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# ZnAl layered double hydroxide based catalysts (with Cu, Mn, Ti) used as noble metal-free three-way catalysts

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## Abstract

This research presents a novel approach for developing noble metal-free three-way catalysts by using ZnAl layered double hydroxide (LDH) co-precipitated with Mn and Ti. Cu was added as the catalytically active metal. Two methods were explored, namely the addition of Cu during the co-precipitation step and a post-synthesis method using the LDH memory effect. In case the starting material had LDH characteristics high amounts of CuO were adsorbed on the support and high dispersion degrees of CuO were obtained. A four-cycle three-way catalysis test was used for evaluation. The smaller CuO particle size resulted in better performance for oxidation reactions. The addition of Mn had a positive effect on the general performance of the catalysts, while the presence of Ti mainly improved the NO conversion.

26 The developed materials showed good stability in consecutive catalytic testing cycles and even  
27 show some NO conversion under stoichiometric conditions. The developed CuO-based ZnAl  
28 layered double hydroxide-based materials are very promising catalysts for Three-way  
29 catalysis, allowing to reduce the precious metal content compared to the classical catalysts  
30 composition.

## 31 Keywords

32 Layered Double Hydroxides, memory effect, three-way catalysts, copper oxide, NO reduction,  
33 CO oxidation

## 34 Highlights

- 35 • High loadings of well-dispersed CuO nanoparticles on layered double hydroxide (LDH)
- 36 • Superior catalytic performance due to well-dispersed CuO nanoparticles
- 37 • Amorphous aluminium oxide in the LDH material works as support dispersing CuO
- 38 • Increased performance with Mn addition and NO conversion with Ti addition

## 39 Introduction

40 While the technology for the production of alternatively powered vehicles has greatly  
41 advanced in the past few years and the sales of diesel cars continues to fall, only 10.5 % of the  
42 passenger cars registered in 2020 in the European Union relied only on electricity (European  
43 Automobile Manufacturers Association, 2021). Petrol cars are equipped with a three-way  
44 catalytic converter whose function is to remove CO, NO<sub>x</sub> and hydrocarbons. The key  
45 components of current three-way converters are the precious metals Pt, Pd, and Rh (Kašpar  
46 et al., 2003; Kumar et al., 2017). These materials are however considered as critical raw  
47 materials (CRM) because of their great economic importance, while their supply is at risk. This

48 means that their use in new technologies must be limited as much as possible (European  
49 Commission, 2018). This also causes a revived interest in the use of alternative materials to  
50 replace the CRMs.

51 Although having a reaction rate of about 50 times lower than Pd for CO (Kummer, 1980), CuO  
52 is the preferred catalyst to focus on as it is not a CRM (European Commission, 2018). While  
53 lower reaction rates could be compensated by using larger amounts of CuO, a good dispersion  
54 is essential for the catalytic activity of CuO, which is challenging to achieve with larger amounts  
55 on a given surface area (Bennici et al., 2004). At the same time, large amounts of CuO can  
56 agglomerate easier when exposed to the high temperatures of the exhaust gas. In order to  
57 avoid this, it is important to design a catalyst with a high surface area, while having highly  
58 dispersed CuO deposited on the surface of the support. Another way of increasing or changing  
59 the catalytic properties of a material is by the addition of a co-catalyst material. Specific metal  
60 oxides can work together in a synergetic system, leading to a better performance than the  
61 sum of the separate materials (Cavani et al., 1991; Weng and Delmon, 1992; Liu et al., 2019).  
62 These mixtures of metal oxides (MO) will be exposed for a long time at high temperatures,  
63 under different gas mixtures, and the formation of mixed metal oxides (MMO), such as spinels  
64 and perovskites, is likely to occur. This can directly influence the performance of the catalyst  
65 (Cousin and Ross, 1990).

66 In this work, layered double hydroxide (LDH), with different compositions, are synthesized and  
67 used as precursors to further tune three-way catalyst compositions of MMOs. MMOs derived  
68 from LDH are particularly interesting in catalytic applications due to their large surface area,  
69 homogeneous distribution of the different components and basic character (Cavani et al.,  
70 1991; Gennequin et al., 2010; Kowalik et al., 2013; Mrad et al., 2015). The LDH structure

71 consists of positively charged hydroxide layers, resulting from the mixture of divalent (such as  
72  $Mg^{II}$  or  $Zn^{II}$ ) and trivalent (such as  $Al^{III}$ ) or tetravalent atoms (such as  $Ti^{IV}$ ) (Saber and Tagaya,  
73 2003, 2008; E. M. Seftel et al., 2008; Li et al., 2017) with interstitial layers of anions (such as  
74  $CO_3^{2-}$ ) that compensate for the positive charge of the metal hydroxide layers (Gennequin et  
75 al., 2010). Furthermore, water molecules are also located in the interstitial layers (Cavani et  
76 al., 1991). One of the most interesting properties of the LDH is that after calcination, the  
77 resulted MMO-type material is able to reconstruct the original LDH structure as soon as the  
78 materials are brought into contact with water and an anion source. The phenomenon is often  
79 referred to as “the memory effect” (Cavani et al., 1991; Marchi and Apesteguía, 1998; Erickson  
80 et al., 2005; Carja et al., 2010; Gennequin et al., 2010; Kowalik et al., 2013).

81  $Cu(O)$  was used as the main active material and was either inserted in the material during the  
82 initial LDH synthesis via co-precipitation or afterwards in an impregnation step by making use  
83 of the “memory effect” property of LDH.

84 While the most common LDH consist of Mg and Al (Cavani et al., 1991), this research is based  
85 on Zn and Al LDH in a 2:1 ratio. Here, Zn was chosen mainly for its structural role in the LDH  
86 synthesis as well as its beneficial synergistic effect for CO oxidation when used together with  
87 Cu and Ti (Saber and Zaki, 2014). Zn was partially substituted by Cu in a molar ratio of 1:1, as  
88 this is the maximal amount of Cu that could be isomorphously substituted within the hydroxyl  
89 layers of LDH. Promising research results for CO oxidation (Saber and Zaki, 2014) and NO  
90 reduction (Yuan et al., 2013, 2014; Chen et al., 2014) were reported by using LDH based  
91 catalysts featuring Ti. Therefore, several compositions of LDH containing Ti were further  
92 prepared by the isomorphous substitution of trivalent  $Al^{III}$  atoms with tetravalent  $Ti^{IV}$  atoms  
93 in order to improve the performances of the three-way MMO catalysts. Furthermore,

94 different formulations with Mn were synthesized, based on the promising research towards  
95 NO<sub>x</sub> abatement of the Mn containing LDH (Obalová et al., 2007, 2009; Li et al., 2010, 2012; Li  
96 et al., 2014; Pacultová et al., 2016).

97 The aim of this work is two-fold. Firstly, to show how LDH can be used as a basis for the  
98 development of Cu-based three-way catalysts and that the reconstructive properties of LDH  
99 can be utilized to further optimise CuO nanoparticles on the surface of the material as  
100 opposed to the traditional co-precipitation. Secondly, to assess how different metals (Mn, Zn,  
101 Al, Ti) influence the catalytic performance of CuO by interacting with it or by forming mixed  
102 metal oxides.

## 103 Experimental

### 104 Catalyst preparation

105 Two synthesis strategies were explored for the preparation of copper-containing three-way  
106 catalysts, e. g. direct co-precipitation at constant pH and reconstruction route by taking  
107 advantage of the memory effect property of the LDH, respectively. The commercial  
108 benchmark reference catalysts (0.5% Pd/CeZrO<sub>2</sub> and “TWC”, a grounded commercial three-  
109 way catalyst) have been supplied by Johnson Matthey, UK.

#### 110 a) Co-precipitation method

111 The initial materials were synthesized by a common co-precipitation method for LDH, at fixed  
112 pH. In a typical synthesis, 120 mL of a solution of the mixed metal salts was prepared by  
113 dissolving the metal salts listed in the supplementary information in the appropriate amount  
114 of water for a total concentration of 0.5 M. The molar ratios of the metals were according to  
115 **¡Error! No se encuentra el origen de la referencia..** The resulting mixed metals solution was

116 mixed with 40 mL of a 1 M Na<sub>2</sub>CO<sub>3</sub> solution and was added dropwise and simultaneously with  
 117 a 2 M NaOH solution to a beaker while stirring. The flow of the two solutions was regulated  
 118 as such to maintain the pH value at 7.5. The obtained precipitate was aged under stirring for  
 119 24 h at room temperature. The obtained precipitate was collected by vacuum filtration,  
 120 washed thoroughly with water and dried overnight in an oven at 80°C(E. M. Seftel et al., 2008;  
 121 E. Seftel et al., 2008).

122 For simplicity reasons, the synthesized materials are labelled based on the elements in their  
 123 composition. The names of the samples and the ratios between the elements are displayed in  
 124 Table 1.

Sample name	Elements	Atomic ratio
ZA	Zn, Al	2:1
ZCA	Zn, Cu, Al	1:1:1
ZAT	Zn, Al, Ti	4:1:1
ZCAT	Zn, Cu, Al, Ti	2:2:1:1
MZAT	Mn, Zn, Al, Ti	2:2:1:1
MCAT	Mn, Cu, Al, Ti	2:2:1:1
MA	Mn, Al	2:1
MT	Mn, Ti	2:1

125 *Table 1: Composition details of the catalysts prepared via the co-precipitation method.*

126 **b) Reconstruction method**

127 By following the same procedure described above, copper-free LDH were also prepared and  
 128 further used to explore the second synthesis strategy based on the memory effect.

129 These materials were calcined at 400°C for 4 h (ramping speed of 1°C/min), followed by a slow  
130 cooling down to room temperature. 1 g of these MMOs were suspended into 250 mL of 0.1  
131 M copper acetate solution under stirring for 30 minutes to allow the reconstruction of the  
132 initial layered structure and intercalate the copper acetate anions. Finally, the solid fraction  
133 was separated by centrifugation, washed thoroughly with water and dried overnight in an  
134 oven at 80°C (Carja et al., 2013).

135 The materials after calcination are additionally labelled with the temperature details (for  
136 example “-400”). The materials that were impregnated with Cu according to the  
137 reconstruction method are indicated with “C/” before their name.

138 Finally, all the copper-loaded materials, as prepared by the two strategies as described above,  
139 were calcined at 600°C for 4 h (ramping speed of 1°C/min) before their use in the catalytic  
140 experiments.

#### 141 [Physicochemical characterisation](#)

142 The materials were characterized by X-ray powder diffraction (XRD) using an X-ray  
143 diffractometer (PANalytical X’Pert PRO MPD) equipped with CuK $\alpha$  radiation ( $\lambda = 0.15406$  nm).  
144 The analysis was performed from  $2\theta = 5 - 70^\circ$  with a scanning speed of 0.04°/4s. The average  
145 crystallite size was estimated using the Scherrer equation, which is explained in the  
146 supplementary information.

147 N<sub>2</sub>-physisorption was carried out at -196°C on a Quantachrome Quadrasorb SI automated gas  
148 adsorption system. Prior to the measurements, the samples were outgassed at 60°C for 16 h  
149 except for the materials calcined at 400 or 600°C. These were outgassed at 200°C for 16 h. The  
150 specific surface area was calculated using the Brunauer-Emmet-Teller (BET) equation. The



151 Barret-Joyner-Halenda (BJH) method was applied to estimate the pore size distribution. The  
152 total pore volume was determined at  $P/P_0= 0.95$ .

153 SEM-EDX was used to determine the surface composition, using a FEG-ESEM-EDX, FEI Quanta  
154 250 by scanning the whole field instead of discrete points.

155 TEM characterization of CuO nanoparticles was performed by high-angle annular dark field  
156 scanning transmission electron microscopy (HAADF-STEM) and energy dispersive X-ray  
157 spectroscopy (EDS) using an aberration-corrected cubed FEI Titan microscope equipped with  
158 a Super X EDS detector operating at 300 kV.

### 159 Catalytic activity measurements

160 The catalytic tests were performed in a quartz fixed bed reactor. In a typical experiment, 0.200  
161 g of the powder catalyst (100 - 200  $\mu\text{m}$ ) was mixed with 0.200 g silicon carbide to avoid  
162 limitations related to mass and heat transfer phenomena. The catalytic performance was  
163 evaluated from the conversions of the different gases in a multi-cycle temperature-  
164 programmed experiment. The course of the temperature as a function of time during the  
165 experiment is illustrated in Supplementary Figure 1, in a relevant time-scale. More details can  
166 be found in the supplementary information.

## 167 Results

### 168 Structural characterization and crystallinity of the catalysts

169 With X-ray diffraction (XRD) the crystal structures of the catalysts through the different steps  
170 of the synthesis can be identified. Using the Scherrer equation, the average CuO crystallite  
171 size is estimated from the (111) reflection at  $38.7^\circ$  and reported in Table 2. Porosity analysis of  
172 the catalysts was performed by  $\text{N}_2$ -sorption. The isotherms along with the surface area of the

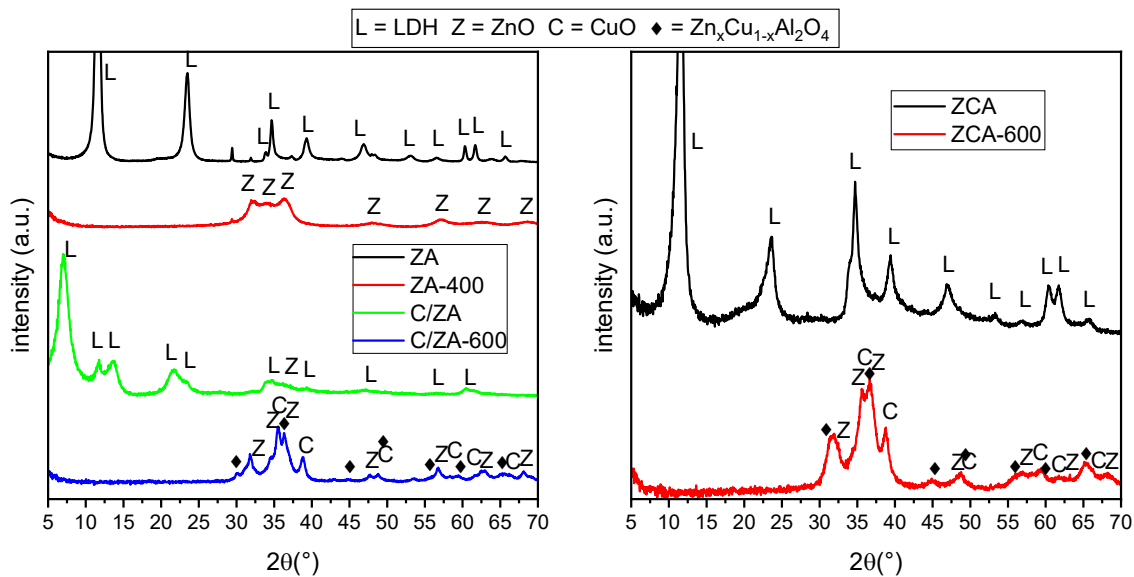
173 materials throughout the different steps of the synthesis are presented in the supplementary  
174 information.

### 175 **Cu loading on ZnAl-LDH**

176 The X-ray diffractograms of C/ZA and ZCA through the different steps of the synthesis are  
177 presented in Figure 1. In both the materials after co-precipitation (ZA and ZCA) the typical  
178 diffraction patterns for a ZnAl carbonate intercalated layered double hydroxide (LDH) (PDF 01-  
179 075-2983 (Gates-Rector and Blanton, 2019)) are present. The typical (003) and (006) basal  
180 reflections centred at 11.7 and 23° are associated with the stacking of the layers (Cavani et al.,  
181 1991).

182 The calcination of ZA at 400°C leads to the collapse of the LDH structure and ZnO is crystallised,  
183 which is observed by its typical pattern (Zincite, PDF 00-036-1451 (Gates-Rector and Blanton,  
184 2019)). After the reconstruction method, the structural reconstruction is confirmed by the  
185 reappearance of the typical LDH reflection planes in the XRD pattern in the forthcoming Cu-  
186 containing material, C/ZA. The reflections are however shifted in comparison to the original  
187 materials, especially the (003) diffraction plane. For the LDH materials, this diffraction plane  
188 is used to calculate the interlayer distance in the structure, providing direct information of the  
189 anions identity and orientation within the interlayer gallery. For example, in the case of ZA,  
190 the interlayer distance is 0.76 nm ( $d_{003}$ ) typical for the carbonate-containing LDH, while for  
191 C/ZA it has increased to 1.25 nm ( $d_{003}$ ) which indicates the intercalation of the acetate anions  
192 during the reconstruction process in the presence of copper acetate aqueous solution (Prevot  
193 et al., 2008). At the same time, for C/ZA, the reflection at 11.7° ( $d_{003}$ ) occurs, which  
194 corresponds to the original carbonate LDH structure.

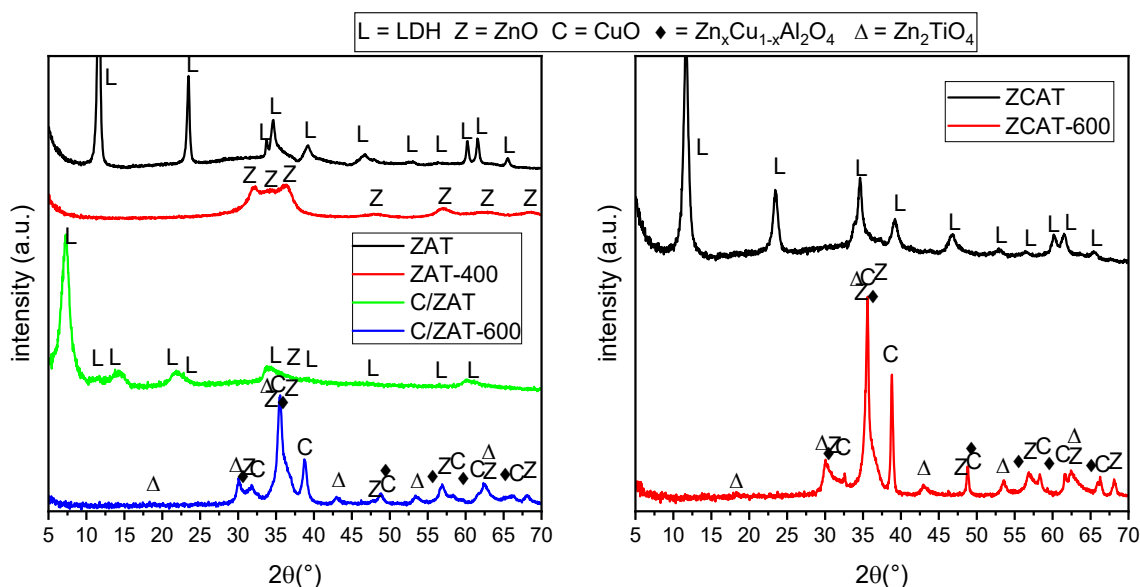
195 The three-way catalysts are obtained after calcination at 600°C. Both cases (C/ZA-600 and  
 196 ZCA-600), feature the XRD patterns of CuO (tenorite, PDF 45-937) as well as of ZnO together  
 197 with a spinel phase. The general formula for a spinel structure is  $A^{2+}B^{3+}_2O_4$  (A – tetrahedral  
 198 site, B – octahedral site). The atomic ratio for spinel generally is  $M^{2+}:M^{3+} = 1:2$  as compared to  
 199  $M^{2+}:M^{3+} = 2:1$  for the LDH (Cavani et al., 1991; Zhao et al., 2017). This means that there will be  
 200 anyway a surplus of  $M^{2+}$  if the LDH is completely transformed into spinel, and this is confirmed  
 201 by the ZnO or CuO formation in the calcined materials. The spinel phase can be associated  
 202 with  $ZnAl_2O_4$  (PDF 00-005-0669). However, after the calcination at 600°C, a limited amount of  
 203 copper can penetrate the spinel structure by substituting one of the elements in the  
 204 octahedral or tetrahedral sites. Due to the fact that Cu and Zn have similar values of ionic  
 205 radius, it is very difficult to make a distinction between the two atoms within the spinel phase  
 206 based on the XRD patterns.



207  
 208 *Figure 1: X-ray diffractograms at different stages of the synthesis of C/ZA (left) and ZCA (right). ZA and ZCA are obtained by*  
 209 *co-precipitation. Cu-containing C/ZA is obtained by the reconstruction method. Note: the intensity of the diffractogram of ZA*  
 210 *has been divided by 2 in order to be displayed together with the others.*

211 **Cu loading on ZnAlTi-LDH**

212 The diffractograms of C/ZAT and ZCAT at the different stages of the synthesis are compared  
 213 in Figure 2. Once more, in both ZAT and ZCAT the typical pattern of LDH after co-precipitation  
 214 is present. Upon calcination at 400°C, ZAT loses the LDH structure with the concomitant  
 215 formation of the ZnO phase. The reconstruction process in the presence of the copper acetate  
 216 solution leads to the intercalation of the acetate anions within the interlayer gallery of the  
 217 obtained LDH (C/ZAT). Further calcination at 600°C of the materials obtained by both co-  
 218 precipitation and reconstruction (C/ZAT-600 and ZCAT-600) are characterized by similar XRD  
 219 patterns with the features of CuO and ZnO combined with a ZnAl<sub>2</sub>O<sub>4</sub> spinel phase. Due to the  
 220 additional presence of the Ti<sup>4+</sup> cations within the LDH phase, the concomitant formation of a  
 221 Zn<sub>2</sub>TiO<sub>4</sub> phase was further identified (PDF 01-082-1438 (Gates-Rector and Blanton, 2019)).



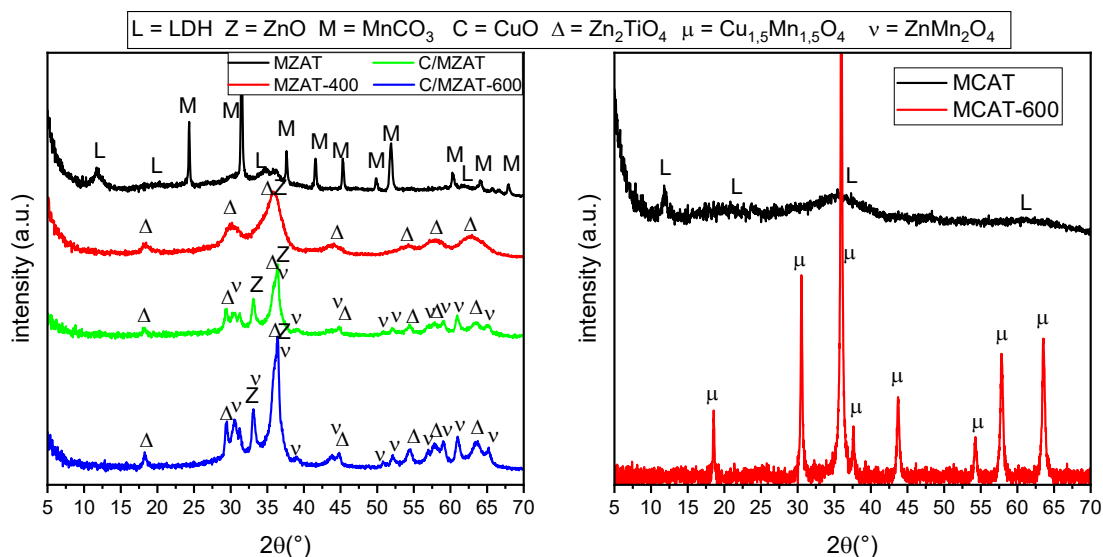
222  
 223 *Figure 2: X-ray diffractograms at different stages of the synthesis of C/ZAT (left) and ZCAT (right). ZAT and ZCAT are*  
 224 *obtained by co-precipitation, while C/ZAT contains Cu deposited by the reconstruction method.*

225 **Cu loading on MnZnAlTi-LDH and MnCuAlTi-LDH**

226 The diffractograms of C/MZAT and MCAT in the different stages of the synthesis are displayed  
 227 in Figure 3. Less intense XRD diffraction lines characteristic to the LDH phase may be identified

228 for the as-prepared MZAT and MCAT material via the co-precipitation method. As such, the  
229 (003) reflection characteristic to the LDH phase may be identified together with the reflections  
230 of a  $\text{MnCO}_3$  (Rhodochrosite, PDF 01-073-4352 (Gates-Rector and Blanton, 2019)) phase. Upon  
231 calcination of the MZAT material at  $400^\circ\text{C}$ , the  $\text{Zn}_2\text{TiO}_4$  phase is formed, which in the case of  
232 the previously discussed ZAT material was not formed at this stage of the synthesis.  
233 Furthermore, when the reconstruction procedure is applied, ZnO particles combined with a  
234  $\text{ZnMn}_2\text{O}_4$  (Hetaerolite, PDF 00-024-1133) phase are formed (C/MZAT). Therefore, for this  
235 initial metal composition in the LDH structure, the reconstruction process via the memory  
236 effect property was not achieved, probably due to the low crystallinity of the initially obtained  
237 layered structure and as well as the presence of the additional  $\text{MnCO}_3$  phase. Further  
238 calcination at  $600^\circ\text{C}$  leads to very significant differences between the obtained three-way  
239 catalytic materials, e. g. the C/MZAT-600 and the MCAT-600, respectively. This can be  
240 attributed to the presence of Zn. In the XRD pattern of C/MZAT-600 all the peaks may be  
241 associated to structures that contain Zn, namely  $\text{Zn}_2\text{TiO}_4$ ,  $\text{ZnMn}_2\text{O}_4$  and ZnO, respectively,  
242 while in the case of MCAT-600 these structures are not formed due to the absence of Zn ions  
243 in the initial synthesis mixtures. The presence of Cu containing phase and its content is proven  
244 by measuring SEM-EDX as it will be discussed in a later section. In the latter case, a spinel-like  
245 phase, identified as  $\text{Cu}_{1.5}\text{Mn}_{1.5}\text{O}_4$  (PDF 01-070-0262), is formed from the reaction of the high  
246 amounts of  $\text{Mn}_2\text{O}_3$  and CuO, which is in agreement with previously reported studies in  
247 literature (Blasse, 1966; Vandenberghe et al., 1973, 1976; Behar et al., 2012; Biemelt et al.,  
248 2016; Van Everbroeck et al., 2020).

249



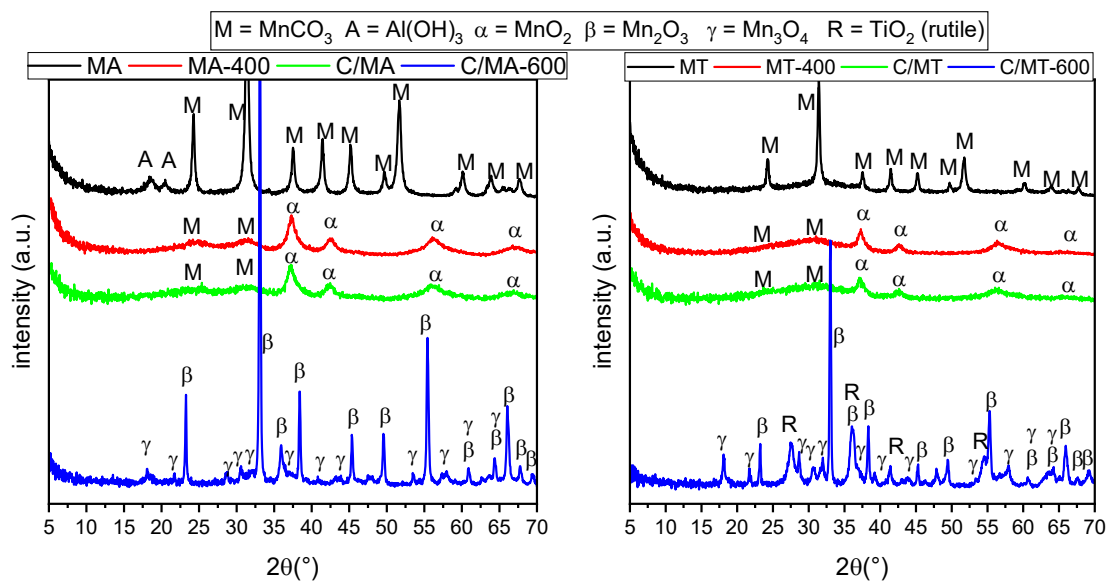
250

251 *Figure 3: X-ray diffractograms at different stages of the synthesis of C/MZAT (left) and MCAT (right). MZAT and MCAT are*

252 *obtained by co-precipitation. Cu-containing C/MZAT is obtained by the reconstruction method.*

### 253 **Cu loading on MnAl and MnTi-LDH**

254 According to the diffractograms of C/MA and C/MT at the different stages of the synthesis, in  
 255 neither of the materials the LDH structure was formed after co-precipitation (Figure 4). As a  
 256 consequence, no LDH structures were formed after the reconstruction step. The only  
 257 crystalline structure in the co-precipitated materials was  $\text{MnCO}_3$ , together with some  $\text{Al}(\text{OH})_3$   
 258 (Bayerite, PDF 01-077-0250 (Gates-Rector and Blanton, 2019)) in the case of the as-  
 259 synthesized MA material. Calcining at  $400^\circ\text{C}$  and applying the reconstruction procedure leads  
 260 to very similar materials. There are still some remnants of the  $\text{MnCO}_3$  phase together with  
 261  $\text{MnO}_2$  (Akhtenskite, PDF 01-089-5171 (Gates-Rector and Blanton, 2019)). Finally, the heat  
 262 treatment of the catalysts at  $600^\circ\text{C}$  leads to the transformation of the  $\text{MnO}_2$  phase into a  
 263 mixture of  $\text{Mn}_2\text{O}_3$  (Bixbyite PDF 31-825) and  $\text{Mn}_3\text{O}_4$  (PDF 86-2337), while there is no indication  
 264 of a spinel phase formation, such as  $\text{Cu}_{1.5}\text{Mn}_{1.5}\text{O}_4$  in the XRD patterns. A  $\text{TiO}_2$  rutile phase (PDF  
 265 00-021-1276) can additionally be found in MT-600.



266

267 *Figure 4: X-ray diffractograms at different stages of the synthesis of C/MA (left) and C/MT (right). MA and MT are obtained*

268 *by co-precipitation and C/MA contains Cu deposited by the reconstruction method.*

## 269 Quantification of Cu loading in the three-way catalysts

270 The Cu content of the final materials as determined by SEM-EDX and the average CuO

271 crystallite size before and after the catalytic test is summarized in Table 2. The Cu content is

272 around 35-40% for the materials that had Cu included in the first co-precipitation synthesis

273 step (ZCA-600, ZCAT-600 and MCAT-600), but it is further increased by applying the

274 reconstruction synthesis route on the calcined LDH materials C/ZA-600 and C/ZAT-600. Lower

275 copper loadings are achieved in the case of the parent materials that did not display the

276 characteristics of a well-layered structure after the co-precipitation process, such as C/MZAT,

277 C/MA and C/MT. The precursors used in their preparation did not present only an LDH-type

278 structure but a combination with other crystalline phases or sometimes a more amorphous

279 character. It should be noted that, for all the prepared catalysts, the exact same concentration

280 of copper acetate solution is used. Large variations in the crystallites sizes of the CuO

281 nanoparticles, e. g. 7 nm and 26 nm, respectively occur in ZCA-600 and ZCAT-600 prepared via

282 the co-precipitation method. On the other hand, by following the reconstruction method,

283 similar sizes of the CuO crystallites of  $\pm 10$  nm, are achieved, namely for the C/ZA-600 and  
 284 C/ZAT-600 three-way catalysts. After the catalytic test, the CuO crystallite sizes increase for  
 285 most of the materials, except for ZCAT-600. While 18 nm CuO crystals are formed for MCAT-  
 286 600 after the test, no CuO crystals are XRD discernible in any of the other Mn-containing  
 287 materials, indicating that these CuO crystallites have sizes that are probably below 3 nm.

288

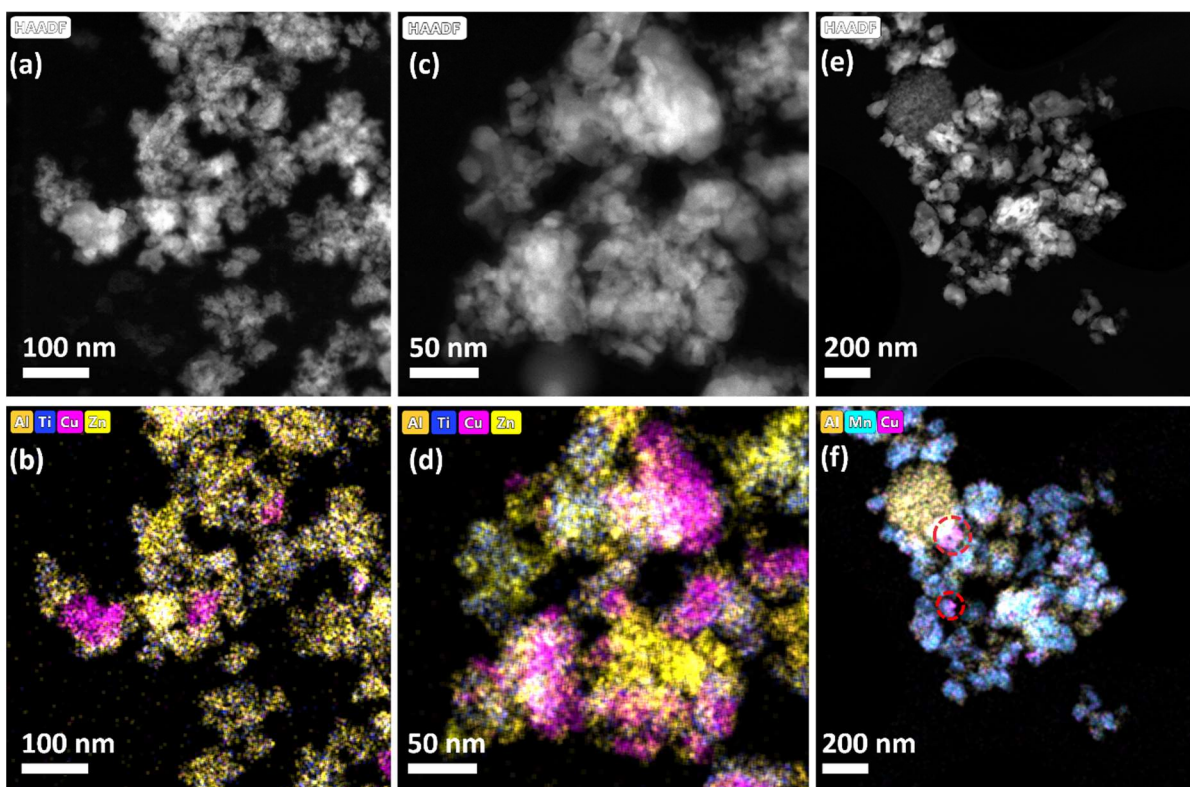
Sample	Cu content (wt%)	CuO crystallite size	
		(before cat. test) (nm)	CuO crystallite size (after cat. test) (nm)
ZCA-600	37	7	14
C/ZA-600	62	11	15
ZCAT-600	35	26	21
C/ZAT-600	56	10	15
MCAT-600	41	-	18
C/MZAT-600	10	-	-
C/MA-600	11	-	-
C/MT-600	11	-	-

289 *Table 2: Cu content and estimated CuO crystallite size of the final materials (before and after the catalytic test). Copper*  
 290 *content determined by SEM-EDX. CuO size calculated with the Scherrer equation.*

291 Energy-dispersive X-ray spectroscopy (STEM-EDS) maps were acquired from different areas  
 292 on samples ZCAT-600, C/ZAT-600 and C/MA-600 to further investigate the size of the CuO  
 293 nanoparticles and their distribution. Figure 5 shows HAADF-STEM images and their respective  
 294 elements composition, obtained on representative areas for each sample. Whereas for ZCAT-  
 295 600 and C/ZAT-600 samples it is easy to find isolated zones that clearly indicate the presence



296 of Cu (Figure 5b, d), such areas are absent in the C/MA-600 sample due to a high degree of  
297 agglomeration between particles. Even so, some isolated CuO nanoparticles are marked in  
298 Figure 5f with red circles. CuO nanoparticles adopt a rectangular shape in all three cases, with  
299 an average nanoparticle size, determined by STEM-EDS, of  $35 \pm 17$  nm,  $15 \pm 11$  nm and  $20 \pm 8$   
300 nm for the ZCAT-600, C/ZAT-600 and C/MA-600 composites respectively (seeSupplementary  
301 figure 8). These values do not exactly match the sample size obtained in bulk, which can be  
302 understood since TEM is a very local technique. Nevertheless, the particle size trend is  
303 confirmed, with bigger particles for the ZCAT-600 sample in comparison to the C/ZAT-600. On  
304 the other hand, for the C/MA-600 sample, CuO nanoparticles are observed in small quantities,  
305 which explains why they are not detected by XRD.



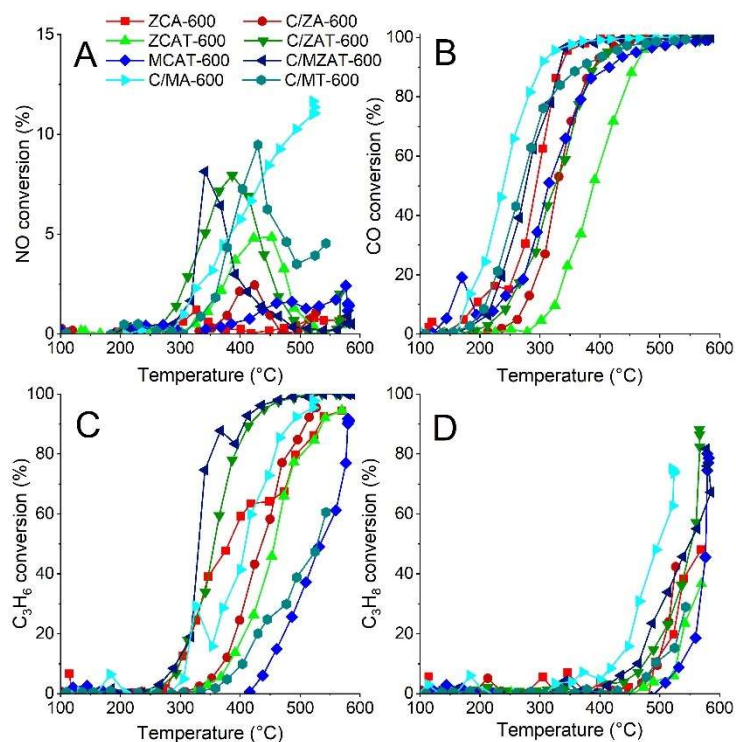
307 *Figure 5: HAADF-STEM images and their respective EDS elemental distribution for (a, b) ZCAT-600, (c, d) C/ZAT-600 and (e, f)*  
308 *C/MA-600 composites. For the EDS maps, Al is marked in brown, Ti in dark blue, Cu in pink, Zn in yellow and Mn in light blue.*

309

## 310 Catalytic activity testing

311 As described in the experimental section, the catalytic test consists of four heating and cooling  
312 cycles in three different gas compositions, after an initial pre-treatment cycle. The conversions  
313 of NO, CO, propene (C<sub>3</sub>H<sub>6</sub>) and propane (C<sub>3</sub>H<sub>8</sub>) as a function of temperature during heating in  
314 the different cycles are displayed in Figure 6 and Supplementary Figure 9 to 11. The  
315 temperatures at which 50% conversion (Light-off temperature, T<sub>50</sub>) is reached and, in case of  
316 the NO conversion, the maximal conversion and temperature at which this conversion is  
317 reached can be found in tables in the supplementary info. Light-off temperatures at 50%  
318 conversion are typically used in temperature-programmed experiments to indicate the onset  
319 temperature of significant catalytic activity (Biasin et al., 2019).

320 In the first stoichiometric cycle (Figure 6Figure 6), NO conversion generally does not reach over  
321 10%. All the NO conversions display local maxima, only for C/MA-600 the NO conversion keeps  
322 rising until the end of the measurements. The maximum NO conversion is notably higher for  
323 C/ZAT-600, C/MZAT-600, C/MA-600 and C/MT-600. The temperature of 50% CO conversion is  
324 the lowest for C/MA-600, C/MT-600 and C/MZAT-600, while ZCAT-600 differentiates the most  
325 from the others for a much higher T<sub>50</sub>. On the other hand, the propene conversion curves are  
326 much more diverse. While C/MZAT-600 and C/ZAT-600 have a T<sub>50</sub> around 330 and 350°C,  
327 C/MT-600 and MCAT-600 only reach this above 520°C. Propane conversion generally only  
328 starts above 500°C. The only exception is in the case of C/MA-600, which reaches even 50%  
329 under 500°C.



330

331 *Figure 6: Temperature programmed conversion curves under stoichiometric conditions, first cycle, A: NO conversion, B: CO*

332 *conversion, C: propene conversion, D: propane conversion.*

333

## 334 Discussion

335 Layered double hydroxides have been successfully formed even if the Zn<sup>II</sup> atoms were partially  
 336 substituted by Cu<sup>II</sup> or Al<sup>III</sup> atoms are partially substituted by Ti<sup>IV</sup>, in both cases for a substitution  
 337 degree of 50 mol%. The incorporation of large quantities of Mn atoms in the LDH structure is  
 338 however not as straightforward, due of the fact that Mn atoms lead to octahedral distortions  
 339 (similarly to Cu atoms) as a result of the Jahn-Teller effect (Cavani et al., 1991; Fridrichová et  
 340 al., 2017). In the cases of MCAT and MZAT, there is still evidence of the LDH structure, but the  
 341 experiments prove that Cu or Zn are essential to form the LDH structure with Al (and Ti).  
 342 Aisawa *et al.* reported that when CO<sub>3</sub><sup>2-</sup> is used as the anion source for the synthesis of Mn-Al  
 343 LDH, the LDH could still be formed but MnCO<sub>3</sub> could also be formed, which is indeed the case  
 344 here (Aisawa et al., 2002).

345 While no indications of the LDH-like structure after calcination at 400°C in the XRD results, the  
346 LDH structure is reformed after suspension in an aqueous copper acetate solution, even after  
347 a long time at ambient conditions. The experiments prove that it is essential for the LDH  
348 structure to have a high crystallinity for the starting material in order to exhibit these  
349 reconstructive properties. In the case of MZAT, the low intensity of the LDH characteristic XRD  
350 reflections as well as the presence of the MnCO<sub>3</sub> segregated phase indicate that the material  
351 is very unlikely to reconstruct. Accordingly, the materials that do not have the LDH structure  
352 before calcination will also not form LDH-like structures when mixed with the copper acetate  
353 solution. Besides the characteristic (003) plane of acetate intercalated LDH, the XRD patterns  
354 of C/ZA show (Figure 1) another reflection maximum at 11.7°, characteristic to the original  
355 carbonate LDH. This can be related to the presence of CO<sub>2</sub> in the air or CO<sub>2</sub> in the aqueous  
356 acetate solution (Stanimirova and Balek, 2008).

357 The current experiments suggest that the reconstructed LDH materials are able to take-up  
358 large amounts of copper between their layers. After all the synthesis steps, the materials with  
359 LDH structure retain up to 6 times more copper (around 60 wt%) than those without LDH  
360 structure (around 10 wt%) (Table 2). The thermal decomposition of LDH generally consists out  
361 of two phases. At around 200°C the interstitial water is lost, while between 300 and 450°C  
362 generally the hydroxyl groups and carbonate are removed (Reichle et al., 1986). From XRD, it  
363 is observed that at 400°C some oxides such as ZnO are already crystallised, but the mixed  
364 oxides, such as spinels are not yet detectible in the XRD patterns. This means that the mixed  
365 oxides are still amorphous, so the lamellar microstructure of LDH is still there, albeit not as  
366 crystalline to be discernible with XRD. These LDOs (layered double oxides) obtained by  
367 calcination at mild temperatures are known to be very effective to intercalate anions via the  
368 reconstruction based on the memory effect mechanism (Seftel et al., 2018). Although it seems

369 counterintuitive, LDH are also able to adsorb metal cations from a solution. This can occur  
370 through multiple mechanisms such as precipitation, surface complexation, isomorphic  
371 substitution or chelation (Liang et al., 2013). It can be stated that the intercalation of the  
372 anionic species occurs concomitantly with the rehydroxylation of the brucite-like layers. As  
373 such, the presence of the hydroxyl groups on the surface of the LDH enables all these  
374 mechanisms, explaining why the materials that did not exhibit LDH characteristics in this study  
375 did not take up high amounts of Cu<sup>II</sup>. To determine the exact mechanism that takes place in  
376 the experiments more research is needed, which is beyond the scope of the current study.

377 The estimated CuO crystallite size is much larger for ZCAT-600 in comparison to the other  
378 catalysts, while the Cu content is much lower in ZCAT-600 than in C/ZAT-600 and C/ZA-600  
379 (Table 2). This seems to suggest that the reconstruction method leads to much better  
380 dispersed CuO in the final material, than by adding copper during the LDH synthesis via the  
381 direct co-precipitation route. On the other hand, ZCA-600 contains about the same amount of  
382 copper as ZCAT-600, but the CuO size is considerably smaller. For the calcined catalysts, Cu is  
383 also likely to be found in the mixed oxide phases (such as ZnAl<sub>2</sub>O<sub>4</sub> and Zn<sub>2</sub>TiO<sub>4</sub>) explaining the  
384 difference between ZCAT-600 and ZCA-600; Cu is likely to form CuAl<sub>2</sub>O<sub>4</sub> or together with zinc  
385 in a Zn<sub>x</sub>Cu<sub>1-x</sub>Al<sub>2</sub>O<sub>4</sub> spinel after calcination at high temperature (Friedman et al., 1978; Mista et  
386 al., 2003; Patel et al., 2015, 2017). This is certainly the reason why no CuO phase could be  
387 detected in MCAT-600 by XRD analysis. When mixed together, the solid-state reaction  
388 between the CuO and Mn<sub>2</sub>O<sub>3</sub> is very likely to form a spinel phase even at temperatures lower  
389 than 600°C (Vandenberghe et al., 1973; Van Everbroeck et al., 2020). For C/MA-600 and C/MT-  
390 600, the measured copper content is low, resulting in CuO nanoparticles not being detectable  
391 by XRD analysis (Figure 4). However, after the catalytic tests, the crystallite size of CuO has  
392 increased in some of the catalysts due to the sintering process. Interestingly, in the case of

393 ZCAT-600, there is even a slight decrease of CuO crystallite size; this could be possibly  
394 correlated to the diffusion of copper ions into the mixed oxides. The most notable change of  
395 the CuO crystallites size is occurring within the MCAT-600 catalyst. In this case, before the  
396 catalytic test, no CuO could be detected in XRD, while after the catalytic test, which can be  
397 found in the supplementary information, the XRD analysis indicates the presence of CuO  
398 crystals with sizes of around 18 nm. This may be due to the transformation of  $\text{Cu}_{1.5}\text{Mn}_{1.5}\text{O}_4$   
399 phase into a  $\text{CuMn}_2\text{O}_4$  type phase as it was indicated evidenced by XRD analysis (Supplementary  
400 figure 26). Part of the  $\text{Cu}^{2+}$  segregates from the structure and agglomerates as CuO.

401 The profiles of the CO conversion as a function of temperature under all the different  
402 conditions indicate that the ZCAT-600, which has the largest CuO crystallite size, always  
403 achieves the highest  $T_{50}$ . C/ZA-600 and C/ZAT-600 perform very similar which can be assigned  
404 to the fact that these contain similar loadings of CuO with sizes in the same range. Only in the  
405 lean and second stoichiometric cycle, C/ZA-600 shows a relatively higher  $T_{50}$  than C/ZAT-600.  
406 At the same time, ZCA-600 reaches a lower  $T_{50}$  in the first stoichiometric cycle than C/ZAT-  
407 600, again related to the smaller CuO size. However, in the second stoichiometric cycle, it  
408 performs very comparable to C/ZAT-600, likely due to the growth of CuO crystallites. CO  
409 oxidation at lower temperatures due to smaller or more disperse CuO particles has been  
410 reported before. Higher CuO loading is also advantageous for CO conversion but only if good  
411 dispersion can be assured (Luo et al., 2005; Xin et al., 2016). By using the LDH memory effect  
412 both a high loading and good dispersion was ensured in C/ZAT-600 compared to ZCAT-600.

413 The Mn containing materials are generally superior for the CO conversion than those without.  
414 C/MA-600 has in every case the lowest  $T_{50}$ , while MCAT-600 has most of the times the highest  
415  $T_{50}$ . Again, MCAT-600 has a much higher copper content and after the test it aggregates into

416 large CuO crystallites, explaining the higher  $T_{50}$ . The remarkable performance of the Cu-Mn  
417 compounds for CO oxidation is known since the first World War (1920), when mixtures of CuO  
418 and  $MnO_2$ , known as hopcalite, were studied to be used in gas masks to prevent CO poisoning  
419 (Lamb et al., 1920; Almquist and Bray, 1923; Bliss, 1955). The difference between the materials  
420 used for gas masks and the materials in this application (three-way catalysis) is that the gas  
421 masks need to operate around room temperature, while for three-way catalysis it operates  
422 for long periods at high temperatures. In the case of three-way catalysis, the materials are  
423 already calcined at  $600^\circ\text{C}$ , so the crystallisation of the compounds explains why they are  
424 unable to convert CO at room temperature. For MCAT-600, the copper manganite structure  
425 is more crystalline than the other compounds, which can also explain the higher  $T_{50}$  for CO. A  
426 possible reason for the best performance of C/MA-600 for CO conversion is related to the  
427 amorphous alumina present, which helps disperse the Cu and Mn mixtures over the entire  
428 material, inhibiting their agglomeration and further crystallization. Furthermore, the higher  
429 surface area of C/MA-600 as compared to C/MT-600 and C/MZAT-600 also contributes to  
430 better dispersion. Moreover, the propene/propane conversion profiles and the benchmarking  
431 of the catalyst are discussed in detail in the Supplementary Information. A further discussion  
432 on the mechanisms that take place on CuO and copper manganite can be found in the  
433 literature (Cocke and Vepřek, 1986; Vepřek et al., 1986; Buciuman et al., 1999; Van Everbroeck  
434 et al., 2020).

435 The formation of  $N_2O$  and  $NH_3$  can occur from NO reduction, as reaction intermediates (Jobson  
436 et al., 1994; Okamoto and Gotoh, 1997; Okamoto et al., 1997; Granger et al., 2004; Nevalainen  
437 et al., 2018). In the current experiments, no  $N_2O$  is detected at low temperatures in  
438 stoichiometric and lean conditions, while in rich conditions, small quantities ( $< 20$  ppm) of  $N_2O$   
439 are sometimes detected above  $400^\circ\text{C}$ . Overall, the selectivity towards  $N_2O$  is very low. A

440 further discussion is on the NO conversion and selectivity is presented in the Supplementary  
441 Information. Furthermore, corresponding stoichiometric, lean and rich conditions results as  
442 well as reaction rates taking into account (i) specific surface area (Supplementary Figure 16,  
443 17 and 18) and the (ii) amount of reactive sites of the catalyst (Supplementary Figure 19, 20  
444 and 21) are detailed in the Supplementary Information.

445 To demonstrate the stability of the materials a structural investigation of the spent catalysts  
446 is also presented in the Supplementary Information.

## 447 Conclusions

448 Layered double hydroxides with various compositions were prepared to obtain copper oxide  
449 containing three-way catalysts. The Cu source was added following two different methods, (a)  
450 during the LDH synthesis via co-precipitation and (b) by reconstruction of the calcined LDOs  
451 material based on the unique memory effect property. Larger amounts of CuO nanoparticles  
452 can be loaded on the three-way catalysts by using the reconstruction method, but a well-  
453 crystallized starting LDH structure is necessary for an effective reconstruction process.

454 Catalytic testing of the obtained three-way catalysts showed that the CO and the  
455 hydrocarbons, e. g. propene and propane, were generally converted at lower temperatures  
456 when CuO crystallites were smaller and well-dispersed. A synergistic effect was observed  
457 when Mn atoms were present in the catalysts structure. Further, it was demonstrated that a  
458 better dispersion of CuO was more important than achieving high loadings of CuO  
459 nanoparticles on the three-way catalysts. The presence of Al in the LDH layers creates a more  
460 appropriate support material, compared to the presence of Ti in the layers, due to the fact  
461 that it enabled a higher Cu dispersion and provided a higher surface area. The main drawback  
462 observed in the case of Ti-containing catalysts is directly related to its crystallisation to rutile,



463 after long exposure to high temperatures. However, the addition of Zn prevented the  
464 crystallisation of rutile by the formation of  $Zn_2TiO_4$ , which was advantageous for the propene  
465 conversion process. On the other hand, the presence of Ti proved to be beneficial for the  
466 conversion and the selectivity towards  $N_2$  in the NO conversion. Highly dispersed CuO on the  
467 supporting material was again one of the most important factors to reach a high catalyst  
468 activity. These conclusions offer new insights in the design of LDH-based MMOs for the  
469 reduction of CRMs in three-way catalytic convertors.

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## 725 Supplementary information

### 726 Reagents

727 All chemicals were purchased from Sigma-Aldrich/Merck and used without any further purification.

728 The used chemicals are:  $\text{Cu}(\text{NO}_3)_2 \cdot 3\text{H}_2\text{O}$  (99.0 wt%),  $\text{Zn}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$  (99 wt%),  $\text{Al}(\text{NO}_3)_3 \cdot 9\text{H}_2\text{O}$  (99 wt%),

729  $\text{TiCl}_4$  (99.9 wt%),  $\text{MnCl}_2$  (97.0 wt%), NaOH (98.5 wt%),  $\text{Na}_2\text{CO}_3$  (99.95-100.05 wt%), Copper acetate (98

730 wt%).

### 731 Scherrer equation:

732 The average crystallite size was estimated using the Scherrer equation:

$$733 \quad d = \frac{K\lambda}{\beta \cos(\theta)}$$

734 Where,  $d$  = crystallite size,  $\lambda$  = the wavelength of the X-ray source (0.15406 nm),  $\beta$  = the full

735 width at half maximum of the chosen diffraction peak ( $2\theta$ ),  $\vartheta$  = Bragg angle and  $K$  = the shape

736 constant chosen as 0.9. Using this equation, the morphological effects and disorder to

737 reflection widths were not taken into account, so the obtained values were merely used for

738 relative comparison of the materials (Hargreaves, 2016).

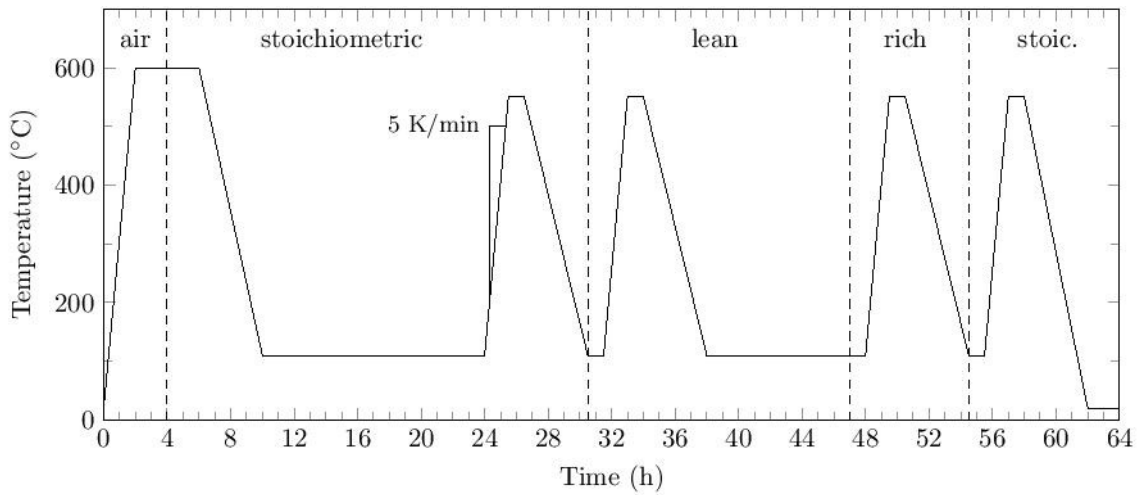
739

### 740 Catalytic activity measurements

741 The experiment consisted of four consecutive heating (at 5°C/min) and cooling cycles between 110

742 and 550°C, in different conditions (stoichiometric, lean, rich), preceded by a pre-treatment at 600°C

743 in air and under stoichiometric condition.



744

745

*Supplementary Figure 1: Procedure for catalytic testing.*

746

The gas composition for each of the conditions is described in Supplementary **¡Error! No se encuentra**

747

**el origen de la referencia..** The flow rate was adjusted to correspond to a Gas Hourly Space Velocity of

748

60 000 mL h<sup>-1</sup> g<sup>-1</sup>. The composition of the gas at the outlet was monitored by a Thermo Scientific 42i-

749

HL chemiluminescence analyzer for NO<sub>x</sub> detection and an Agilent Technologies 490 micro GC for the

750

quantification of N<sub>2</sub>, N<sub>2</sub>O, H<sub>2</sub>, CO, propane, propene and CO<sub>2</sub>. The displayed conversions were

751

calculated from the heating cycles, using the following formula:

752

$$X_i(\%) = \left(1 - \frac{[i]}{[i]_0}\right) \cdot 100$$

753

Where  $X_i$  = the conversion of compound  $i$ ,  $[i]$  is the concentration of  $i$  at the outlet, and  $[i]_0$  is the inlet

754

concentration of  $i$ .

755

The concentration of NH<sub>3</sub> was calculated using the following formula assuming that all NO not

756

converted to N<sub>2</sub> or N<sub>2</sub>O leads to NH<sub>3</sub>:

757

$$[NH_3] = [NO]_0 - ([NO] + 2 [N_2] + 2 [N_2O])$$

758

The selectivity of NO towards N<sub>2</sub> ( $S_{N_2}$ ), N<sub>2</sub>O ( $S_{N_2O}$ ) and NH<sub>3</sub> ( $S_{NH_3}$ ) were calculated as follows:

759

$$S_{N_2} = \frac{2 [N_2]}{[NO]_0 - [NO]}$$

760 
$$S_{N_2O} = \frac{2 [N_2O]}{[NO]_0 - [NO]}$$

761 
$$S_{NH_3} = \frac{[NH_3]}{[NO]_0 - [NO]}$$

762 The yield ( $Y_i$ ) was determined as:

763 
$$Y_i = X_i \cdot S_i$$

764

Condition/Gas(vol%)	NO	CO	C <sub>3</sub> H <sub>6</sub>	C <sub>3</sub> H <sub>8</sub>	H <sub>2</sub>	O <sub>2</sub>	CO <sub>2</sub>	H <sub>2</sub> O	He
<i>Stoichiometric</i>	0.1	0.7	0.045	0.022	0.23	0.73	15	10	Balance
<i>Lean</i>	0.1	0.5	0.03	0.015	0.17	0.90	15	10	Balance
<i>Rich</i>	0.1	0.9	0.06	0.030	0.30	0.55	15	10	Balance

765 *Supplementary Table 3: Gas composition of the different conditions during the catalytic experiment.*

766

767 The specific reaction rate (Spec. RR) is calculated by dividing the reaction rate (RR) by the BET specific  
768 surface area (SSA):

769 
$$Spec. RR = \frac{RR}{SSA}$$

770 The reaction rate (RR) is calculated as follows (example for CO):

771 
$$RR_{CO} = \frac{[CO]_0 \cdot X_{CO} \cdot F \cdot p}{T \cdot R \cdot g_{cat}}$$

772  $F$  is the total flow rate,  $p$  the pressure (1.013 bar),  $T$  the temperature,  $R$  the gas constant.  $g_{cat}$  the  
773 weight of the catalyst.

774 The turnover frequency is calculated by dividing the reaction rate (RR) by the number of moles Cu in  
775 the catalyst, based on the SEM-EDX results:

776

$$TOF = \frac{RR}{\# \text{ mol CuO} / g_{cat}}$$

777

778 [Textural characteristics of the catalysts](#)

779 The values for the BET specific surface area (SSA) of the different synthesized materials are  
 780 presented in Supplementary table 2. The evolution of the surface area through the different  
 781 synthesis steps can be deduced from left to right. Isotherms and additional textural properties  
 782 are presented in Supplementary figure 2 to 7. For the initial materials, the copper-containing  
 783 LDHs have each time a higher surface area than their non-copper containing LDHs (e. g. ZCA  
 784 vs. ZA or ZCAT vs. ZAT). At the same time, titanium and manganese presence in the materials  
 785 generally tends to increase the surface area (e. g. the case of ZAT, ZCAT, MZAT, MCAT or MT).  
 786 Comparing the three-way catalysts obtained by the thermal treatment at 600°C to the initial  
 787 ones, it can be concluded that the results are quite divergent; ZA, ZCA and MA increase their  
 788 surface area along with the synthesis steps, while the others decrease their surface area along  
 789 the process.

790

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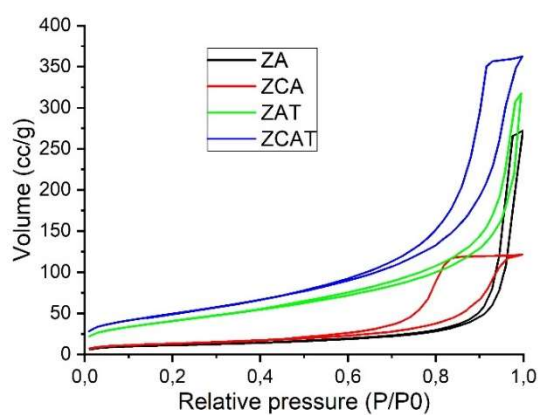
SSA (m <sup>2</sup> /g)	Three-way catalysts			
	Non-calcined	Calcined 400°C	Reconstructed	(calcined 600°C)
ZA	38	78	14	66
ZCA	45	-	-	63
ZAT	148	105	58	80
ZCAT	179	-	-	54

MZAT	195	130	113	46
MCAT	259	-	-	117
MA	66	188	98	73
MT	159	157	60	28

793 *Supplementary table 2: BET specific surface area (SSA) of the synthesized materials in the different stages (non-calcined*  
794 *material, calcined at 400°C, after reconstruction in Cu-containing aqueous solutions and the obtained three-way catalysts*  
795 *produced by the calcination at 600°C).*

796

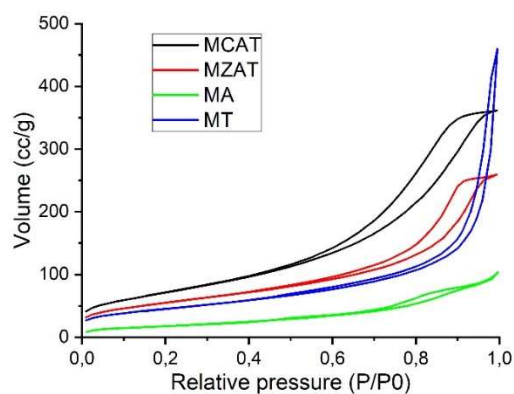
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798

799 *Supplementary figure 2: N<sub>2</sub>-sorption isotherm of the materials prepared by co-precipitation (ZA, ZCA, ZAT, ZCAT).*

800



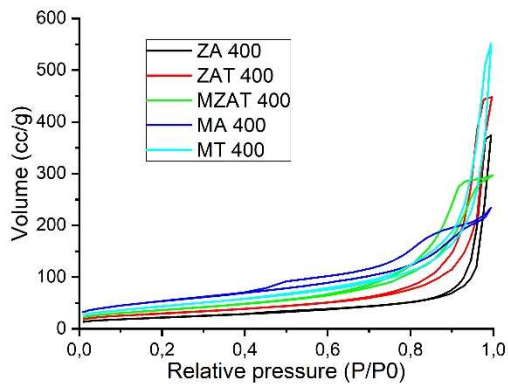
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802 *Supplementary figure 3: N<sub>2</sub>-sorption isotherm of the materials prepared by co-precipitation (MCAT, MZAT, MA, MT).*

803

804



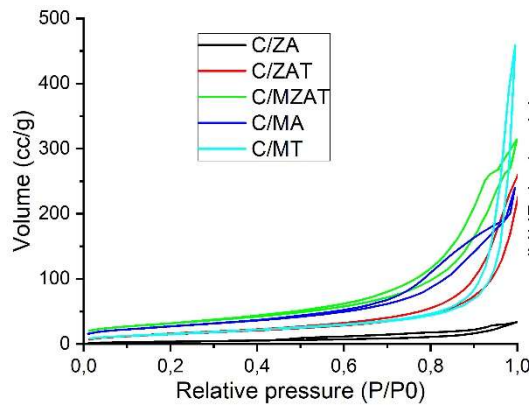


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806

Supplementary figure 4:  $N_2$ -sorption isotherm of the materials calcined at 400 °C.

807

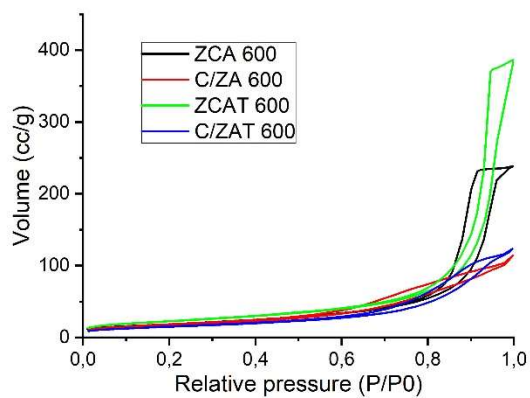


808

809

Supplementary figure 5:  $N_2$ -sorption isotherm of the materials after impregnation with Cu by the reconstruction method.

810



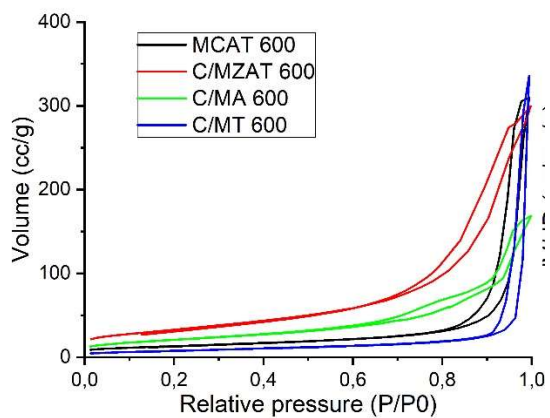
811

812

Supplementary figure 6:  $N_2$ -sorption isotherm of the final materials calcined at 600°C (ZCA 600, C/ZA 600, ZCAT 600, C/ZAT 600).

813

814



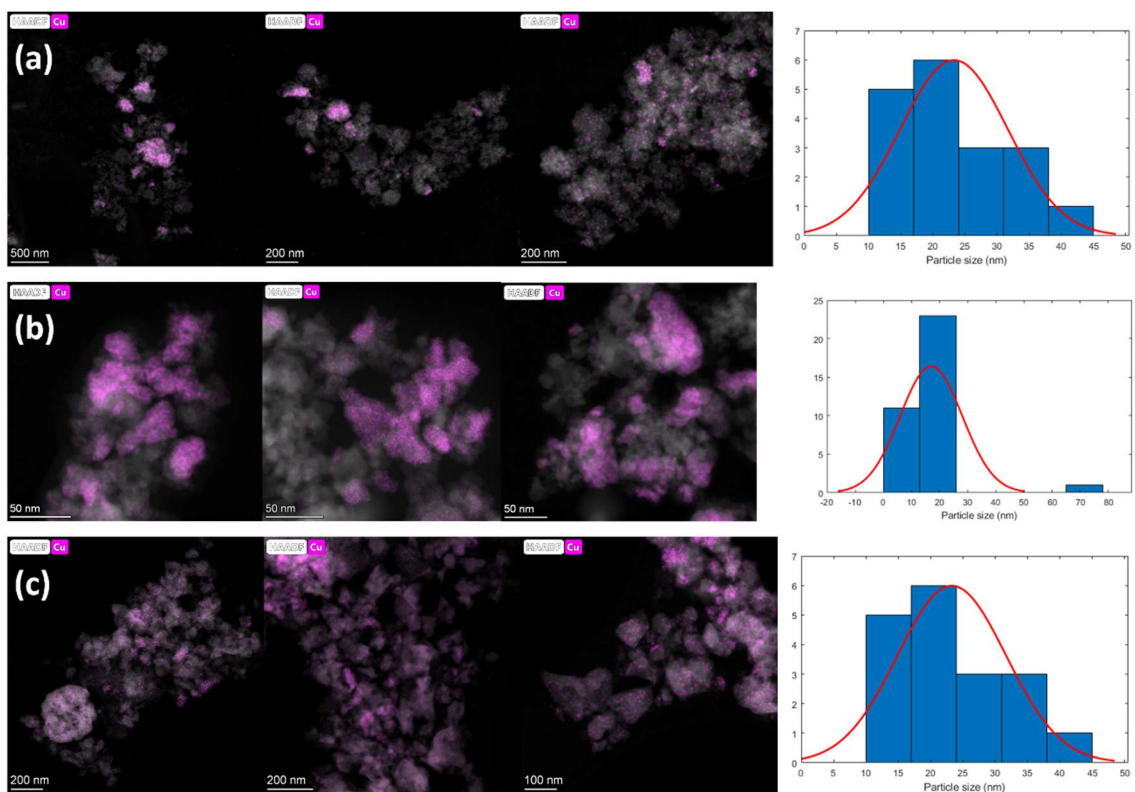
815

816 *Supplementary figure 7: N<sub>2</sub>-sorption isotherm of the final materials calcined at 600°C (MCAT 600, C/MZAT 600, C/MA 600,*  
 817 *C/MT 600).*

818

819

820 EDS mapping



821

822 *Supplementary figure 8: HAADF-STEM image with the EDS Cu mapping of different areas and CuO nanoparticles size*  
 823 *distribution of (a) ZCAT-600, (b) C/ZAT-600 and (c) C/MA-600.*

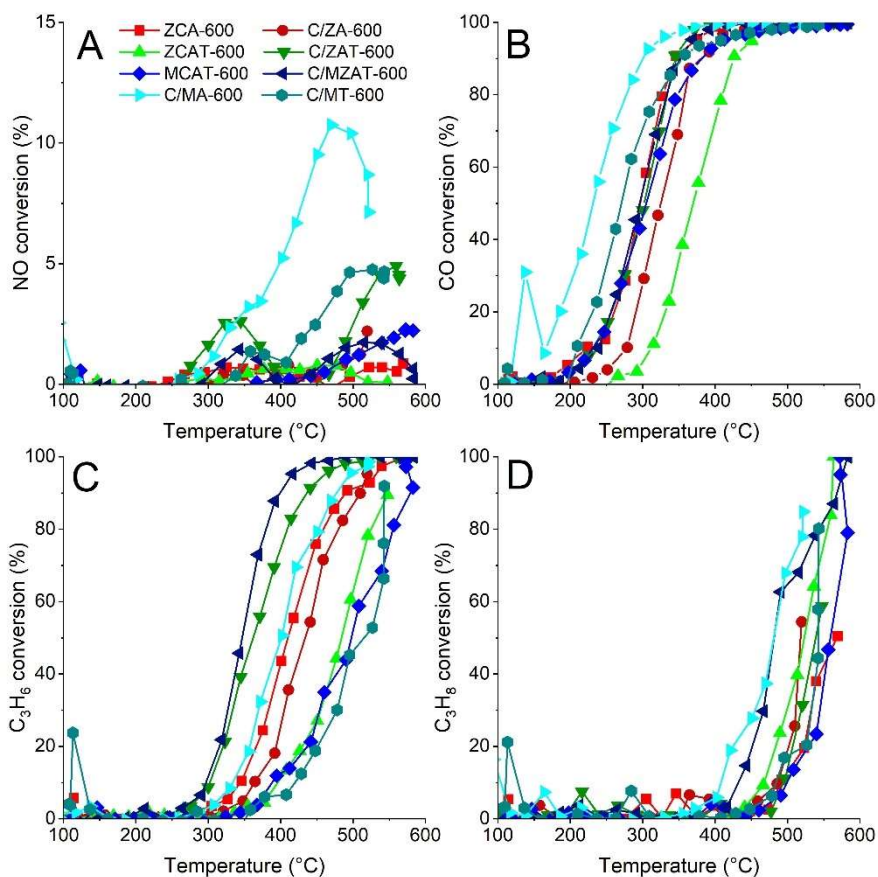
824

825

Stoic 1	Temperature 50 %			temp.	
	conversion			max. NO	for max.
	CO	C <sub>3</sub> H <sub>6</sub>	C <sub>3</sub> H <sub>8</sub>	conv.	NO conv.
ZCA-600	293	381	n.r.	1.2	327
C/ZA-600	329	436	n.r.	2.4	424
ZCAT-600	389	457	n.r.	4.9	453
C/ZAT-600	325	356	552	8	387
MCAT-600	314	534	577	2.4	576
C/MZAT-600	276	331	549	8.1	341
C/MA-600	239	409	497	11.7	522
C/MT-600	269	528	n.r.	9.5	429
Comm. TWC	217	262	454	47.1	285
0.5% Pd/CeZrO <sub>2</sub>	136	177	532	27	232

828 *Supplementary table 3: temperatures at which 50 % conversion is reached and maximal NO conversion in the first*  
829 *stoichiometric cycle. N.r. = not reached.*

830 Under lean conditions (Supplementary Figure 9), one would expect lower NO conversions and  
831 higher conversions of CO and the organic compounds, due to the relatively higher amount of  
832 O<sub>2</sub> present. The maximal NO conversion is indeed lower (< 5%), except for C/MA-600, which  
833 still manages to convert up to 10.7%. CO and propene conversions are very comparable to the  
834 first stoichiometric cycle, while propane conversion is slightly improved.



835

836

Supplementary Figure 9: Temperature programmed conversion curves under lean conditions, A: NO conversion, B: CO

837

conversion, C: propene conversion, D: propane conversion.

838

Lean	Temperature 50% conversion			max. NO conv.	temp. for max. NO conv.
	CO	C <sub>3</sub> H <sub>6</sub>	C <sub>3</sub> H <sub>8</sub>		
ZCA-600	297	410	568	0.9	569
C/ZA-600	325	434	518	2.2	520
ZCAT-600	370	484	539	0.8	451
C/ZAT-600	302	362	523	4.9	560

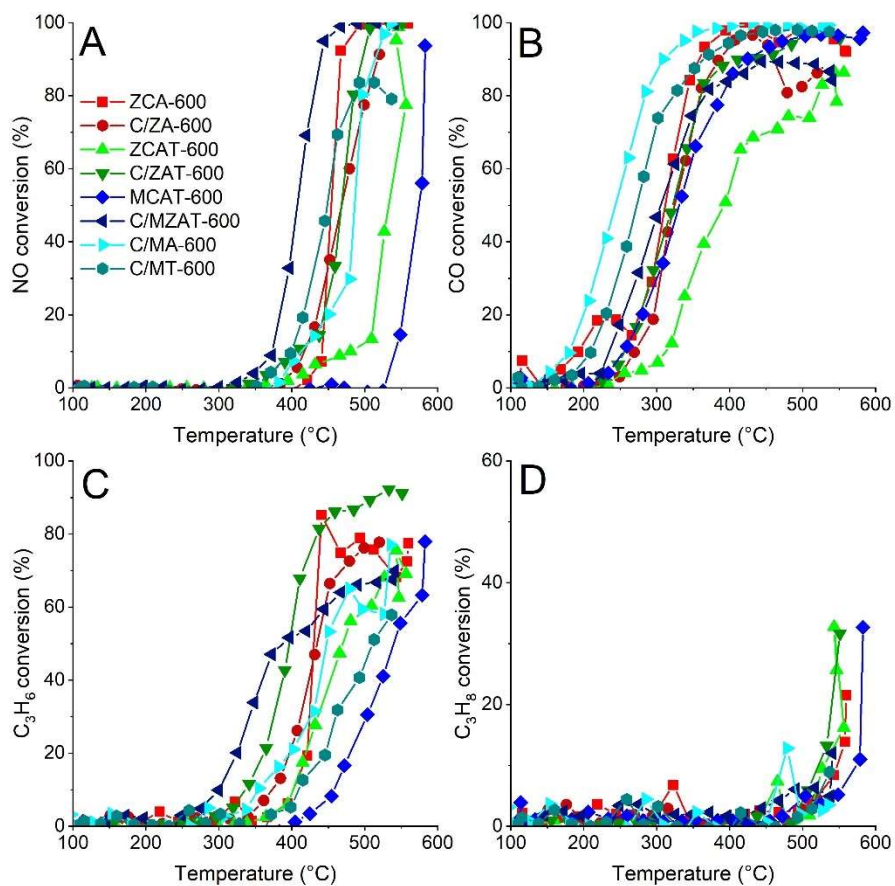
MCAT-600	311	525	570	2.3	573
C/MZAT-600	296	347	481	1.7	515
C/MA-600	231	401	481	10.7	470
C/MT-600	271	515	542	4.8	527
Comm. TWC	198	213	466	17.5	305
0.5% Pd/CeZrO <sub>2</sub>	188	247	521	18	368

839 *Supplementary Table 4: temperatures at which 50% conversion is reached and maximal NO conversion in the lean cycle. N.r.*

840 *= not reached.*

841

842 In the rich condition (Supplementary Figure 10), less O<sub>2</sub> is present so NO is more likely to  
843 oxidise CO and the hydrocarbons. This is clearly the case here, as NO conversion reaches  
844 nearly 100% in all the cases. C/MZAT-600 stands out with the lowest T<sub>50</sub>, while ZCAT-600 and  
845 MCAT-600 perform considerably worse. The T<sub>50</sub> for CO and propene are slightly higher, but  
846 the conversion of propane is definitely worse as none of the studied three-way catalysts are  
847 able to convert up to 50% propane under 550°C.



848

849 *Supplementary Figure 10: Temperature programmed conversion curves under rich conditions, A: NO conversion, B: CO*  
 850 *conversion, C: propene conversion, D: propane conversion.*

851

Rich	Temperature 50% conversion			
	CO	C <sub>3</sub> H <sub>6</sub>	C <sub>3</sub> H <sub>8</sub>	NO
ZCA-600	312	430	n.r.	454
C/ZA-600	325	435	n.r.	468
ZCAT-600	392	470	n.r.	533
C/ZAT-600	322	396	n.r.	468
MCAT-600	331	540	n.r.	574
C/MZAT-600	306	388	n.r.	407

C/MA-600	243	448	n.r.	486
C/MT-600	273	511	n.r.	449
Comm. TWC	240	333	441	286
0.5% Pd/CeZrO <sub>2</sub>	137	352	456	136

852 *Supplementary Table 54: temperatures at which 50% conversion is reached in the rich cycle. N.r. = not reached.*

853

854

### 855 Propene and propane discussion

856 The general trend that a higher  $T_{50}$  is associated with the presence of larger CuO crystallite  
857 sizes is not observed from the propene conversion profiles. The addition of manganese in the  
858 three-way catalysts does not improve the propene conversion much. The three-way catalysts  
859 C/ZAT-600 and C/MZAT-600 perform the best, showing every time the lowest  $T_{50}$  and  
860 achieving more than 90% conversion at the lowest temperatures, while the lowest catalytic  
861 conversions are found with MCAT-600 and C/MT-600. C/MA-600 on the other hand performs  
862 better and its profile is even similar to C/MZAT-600 in the last stoichiometric cycle. This  
863 demonstrates again that using alumina certainly represents an advantage over titania,  
864 probably because it behaves better as a mesoporous support than titania which further  
865 crystallizes into a rutile structure during the catalytic tests. This assumption is also supported  
866 by the superior activity of the C/ZAT-600 and C/MZAT-600, in which the solid state reaction  
867 between the ZnO and TiO<sub>2</sub> leads to the formation of Zn<sub>2</sub>TiO<sub>4</sub> thus inhibiting the crystallisation  
868 into a rutile phase.

869 Not much distinction can be made between the different three-way catalysts in the propane  
870 conversion profiles. Only C/MA-600 clearly shows the best performances compared to all the  
871 studied three-way catalysts, with its much lower  $T_{50}$ . Similarly as hypothesized for the CO and

872 propene conversion, this may be assigned to the presence of the amorphous alumina which  
 873 inhibits agglomeration of the copper and manganese, while the surface area enhances their  
 874 dispersion.

875

876 The purpose of the second stoichiometric cycle is to compare it with the first one, in order to  
 877 check whether the performance of the catalysts is declining or not after this long dynamic  
 878 procedure. For all the samples, except C/MA-600, the maximal NO conversion is lower, under  
 879 6%. Similar CO conversions as in the first catalytic cycle are observed. Surprisingly, some  
 880 catalysts even have a lower  $T_{50}$  compared to the first catalytic cycle. For example, the  $T_{50}$  for  
 881 MCAT-600 is 62°C lower than in the first cycle (314°C and 252°C). For the propene conversion  
 882 the situation is the same. There is a difference of 114°C for MCAT-600 between the first and  
 883 the second catalytic cycle (534°C and 420°C), while the propane conversion is very similar  
 884 between the two catalytic cycles.

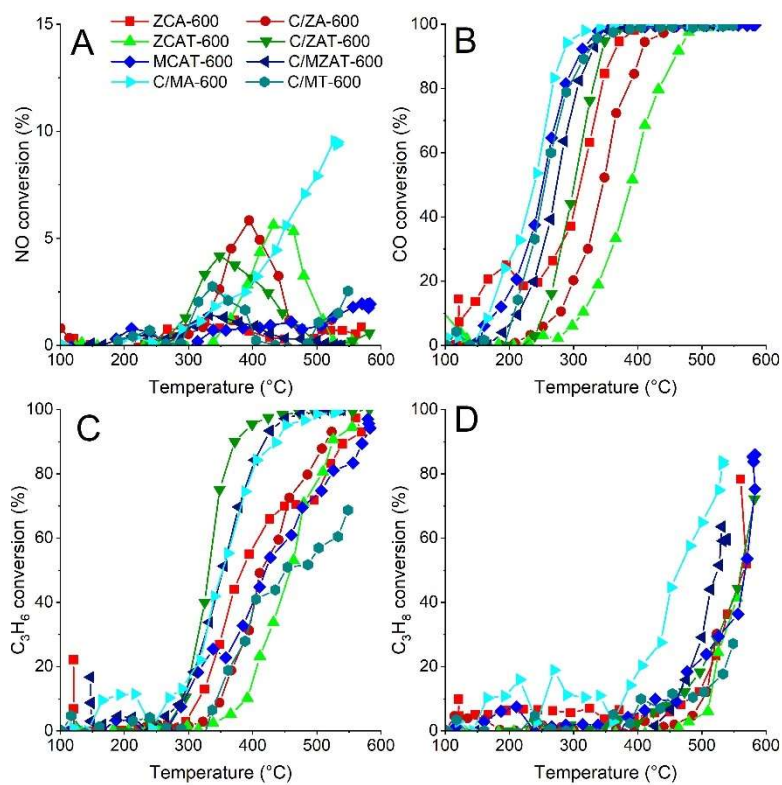
885

Stoic 2	Temperature 50% conversion			temp. max. for max. NO NO conv. conv.	
	CO	C <sub>3</sub> H <sub>6</sub>	C <sub>3</sub> H <sub>8</sub>		
ZCA-600	310	384	567	1.2	371
C/ZA-600	346	370	n.r.	5.8	395
ZCAT-600	390	459	n.r.	5.6	432
C/ZAT-600	301	332	561	4.2	348



MCAT-600	252	420	568	2	583
C/MZAT-600	274	353	524	1.4	332
C/MA-600	239	353	464	9.5	527
C/MT-600	255	452	n.r.	2.8	337
Comm. TWC	228	268	456	52	358
0.5% Pd/CeZrO <sub>2</sub>	157	231	507	39	259

886 *Supplementary Table 6: Temperatures at which 50% conversion is reached and maximal NO conversion in the second*  
887 *stoichiometric cycle N.r. = not reached.*



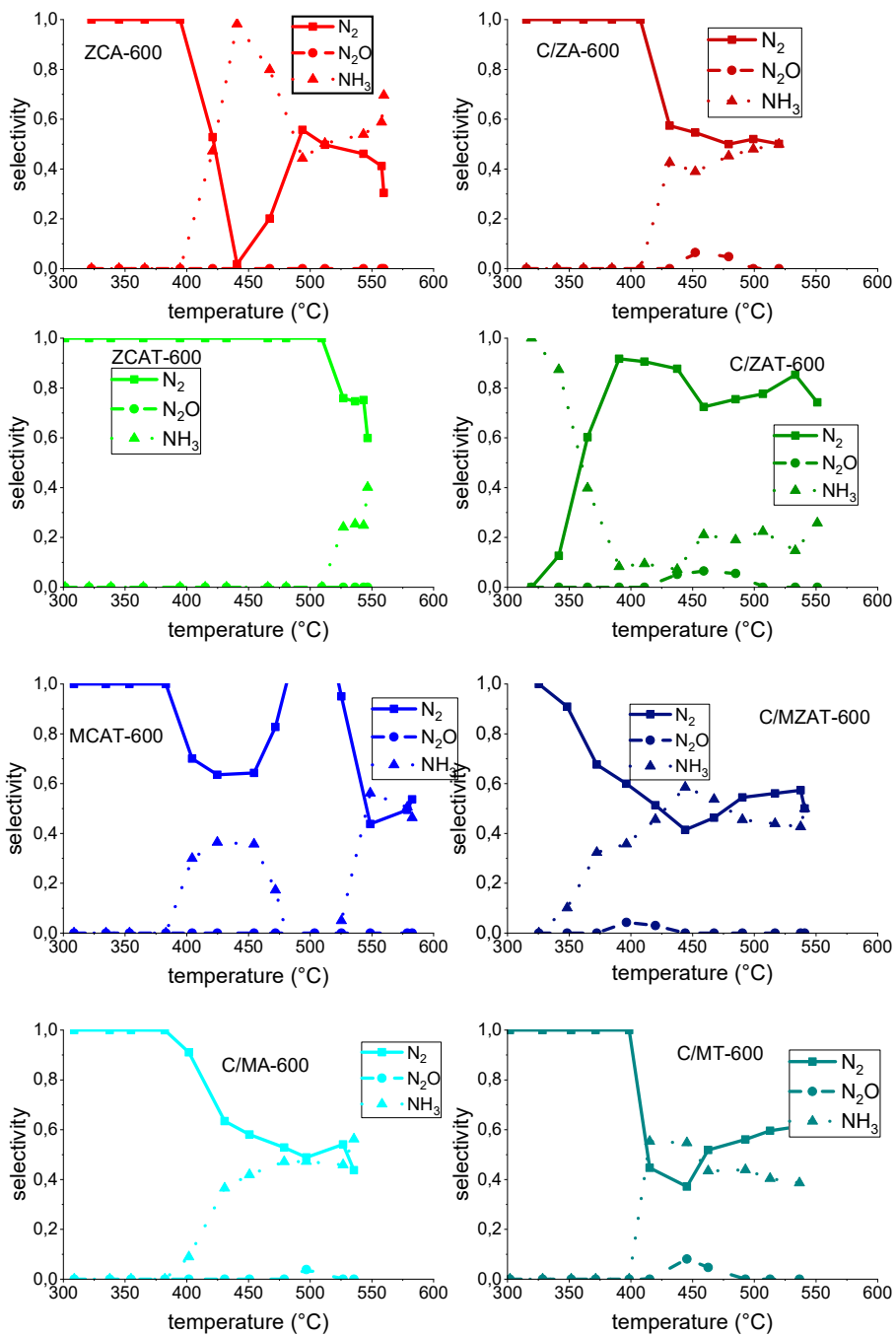
888  
889 *Supplementary Figure 11: Temperature programmed conversion curves under stoichiometric conditions, second cycle, A: NO*  
890 *conversion, B: CO conversion, C: propene conversion, D: propane conversion.*

891  
892

893 [NO conversion discussion](#)

894 In stoichiometric and lean conditions, all NO reduced during the test is selectively converted  
895 to N<sub>2</sub>. In the rich condition, NH<sub>3</sub> and N<sub>2</sub>O are also formed through side reactions. The selectivity  
896 of the NO conversion in the rich condition for all the catalysts is displayed in Supplementary  
897 Figure 12 and the corresponding yields can be found in Supplementary Figure 13. The  
898 selectivity towards N<sub>2</sub>O is very low for all the studied three-way catalysts, however, the  
899 selectivities towards N<sub>2</sub> and NH<sub>3</sub> are almost equal at high temperatures. Overall, the yields  
900 towards N<sub>2</sub> are higher for the catalysts containing titanium (ZCAT-600, C/ZAT-600, C/MT) as  
901 compared to the catalysts that do not contain titanium (ZCA-600, C/ZA-600, C/MA).

902



903

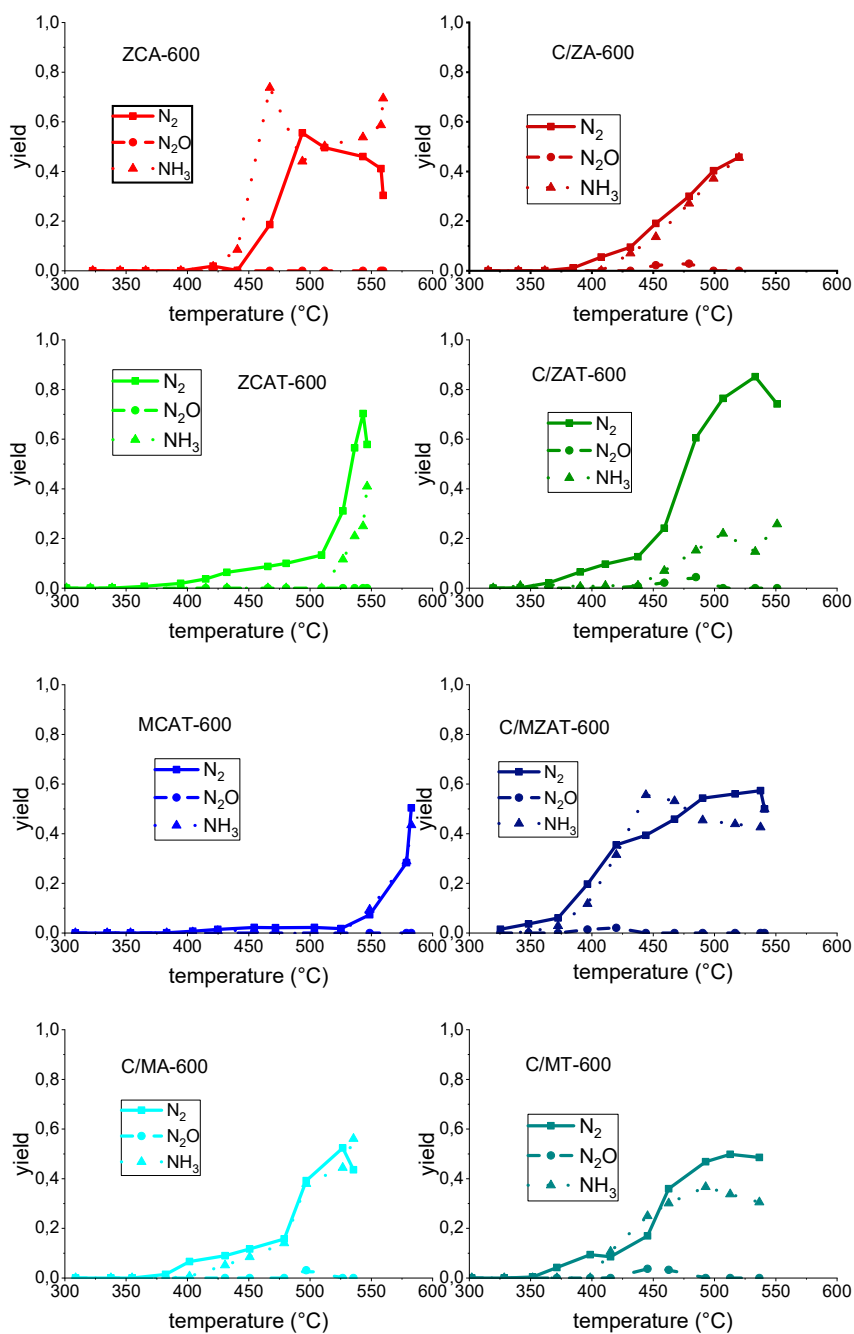
904

Supplementary Figure 12: Selectivity of NO towards N<sub>2</sub>, N<sub>2</sub>O and NH<sub>3</sub> during the rich condition for all catalysts.

905

906

907



908

909

Supplementary Figure 13: Yield of NO conversion towards  $N_2$ ,  $N_2O$  and  $NH_3$  for all catalysts.

910 In stoichiometric and lean conditions, the NO conversion profiles exhibit typical volcano plots.

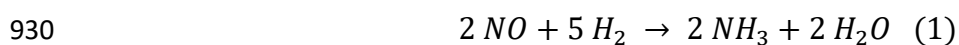
911 This occurs because the selectivity for the reactions with NO drops for reactions with  $O_2$ . In

912 this way, four local maxima for the four different reactants ( $H_2$ , CO, propene, propane) can

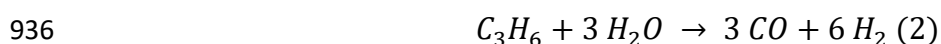
913 be expected, which explains the often strange shape of the NO conversion profiles. The

914 catalysts used here, under these specific conditions, seem to mainly convert NO with propene  
915 or H<sub>2</sub>. As an example, for the profile of C/MZAT-600 there is a sharp rise in the NO conversion  
916 at around 300°C, which is also the temperature where propene and H<sub>2</sub> conversion starts (H<sub>2</sub>  
917 conversion profile can be found in Supplementary Figure 14). Analysing the profiles of the first  
918 stoichiometric cycle, it suggests that the materials containing titanium produce higher  
919 maximal NO conversions. However, while C/MA-600 shows medium conversion at low  
920 temperatures, the NO conversion is much higher than for all the other catalysts at high  
921 temperatures. This can be related to the propene conversion observed in this case, being less  
922 steep than for example C/MZAT-600 and to the propane conversion, which starts at a lower  
923 temperature than for the other catalysts. In the lean cycle, the multiple volcano plots are more  
924 clear and again, the highest conversion is found for C/MA-600, but C/ZAT and C/MT still stand  
925 out to the other catalysts.

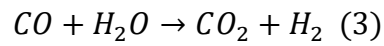
926 In the rich conditions, NH<sub>3</sub> formation is also possible (reaction 1) and the selectivity profiles  
927 demonstrate that this is certainly the case here. NH<sub>3</sub> formation is also undesired as it is  
928 poisonous for the catalyst, corrosive for the system and can form particulate matter (Wang et  
929 al., 2015; Nevalainen et al., 2018).



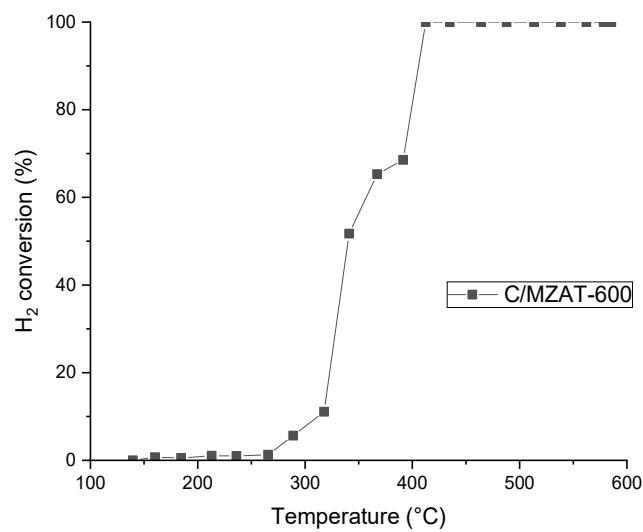
931 However, evaluating the H<sub>2</sub> conversion plots in the rich conditions (Supplementary Figure 15),  
932 it can be concluded that the H<sub>2</sub> conversion reaches 100% before the temperature at which the  
933 NO conversion starts, mainly due to the reaction with O<sub>2</sub>. A source of the extra H<sub>2</sub> required for  
934 NH<sub>3</sub> formation can be steam reforming of propene and the water-gas shift reaction (reaction  
935 2 and 3) (Mejía-Centeno et al., 2012; Nevalainen et al., 2018).



937



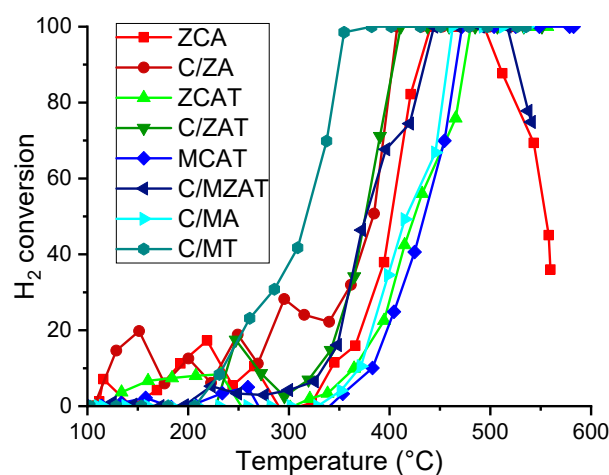
938 As concluded above, the three-way catalysts containing titanium have a higher yield for N<sub>2</sub> at  
939 high temperatures (500°C) than their titania-free correspondents. Yuan *et al.* postulate that  
940 the higher N<sub>2</sub> selectivity of a copper-titanium mixed oxide is due to the existence of more  
941 Lewis sites, which adsorb NO as nitrates on the catalyst surface (Yuan et al., 2014).



942

943

Supplementary Figure 14: H<sub>2</sub> conversion in the first stoichiometric cycle for C/MZAT-600.



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945

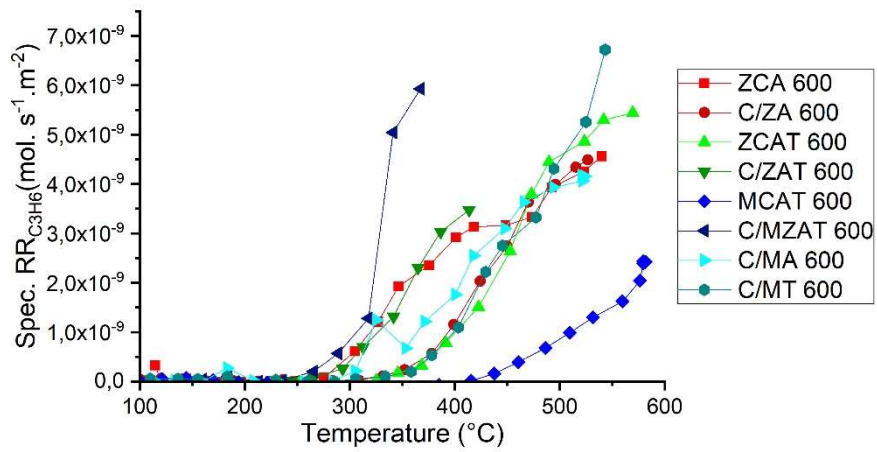
Supplementary Figure 15: H<sub>2</sub> conversion of the materials under rich conditions.

946

947 [Specific reaction rates and turnover frequency](#)

948 The conversion profiles presented earlier are proportional to the reaction rates for 200 mg of  
949 catalyst at different temperatures. Reaction rates can however also be expressed based on  
950 the specific surface area of the catalyst which makes sense as catalysis happens mostly on the  
951 surface. A large surface area will give a catalyst advantage over a catalyst with a smaller  
952 surface area. For this reason, a specific reaction rate is defined, which expresses the amount  
953 of converted molecules per second per  $\text{m}^2$  of catalyst surface area. Specific reaction rate  
954 profiles for CO, NO and propene in the first stoichiometric cycle can be found in  
955 Supplementary Figure 16Supplementary Figure 17Supplementary Figure 18. C/MT-600 and  
956 C/MZAT-600, who were already amongst the best performing catalysts, are standing here  
957 even more out from the others due to their relatively lower surface area. This shows again  
958 that the titania-containing catalysts have a lot of potential but that the small surface area due  
959 the crystallisation of titania is a drawback in these high temperature applications.

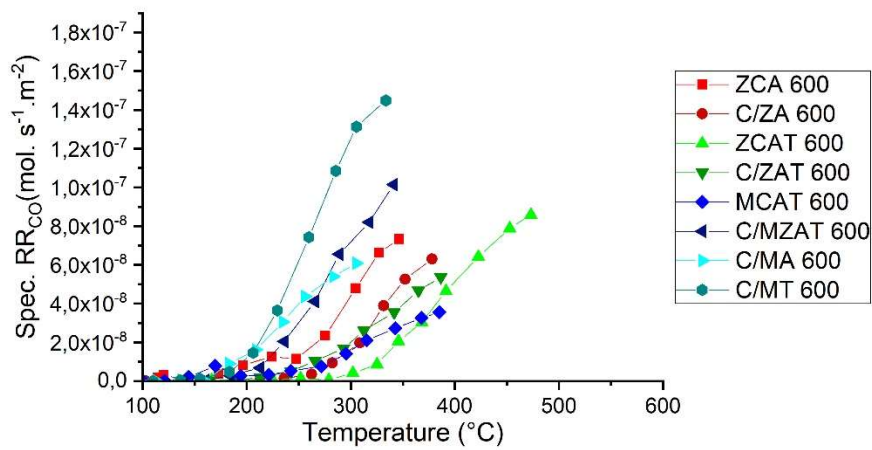
960 Another way to express reaction rates is based on the amount of reactive sites in the catalyst,  
961 which is called the turnover frequency (TOF). Here it is assumed that the reactive sites are  
962 only the Cu-atoms and that every Cu-atom in the material is accessible and taking part in the  
963 catalysis. A more accurate TOF would be probably somewhere between the presented TOF  
964 and the specific reaction rate as Cu-atoms buried deep into the bulk of the material will most  
965 likely not take part in the reactions. Profiles of the TOFs as a function of temperature for CO,  
966 NO and propene in the first stoichiometric cycle can be found as Supplementary Figure 19, 19  
967 and 20. As the Cu content for C/MZAT-600, C/MA-600 and C/MT-600 is much lower and their  
968 catalytic performance is very good, they stand out even more from the others in the TOF  
969 profiles. This emphasizes the fact that Cu deposited as small particles in a post-synthesis  
970 method is much more efficient than in a direct synthesis method.



971

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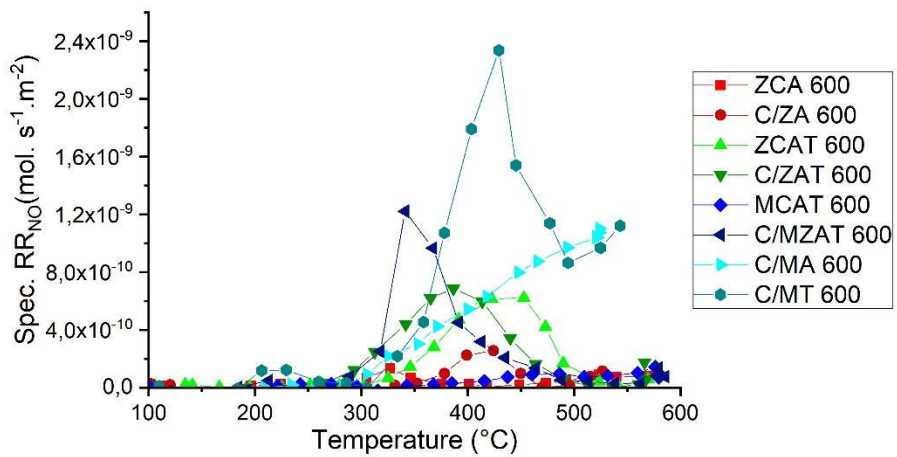
Supplementary Figure 16: Specific reaction rates for  $C_3H_6$  during the first stoichiometric cycle.



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974

Supplementary Figure 17: Specific reaction rates for CO during the first stoichiometric cycle.

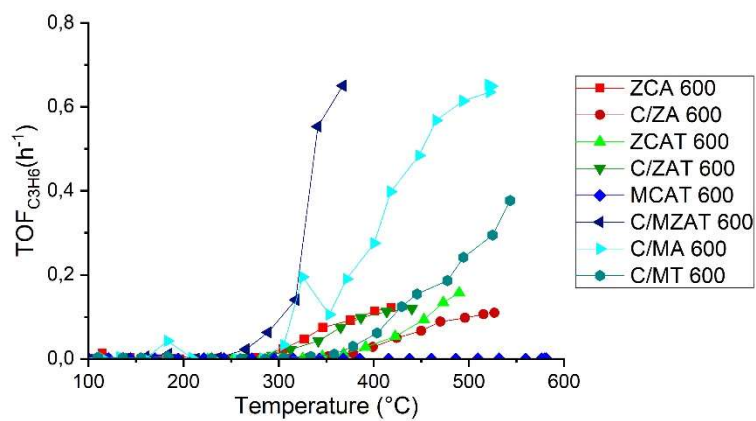


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Supplementary Figure 18: Specific reaction rates for NO during the first stoichiometric cycle.

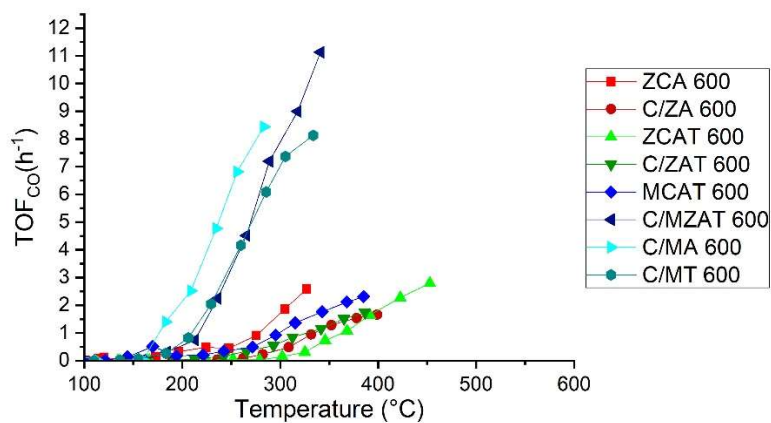




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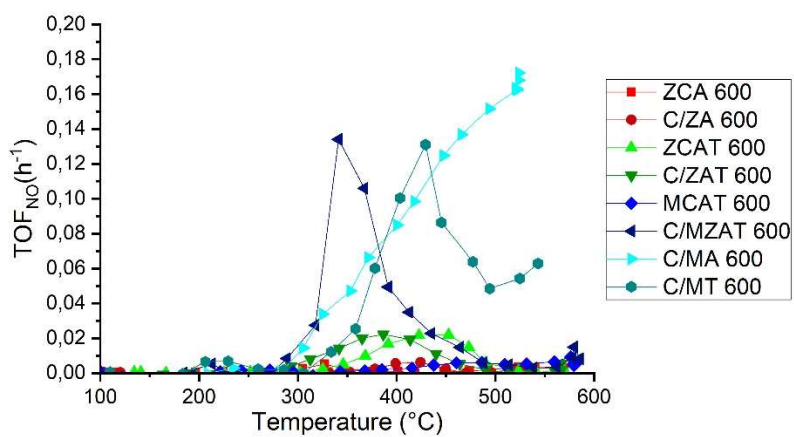
Supplementary Figure 19: Turn-over frequencies for  $C_3H_6$  during the first stoichiometric cycle.



979

980

Supplementary figure 20 Turn-over frequencies for CO during the first stoichiometric cycle.



981

982

Supplementary Figure 21: Turn-over frequencies for NO during the first stoichiometric cycle.

983

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## 985 Comparison with commercial catalysts

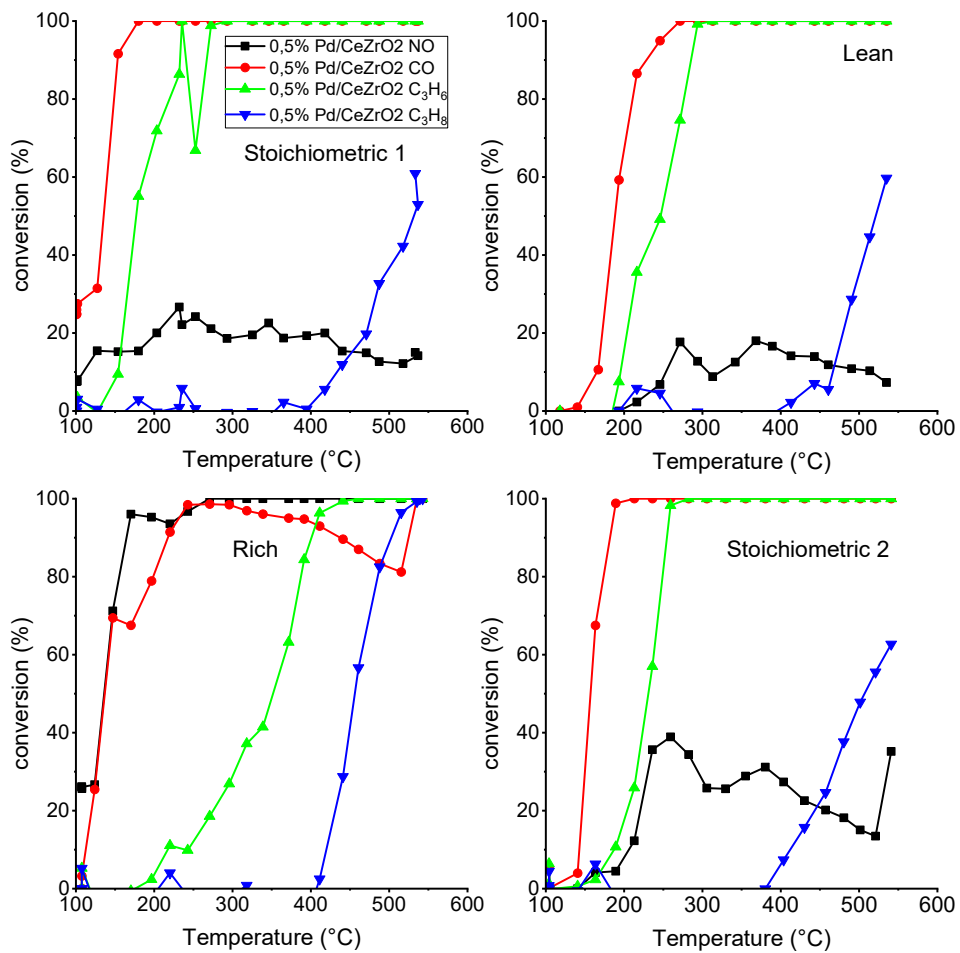
986 The benchmarking of the catalytic results with literature reports is not that straightforward as  
987 small differences in the experimental set-up or in the gas feed compositions may lead to  
988 significant differences in the catalytic performances. On the other hand, extensive research in  
989 the development and testing of three-way catalysts based on LaFeO<sub>3</sub> perovskite materials was  
990 performed by the group of P. Granger (Schön et al., 2015, 2017, 2018; Wu et al., 2019) where  
991 the catalytic testing was done using the same experimental set-up. The best performing non-  
992 PGM containing catalyst in those studies is La<sub>0.67</sub>Fe<sub>0.87</sub>Cu<sub>0.13</sub>O<sub>3</sub> and has a T<sub>50</sub> of 232°C for CO  
993 and 386°C for propene in the first stoichiometric cycle but in the second stoichiometric cycle  
994 this amounts to 255°C and 390°C. The present developed C/MA-600 three-way catalyst shows  
995 a comparable performance for CO conversion in the first stoichiometric cycle (T<sub>50</sub> = 239°C) and  
996 is able to maintain this performance in the second stoichiometric cycle (T<sub>50</sub> = 239°C). The T<sub>50</sub>  
997 of C/MA-600 for the propene conversion is even improving from the first to the second  
998 stoichiometric cycle (from 409 to 353°C), performing much better than the reported  
999 perovskite type material. Further, ZCA-600, C/ZA-600, C/ZAT-600 and C/MZAT-600 developed  
1000 in the present study also show lower T<sub>50</sub> than the above mentioned perovskite-based catalyst  
1001 for propene conversion in the second stoichiometric cycle. Remarkably, the current developed  
1002 three-way catalysts manage to reach in some cases almost 10% NO conversion in  
1003 stoichiometric conditions, while there is virtually no activity reported using the perovskite  
1004 type materials. In conclusion, the main advantages of the catalysts developed in this work are  
1005 the great CO and propene conversion abilities, as well as some NO conversion while also  
1006 showing a good stability over subsequent catalytic cycles.

1007 Ultimately, to further benchmark the performances of the three-way catalysts developed in  
1008 this study, two PGM containing commercial three-way catalysts provided by Johnson Matthey

1009 were tested in the same catalytic conditions, namely a 0.5 wt% Pd loaded onto CeZrO<sub>2</sub> (results  
1010 included in the Supplementary Figure 22) and a grounded piece of monolith of the commercial  
1011 three-way catalytic converter (comm. TWC, results included in Supplementary Figure 23),  
1012 respectively. For these commercial catalysts, the catalytic testing was performed in all three  
1013 experimental conditions (stoichiometric, lean, rich) as well as in a second stoichiometric cycle.

1014 It can be observed that for C/MA-600, the CO conversion is almost comparable to that of the  
1015 commercial TWC, but it is still almost 100°C higher than for Pd/CeZrO<sub>2</sub>, while the propene  
1016 conversion needs to be further improved to reach an efficiency comparable with the T<sub>50</sub> of  
1017 Pd/CeZrO<sub>2</sub>. Further, very good results are obtained for the propane conversion, e.g.  
1018 comparable with the efficiency achieved using the Pd/CeZrO<sub>2</sub> commercial catalyst. In terms of  
1019 NO conversion, lower performances are recorded as compared to the PGM containing  
1020 catalysts. However, this research demonstrates that the developed materials are very  
1021 promising for the use in TWC application as noble metal free catalysts, compared to the  
1022 classical catalysts composition. These new materials have proven to exhibit a considerably  
1023 good performance for the oxidation of CO, propene and propane.

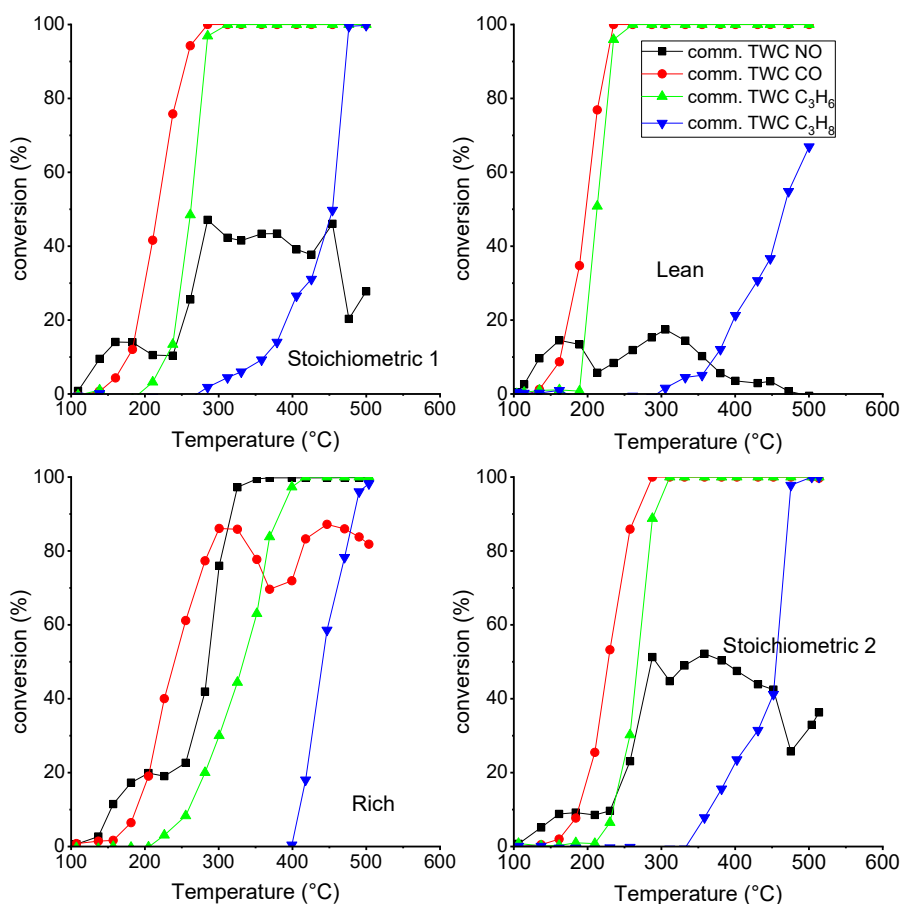
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1025

1026 *Supplementary Figure 22: Conversion profiles for 0.5% Pd/CeZrO<sub>2</sub>. Top left: first stoichiometric cycle, top right: lean cycle,*

1027 *bottom left: rich cycle, bottom right: second stoichiometric cycle.*



1028

1029 *Supplementary Figure 23: Conversion profiles for the commercial TWC catalyst. Top left: first stoichiometric cycle, top right:*

1030 *lean cycle, bottom left: rich cycle, bottom right: second stoichiometric cycle.*

1031 [Evaluation of the stability of the catalysts and structural investigation of the spent](#)

1032 [catalysts](#)

1033

1034 In order to verify the stability of the catalysts after the catalytic tests, for the entire series, the

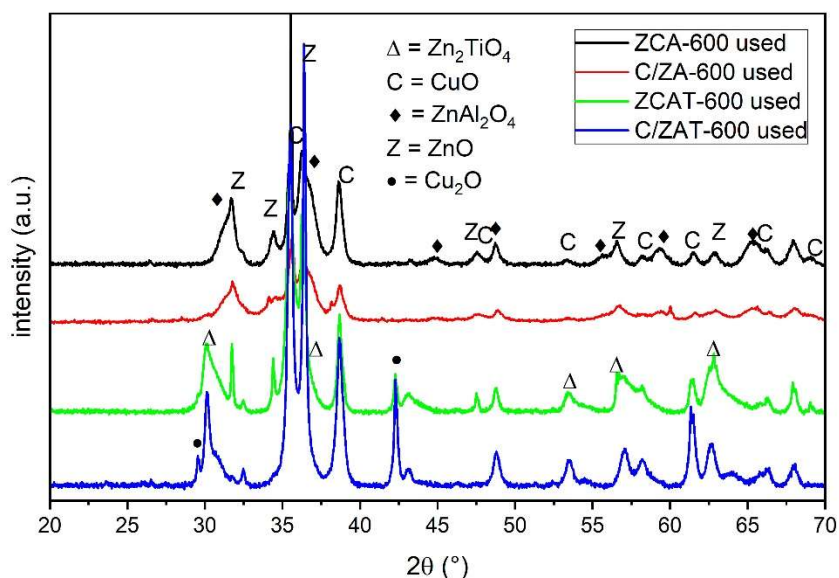
1035 XRD patterns were recorded for the used catalysts (Supplementary Figures 24 and 25). No

1036 structural modification was observed for the ZCA-600 and C/ZA-600 after the catalytic tests,

1037 as the XRD patterns are very comparable, while in the case of ZCAT-600 and C/ZAT-600, the

1038 presence of  $\text{Cu}_2\text{O}$  is noticed after the real conditions of the catalytic tests.

1039 Interestingly, the XRD patterns of MCAT-600 before and after the catalytic test associated with  
 1040 the spinel-like structure (initially  $\text{Cu}_{1.5}\text{Mn}_{1.5}\text{O}_4$ ) are shifted. This shift is clearly noticed for the  
 1041 (115) and (044) planes at  $57.6^\circ$  and  $63.3^\circ$  (iError! No se encuentra el origen de la  
 1042 referencia.26). The newly formed structure is closer to  $\text{CuMn}_2\text{O}_4$  (PDF 00-034-1400), while  
 1043  $\text{TiO}_2$  (rutile) and  $\text{CuO}$  crystal phases also appear now. For the other three manganese  
 1044 containing materials, a  $\text{CuO}$  phase is still not detected in the XRD patterns. Similar to the initial  
 1045 analysis, C/MZAT-600 consists again out of the mixed oxide phases of manganese ( $\text{CuMn}_2\text{O}_4$   
 1046 and  $\text{ZnMn}_2\text{O}_4$ ), while two new structures appear, namely rutile and a slightly different titanate  
 1047 phase, with an ilmenite ( $\text{FeTiO}_3$ , PDF 00-029-0733) structure. This ilmenite-like system is in  
 1048 this case likely a Zn-doped  $\text{MnTiO}_3$  composition, for example  $\text{Mn}_x\text{Zn}_{1-x}\text{TiO}_3$ . After the catalytic  
 1049 test, C/MA-600 patterns present features belonging only to  $\text{Mn}_3\text{O}_4$ , while  $\text{Mn}_2\text{O}_3$  is no longer  
 1050 detected, as before the test. Contrary, the corresponding C/MT-600 consists outof similar  
 1051 components before and after the catalytic test ( $\text{Mn}_3\text{O}_4$ ,  $\text{Mn}_2\text{O}_3$ , rutile).

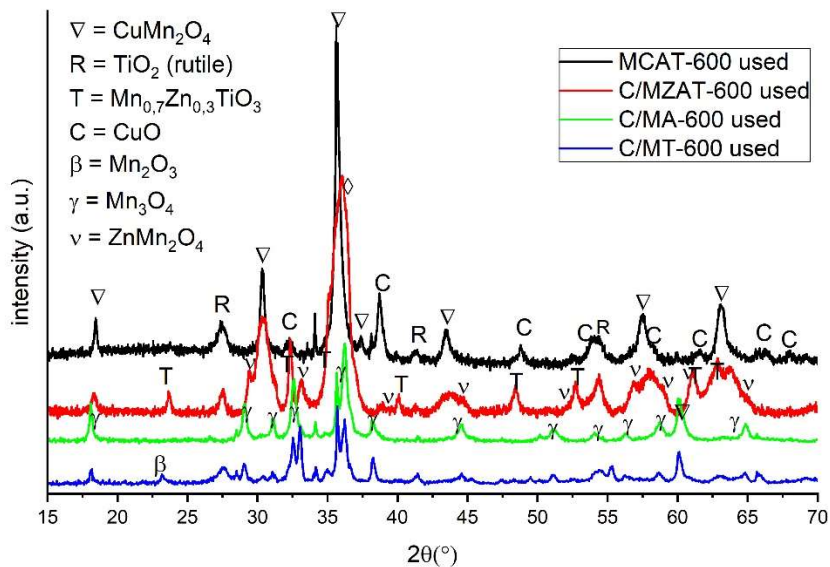


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Supplementary figure 24: XRD-patterns of ZCA-600, ZCAT-600, C/ZA-600 and C/ZAT-600 after the catalytic test.

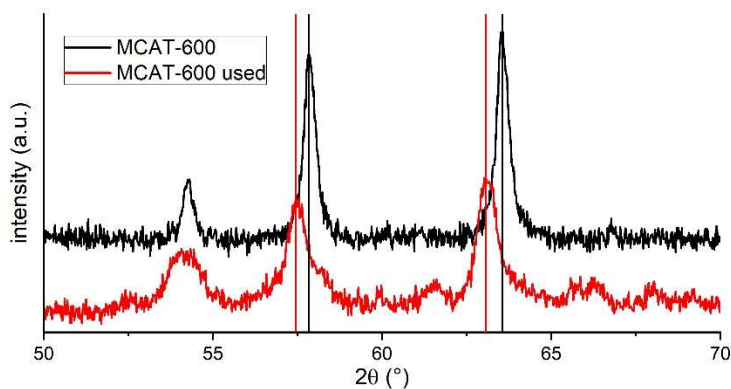
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1056 *Supplementary figure 25: XRD-patterns of MCAT-600, C/MZAT-600, C/MA-600 and C/MT-600 after the catalytic test.*

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1059 *Supplementary figure 26: Detail of the XRD-patterns of MCAT-600 before and after the catalytic test.*

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