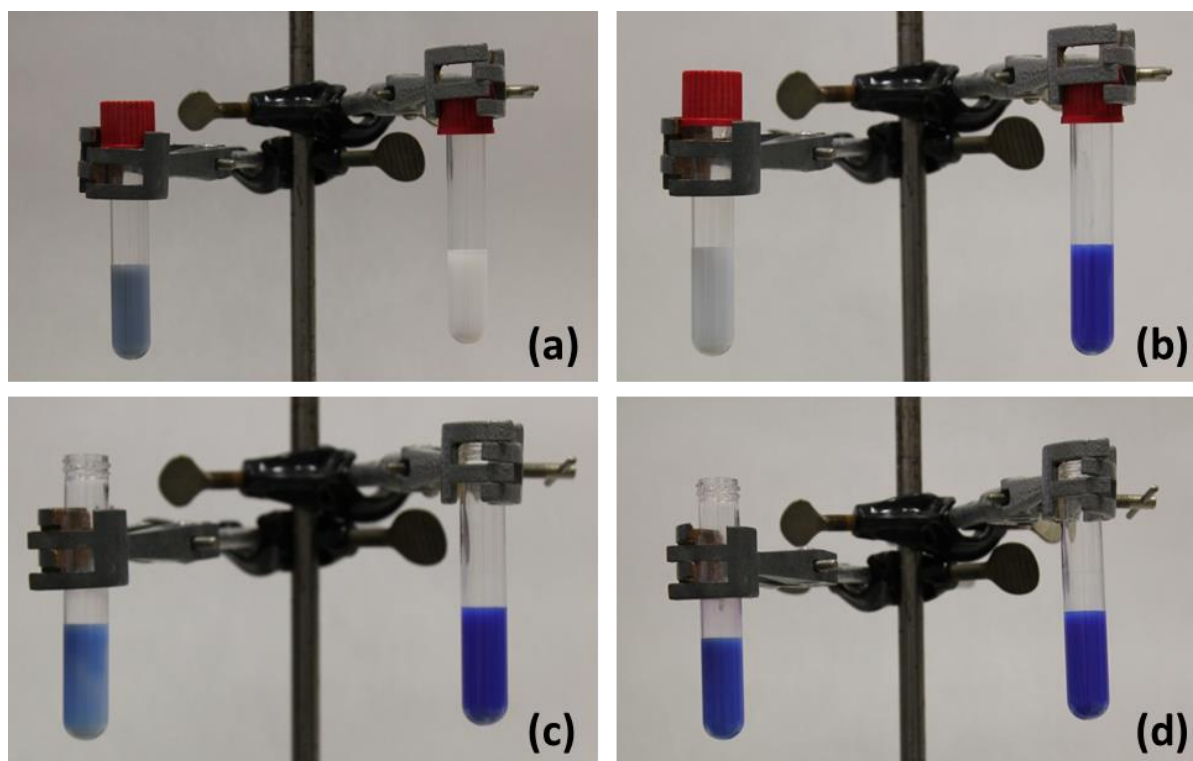
161 As indicated in the introduction, it is our intent to use simple and inexpensive  
162 characterization procedures without having to compromise on accuracy and precision. To  
163 probe the electronic properties of the catalysts we therefore propose a modified electron  
164 titration technique that enables quantitative analysis of defect sites, thereby gaining more  
165 information on the electron-hole pair ( $e^-h^+$ ) recombination process. The method is based on  
166 the light-induced accumulation of electrons on  $TiO_2$  powder in a de-aerated solution in the  
167 presence of a hole scavenger. In Figure 1 (a) equal amounts of  $TiO_2$  powder are suspended in  
168 de-aerated ethanol. The left reaction vessel was illuminated by UV light, while the right vessel  
169 was shielded from illumination using aluminum foil. Charge separation takes place in the  
170 illuminated sample where the holes are scavenged by the organic solvent, causing the electrons  
171 to be trapped at  $Ti^{4+}$  forming  $Ti^{3+}$ , i.e. the electrons accumulate on the  $TiO_2$  powder in the form  
172 of  $Ti^{3+}$ . [27] This can be verified macroscopically by the appearance of a distinct grayish-blue  
173 color (Figure 1 (a)). It is possible to determine the concentration of these accumulated electrons  
174 using thionine (TH) as acceptor molecule. Thionine exists as a monovalent cation and is readily  
175 reduced to its colorless *leuco* form ( $TH^{2-}$ ) via a two-step reduction, as can be derived from  
176 reactions (1) and (2). [28,29]



177 In order for this reaction to occur spontaneously, the reductive power of the trapped  
178 electrons should be strong enough, i.e. the defective sites should have a more negative  
179 electrochemical potential than the TH/TH<sup>2-</sup> redox couple. The redox potential of the dye used in  
180 this study is +0.064 V vs. NHE.[29] Assuming the accumulated electrons are trapped just below  
181 the conduction band edge of TiO<sub>2</sub> (-0.52 V vs. NHE[35]), the stored electrons will indeed react  
182 readily with the thionine molecules, reducing them to their colorless form (reactions (1) and  
183 (2)). This can be seen in Figure 1 (b). A known amount of a de-aerated thionine dye solution is  
184 added to both reaction vessels. In the previously irradiated TiO<sub>2</sub> suspension (left), the stored  
185 electrons reduce the dye to its colorless form, causing the grayish-blue color (indicative of  
186 electron storage) to disappear. The non-illuminated reaction vessel (right) attains the dark blue  
187 color of the thionine dye solution. Using the extinction band of thionine at 602.5 nm ( $\epsilon_{\text{thionine}}$  in  
188 a five-to-one ethanol/water mixture is determined to be 81800 M<sup>-1</sup> cm<sup>-1</sup>), it is possible to  
189 determine the amount of thionine molecules that are reduced. The reader is referred to Figure  
190 S1 in the supplementary information for the UV-VIS absorption spectra of the thionine dye  
191 before and after reduction to its colorless form.

192 Keeping in mind that two electrons are needed to bleach one thionine molecule, the amount  
193 of trapped electrons can thus be calculated. It is important to note that both the stored  
194 electrons on TiO<sub>2</sub> and the reduced *leuco* form of the dye are highly stable under inert  
195 atmosphere but are easily oxidized by coexisting oxygen in the solution (Figure 1 (c) and (d)).  
196 Even small amounts of O<sub>2</sub> present during the measurement would result in quenching of the  
197 signal and consequently lead to large experimental errors. The errors on the values we  
198 obtained are quite low (Figure 2), demonstrating the high accuracy and reproducibility of the

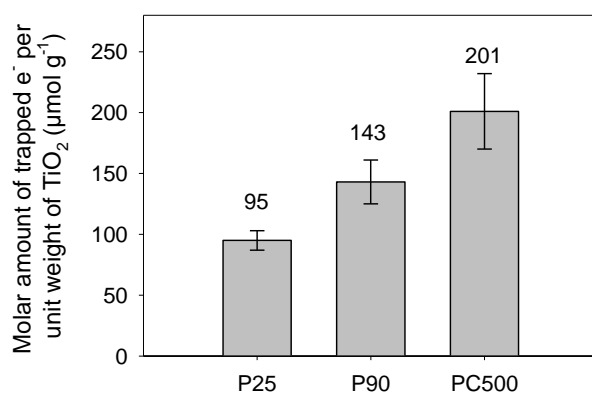
199 electron titration method. Our values are also in the same order of magnitude as the ones  
200 found in literature by applying a comparable analysis.[13,23,27] For instance, Prieto-Mahaney  
201 et al.[13] determined the molar amount of defective sites for commercially available P25 to be  
202  $59 \mu\text{mol g}^{-1}$ , which is somewhat lower than the value of  $95 \mu\text{mol g}^{-1}$  that we obtained. They  
203 used triethanolamine as hole scavenger and methyl viologen as titrant. The redox potential of  
204 methyl viologen is  $-0.45 \text{ V vs. NHE}$  (as compared to  $+0.064 \text{ V vs. NHE}$  for thionine) so it is likely  
205 that the reductive power of some deeper trapped electrons is not sufficient to reduce the  
206 methyl viologen, resulting in a slightly lower value than the one obtained in our study. In  
207 general, such a high number of defects can be largely attributed to the co-existence of different  
208 crystalline phases in the sample and its nanostructured nature.



209  
210 **Figure 1.** Overview of the different steps taking place during the photocatalytic electron titration experiment. (a)  
211 Equal amounts of  $\text{TiO}_2$  were suspended in de-aerated ethanol. The left reaction vessel was illuminated with UV

212 light causing the electrons to accumulate on the powder and the suspension to turn grayish-blue. (b) Situation  
213 after addition of equal amounts of a de-aerated aqueous thionine solution. The stored electrons in the left  
214 reaction vessel reduce the dye to its colorless form. In the right reaction vessel no dye reduction takes place, thus  
215 the suspension gets the dark blue color of the stock thionine solution. (c) Effect of the presence of oxygen five  
216 minutes after opening the reaction vessel. The colorless *leuco*-thionine (left reaction vessel) reverts to its original  
217 colored form. (d) Effect of the presence of oxygen 20 minutes after opening the reaction vessel.

218 From our analysis it can be derived that the capacity for electron trapping follows the order  
219 PC500 > P90 > P25 (Figure 2). A study by Ikeda et al.[27] suggests that the formed  $Ti^{3+}$  species  
220 are produced by trapping electrons at  $-$  and thus quantify the number of  $-$  defect sites in  $TiO_2$ .  
221 Since electron-hole pair recombination is facilitated by defects,[8] it can be stated that the  
222 electron titration method directly yields information on the recombination process and  
223 therefore also on the charge carrier formation efficiency. They demonstrated a linear  
224 correlation between the rate of  $e^-h^+$  pair recombination and the molar amount of defective  
225 sites per unit weight of  $TiO_2$  powder. Considering our results, this implies that P25 is less prone  
226 to charge carrier recombination than P90 and PC500 and thus is the most favorable  
227 photocatalyst in terms of electronic properties.

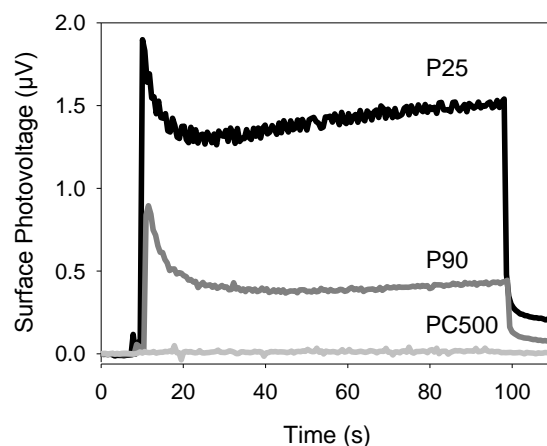


228

229 **Figure 2.** Molar amount of trapped electrons per unit weight for the three commercial TiO<sub>2</sub> powders.

### 230 *Surface photovoltage (SPV) measurements*

231 Surface photovoltage (SPV) measurements enable a detailed characterization of  
232 semiconductors yielding complementary information on the electronic properties. The  
233 technique is based on an illumination-induced change in the surface photovoltage.[36] Contact  
234 between a semiconductor and a conductor leads to the formation of a non-neutral space  
235 charge region near the surface that in turn gives rise to a built-in electric field, commonly  
236 referred to as the surface potential barrier. When the semiconductor is illuminated by UV light,  
237 free charge carriers are generated which will redistribute because of the built-in electric field.  
238 This will alter the surface potential barrier. The difference in surface potential before and  
239 during illumination is defined as the SPV signal.[12,37] The amount of charge carriers formed is  
240 directly related to the potential difference and thus the SPV signal. Therefore, SPV  
241 measurements provide valuable information on the charge carrier formation efficiency of the  
242 semiconductor. The technique has many similarities to the aforementioned electron titration  
243 procedure as it is also fast, simple, cheap, directly applicable to powders and does not require  
244 any sample preparation. Therefore, SPV is an ideal candidate to corroborate and further  
245 complement the results of our adapted electron titration method. The SPV signal of the three  
246 commercial TiO<sub>2</sub> sources can be found in Figure 3.



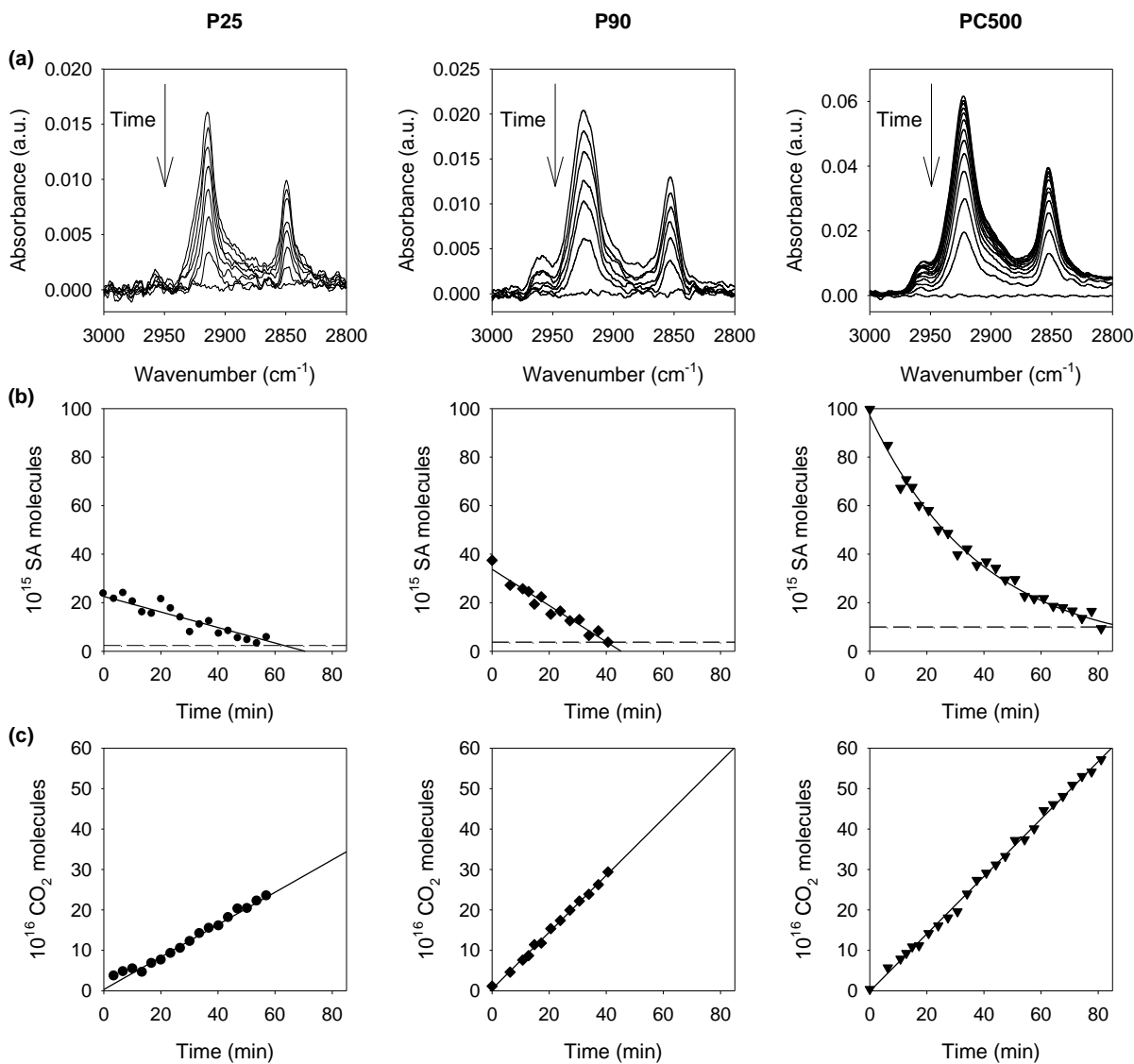
247  
 248 **Figure 3.** Surface photovoltage measurements for P25 (black), P90 (dark gray) and PC500 (light gray). Light was  
 249 switched on after 10 s and turned off at 100 s (sample was illuminated for 90 s).

250 P25 displays the highest SPV signal and thus the best charge carrier formation efficiency  
 251 whereas the signal of PC500 is negligible. These results are in excellent agreement with the  
 252 results obtained by the electron titration method (*vide supra*) according to which P25 has the  
 253 least amount of defect states and is therefore less susceptible to charge carrier recombination.  
 254 PC500 on the other hand only generates a poor amount of stable charge carriers, as it is more  
 255 prone to charge carrier recombination (evidenced by the electron titration experiment),  
 256 probably due to the large amount of non-crystalline material present in the sample (Table 1).

### 257 **3.3 In-situ investigation of the photocatalytic self-cleaning activity**

258 The degradation of a solid layer of stearic acid (SA) is a commonly accepted method to assess  
 259 the self-cleaning photocatalytic activity.[31,38–41] SA is a photo-stable model compound for  
 260 organic fouling on both interior and exterior surfaces. In this method, a thin layer of SA is  
 261 deposited onto the photocatalyst and its destruction over time is monitored by means of FTIR  
 262 since the molecule absorbs strongly in the region from 3000 to 2800  $\text{cm}^{-1}$ . Complete

263 mineralization of SA to CO<sub>2</sub> can be achieved as was evidenced by Mills and Wang who followed  
 264 the simultaneous decrease of SA and increase in CO<sub>2</sub>. [33] In the present research, the  
 265 photocatalytic degradation of SA was monitored using a reaction cell developed in our research  
 266 group that enables the in-situ monitoring of the catalyst surface, while the reaction is ongoing.  
 267 The decrease of the FTIR absorption band of SA from 3000 - 2800 cm<sup>-1</sup> during UV illumination  
 268 corresponding to the decrease of number of SA molecules is shown for the three commercial  
 269 TiO<sub>2</sub> sources as a function of illumination time in Figure 4 (a) and (b).



270

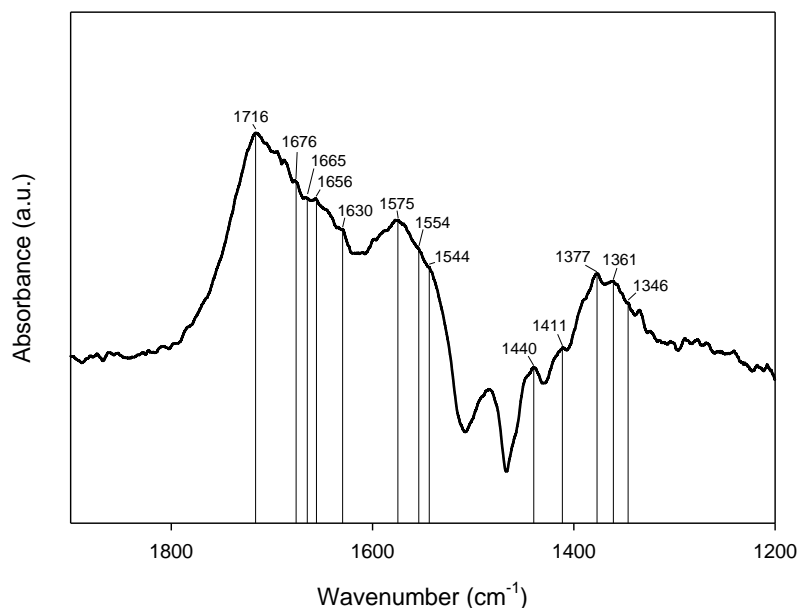


271 **Figure 4.** (a) Decrease of FTIR absorption bands of SA in the range of 3000 - 2800  $\text{cm}^{-1}$  during UV illumination. (b)  
272 Decrease of SA molecules over time, determined from the integrated area from 3000 - 2800  $\text{cm}^{-1}$  under the  
273 corresponding curves of Figure 1 (a). The values are shown until only 10% of the original SA concentration was  
274 present (dashed curve). The kinetic models (solid lines) are also based on the first 90% of conversion. (c)  $\text{CO}_2$   
275 formation in function of illumination time during the first 90% of SA conversion.

276 Since SA appears as a transparent coating over the photocatalyst, light absorption by the  
277 pollutant can be neglected in the kinetic pathway. Depending on the porosity of the  
278 photocatalyst layer, zero or first order kinetics can be expected. It is shown by Ollis that for  
279 non-porous catalyst layers a zero order kinetic model satisfactorily describes the SA  
280 degradation rate, while in the case of a porous catalyst layer, apparent first order behavior can  
281 be expected.[42] Both situations apply in our results, as can be seen in Figure 4 (b). P25 and P90  
282 films exhibit zero order kinetics while PC500 clearly shows first order behavior. Still, both  
283 sample preparation and activity measurement were identical for all three  $\text{TiO}_2$  sources. The  
284 PC500 film, however, appeared to be much coarser, consisting of aggregated islands, whereas  
285 P25 and P90 layers were much more uniform. This intrinsic difference in overall layer  
286 morphology can explain the different kinetic behavior of the PC500 sample that acts as a  
287 porous film, leading to first order kinetics. Despite the difference in kinetic order, in terms of  
288 the disappearance rate of SA, it is clear that the PC500 film is most active while P25 exhibits the  
289 lowest photocatalytic self-cleaning activity.

290 To enable a more appropriate, quantitative, comparison of the full mineralization activity, the  
291  $\text{CO}_2$  formation was also studied by monitoring the peak height of the asymmetric stretch  
292  $\nu_{as}(\text{CO}_2)$  at 2360  $\text{cm}^{-1}$  over time. These results are shown in Figure 4 (c). It can be seen that the  
293  $\text{CO}_2$  formation rate increases linearly with illumination time for all three samples. P25 is the

294 least active catalyst, which is in agreement with the SA removal results. Remarkably, there is no  
295 longer any difference in activity between P90 and PC500. The PC500 sample also shows a clear  
296 linear relationship between CO<sub>2</sub> formation and illumination time, as opposed to its first order  
297 SA removal kinetics. This indicates that for these PC500 films the initial oxidation process  
298 proceeds fast but gives rise to long lived intermediates which are harder to degrade. Minabe et  
299 al. studied the photocatalytic decomposition of SA by TiO<sub>2</sub> and found that they could only  
300 remove 69% of the original amount of SA, also pointing to the formation of intermediates that  
301 are less easily degraded.[43] Since the in-situ reaction cell used in our experiments enables to  
302 monitor the surface while the reactions are ongoing, the FTIR spectra were carefully examined  
303 to see whether some intermediates could be observed. Figure 5 shows the FTIR spectrum of  
304 PC500 in the range of 1900 to 1200 cm<sup>-1</sup> after 90% of the SA was converted. The negative  
305 absorbance from 1530 to 1430 cm<sup>-1</sup> is attributed to the disappearance of stearic acid from the  
306 surface. It is evident from this spectrum that different intermediates like acetaldehyde,  
307 formaldehyde and their oxidation products (acetate, formate) are indeed present on the  
308 surface. Assignment of the different bands can be found in Table 2. These results are in good  
309 agreement with those obtained by Roméas et al.[44] who studied the photomineralization of  
310 palmitic acid by TiO<sub>2</sub> coated self-cleaning glass. They found there was a negative deviation of  
311 the expected reaction stoichiometry due to the formation of intermediates like acetone,  
312 acetaldehyde and formaldehyde.



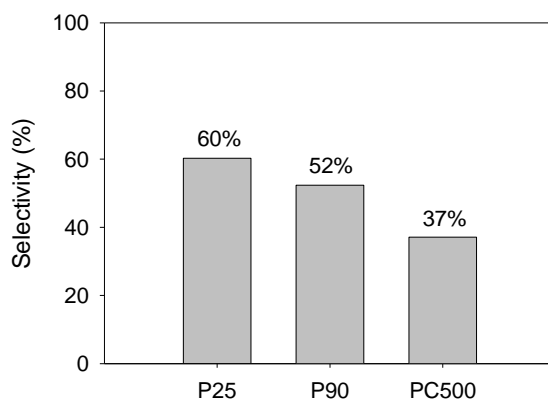
- 313
- 314 **Figure 5.** IR spectrum of PC500 (after H<sub>2</sub>O correction) after 90% of SA was removed by UV light illumination.
- 315 **Table 2.** Assignment of the FTIR bands observed on the surface of PC500 after 90% of SA was removed.

Adsorbed molecule	Vibration mode	Frequency (cm <sup>-1</sup> )	
		Literature[45]	This work
HCOOH <sub>ad</sub>	$\nu(\text{C}=\text{O})$	1713	1716
CH <sub>3</sub> COOH <sub>ad</sub>	$\nu(\text{C}=\text{O})$	1675	1676
HCHO <sub>ad</sub>	$\nu(\text{C}=\text{O})$	1662	1665
CH <sub>3</sub> CHO <sub>ad</sub>	$\nu(\text{C}=\text{O})$	1653	1656
CH <sub>3</sub> CHO <sub>ad</sub>	$\nu(\text{C}=\text{C})$	1625	1630
HCOO <sup>-</sup> <sub>ad</sub>	$\nu_{\text{as}}(\text{COO})$	1574	1575
HCOO <sup>-</sup> <sub>ad</sub>	$\nu_{\text{as}}(\text{COO})$	1554	1554
CH <sub>3</sub> COO <sup>-</sup> <sub>ad</sub>	$\nu_{\text{as}}(\text{COO})$	1547	1544
CH <sub>3</sub> CHO <sub>ad</sub>	$\delta_{\text{as}}(\text{CH}_3)$	1444	1440
HCOO <sup>-</sup> <sub>ad</sub>	$\delta(\text{CH})$	1416	1411

$\text{HCOO}^-_{\text{ad}}$	$\nu_s(\text{COO})$	1378	1377
$\text{HCOO}^-_{\text{ad}}$	$\nu_s(\text{COO})$	1357	1361
$\text{CH}_3\text{COO}^-_{\text{ad}}$	$\delta_{\text{as}}(\text{CH}_3)$	1343	1346

316

317 Complete mineralization of one SA molecule yields 18 molecules of  $\text{CO}_2$ . Since some long-  
 318 lived intermediates are formed in our experiments, the SA: $\text{CO}_2$  ratio will be lower than the  
 319 reaction stoichiometry value of 1:18 i.e. the selectivity toward  $\text{CO}_2$  formation will deviate from  
 320 unity. This drop in selectivity has also been determined in our experiments and is plotted in  
 321 Figure 6. Even though PC500 is the most active catalyst to remove a solid layer of SA, it has the  
 322 lowest selectivity toward  $\text{CO}_2$  formation (37%). P25, the least active catalyst, on the other hand  
 323 has a selectivity of 60%, meaning that 11  $\text{CO}_2$  molecules are formed per SA molecule. Mills and  
 324 Wang also tested the SA degradation capacity of P25 and observed the same drop in the SA: $\text{CO}_2$   
 325 stoichiometric ratio from 1:18 to 1:12 (corresponding to a selectivity of 69%) after ca. 50  
 326 minutes of irradiation.[33] This value matches quite well with the selectivity obtained for P25 in  
 327 our study.



328

329 **Figure 6.** Selectivity toward CO<sub>2</sub> formation for the three tested samples. A selectivity of 100% means full  
330 mineralization of SA to CO<sub>2</sub> is achieved resulting in a SA:CO<sub>2</sub> ratio of 1:18.

### 331 **3.4 Discussion on the structure-activity-selectivity relation**

332 Based on the characterization of the electronic photocatalyst properties (electron titration +  
333 SPV), it is expected that P25 shows the highest photocatalytic activity since its (photo)electronic  
334 properties outperform those of P90 and PC500.[11,12] It has the lowest amount of  
335 defect/recombination sites and the highest SPV value, indicating a good charge carrier  
336 formation efficiency and subsequently high photocatalytic activity. This can be explained by the  
337 fact that P25 is characterized by a high quantum efficiency because of its fully crystalline nature  
338 (few surface and bulk defects[11]), as opposed to PC500 (15 wt% amorphous). In addition, the  
339 superior activity of commercial P25 is often attributed to its polymorphic composition (85 wt%  
340 anatase, 15 wt% rutile) that ameliorates charge separation due to the favorable position of the  
341 valence and conduction bands of both crystal phases. P90 is also fully crystalline but has a  
342 different anatase to rutile ratio (90 wt% anatase, 10 wt% rutile) that deviates more from the  
343 optimal composition found by Su et al. (60 wt% anatase and 40 wt% rutile).[46] In terms of  
344 morphological properties on the other hand, PC500 stands out since it has the smallest primary  
345 particle size and the highest surface area (6 times higher than P25, 3 times higher than P90).

346 Linking both morphological and electronic material properties to the ability to convert a solid  
347 layer of stearic acid for the three commercial TiO<sub>2</sub> sources, it can be concluded that not the  
348 electronic properties but mainly the high surface area is the driving factor for self-cleaning  
349 applications. P25 clearly outperforms P90 and PC500 in terms of charge carrier formation  
350 efficiency but has the lowest activity. PC500 on the other hand displays the largest surface area

351 and also has the highest photocatalytic activity. These results agree well with an earlier study  
352 by our group where PC500 outperforms P90 and P25 for the degradation of acetaldehyde in  
353 air.[12] Interestingly, the trend is reversed for photocatalytic activity measurements in aqueous  
354 phase, where P25 excels over P90 and PC500. During gas phase measurements, the (small)  
355 gaseous pollutants can access the entire surface whereas pollutants in aqueous phase are  
356 hindered by diffusion limitations. A large surface area consisting of small pores is thus only  
357 desirable when mass transfer limitations can be neglected.[12] The consistency between the  
358 driving factors of gas phase photocatalysis and self-cleaning applications can be rationalized  
359 well: The photocatalytic oxidation of organic solids can be considered as a special border case  
360 of gas phase photocatalysis, as the reactions take place at the solid (catalyst) - solid (pollutant) -  
361 gas (ambient) interface. In addition, the supply of O<sub>2</sub> and the removal of CO<sub>2</sub> follow the same  
362 mass transfer laws as those of gas phase reactions, without being obstructed by a liquid barrier  
363 giving rise to diffusion limitations.[7]

364 Next to activity, selectivity is also an important catalyst property. When looking at the ability  
365 to fully mineralize the pollutant, i.e. obtaining high selectivity toward CO<sub>2</sub>, the trend in  
366 photocatalytic activity (PC500 > P90 > P25) is reversed! It can indeed be concluded from Figure  
367 6 that P25 has the highest mineralization efficiency (60%), followed by P90 (52%) and PC500  
368 (37%). Interestingly, this reversed trend corresponds well with the trend observed in the  
369 electronic parameter characterization (P25 > P90 > PC500). It can thus be concluded that in  
370 terms of photocatalytic self-cleaning *activity*, the morphological properties present the driving  
371 parameters. If we however take a look at mineralization *selectivity*, the electronic properties  
372 play a more prominent role. PC500 (largest surface area, least favorable electronic properties)

373 thus has the highest self-cleaning activity i.e. can initially convert the most SA molecules  
374 because of the high amount of active sites. The initial conversion however gives rise to some  
375 long-lived intermediates on the surface that prove to be more difficult to degrade. P25 on the  
376 other hand (lowest specific surface area, best electronic properties) has less active sites so can  
377 convert less SA molecules, but because of its desirable electronic properties is able to  
378 mineralize the pollutant more selectively to CO<sub>2</sub>.

379

#### 380 **4. CONCLUSION**

381 In order to properly assess the photocatalytic properties of self-cleaning TiO<sub>2</sub>, rigorous multi-  
382 aspect characterization is key since differences in macroscopic activity measurements cannot  
383 always be attributed to a single material property. For this reason, we conducted a full  
384 morphological and electronic characterization of three commercial TiO<sub>2</sub> photocatalysts (P25,  
385 P90, PC500) to obtain a coherent structure-activity and -selectivity relation for self-cleaning  
386 applications. Our results indicate that P25 has the most desirable electronic properties  
387 compared to P90 and PC500 but is outperformed by PC500 and P90 in terms of morphological  
388 properties. These favorable morphological properties were shown to lead to a high  
389 photocatalytic self-cleaning activity. The opposite is true when looking at selectivity, where  
390 good electronic properties are the predominant factor. The obtained structure-activity-  
391 selectivity relation provides more mechanistic insight in the self-cleaning behavior of  
392 photocatalytic surfaces. We are hopeful this will boost the development of high-performant  
393 photocatalytic self-cleaning materials and facilitate a rapid transition from the laboratory scale  
394 to commercial applications.

395

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