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# 1     **Recent Progress in Developing Non-Noble Metal Based Photocathodes for** 2                                   **Solar Green Hydrogen Production**

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## 9 10    **Abstract**

11           Photocathodes play a vital role in photoelectrocatalytic water splitting by acting as  
12    catalysts for reducing protons to hydrogen gas when exposed to light. Recent advancements in  
13    photocathodes have focused on addressing the limitations of noble metal-based materials.  
14    These noble metal-based photocathodes rely on expensive and scarce metals such as platinum  
15    (Pt) and gold (Au) as co-catalysts or ohmic back contacts, respectively, rendering the final  
16    system less sustainable and costly when applied at scale. This mini-review summarizes the  
17    important recent progress in the development of non-noble metal-based photocathodes and  
18    their performance in the hydrogen evolution reaction (HER) during photoelectrochemical  
19    water splitting. These advancements bring non-noble metal-based photocathodes closer to their  
20    noble metal-based counterparts in terms of performance, thereby paving the way forward  
21    towards industrial-scale photo-electrolysers or photoelectrochemical cells for green hydrogen  
22    production.

23  
24    **Keywords:** Non-noble metal photocathodes, solar green hydrogen production,  
25    photoelectrochemical water splitting, sustainable energy, renewable energy

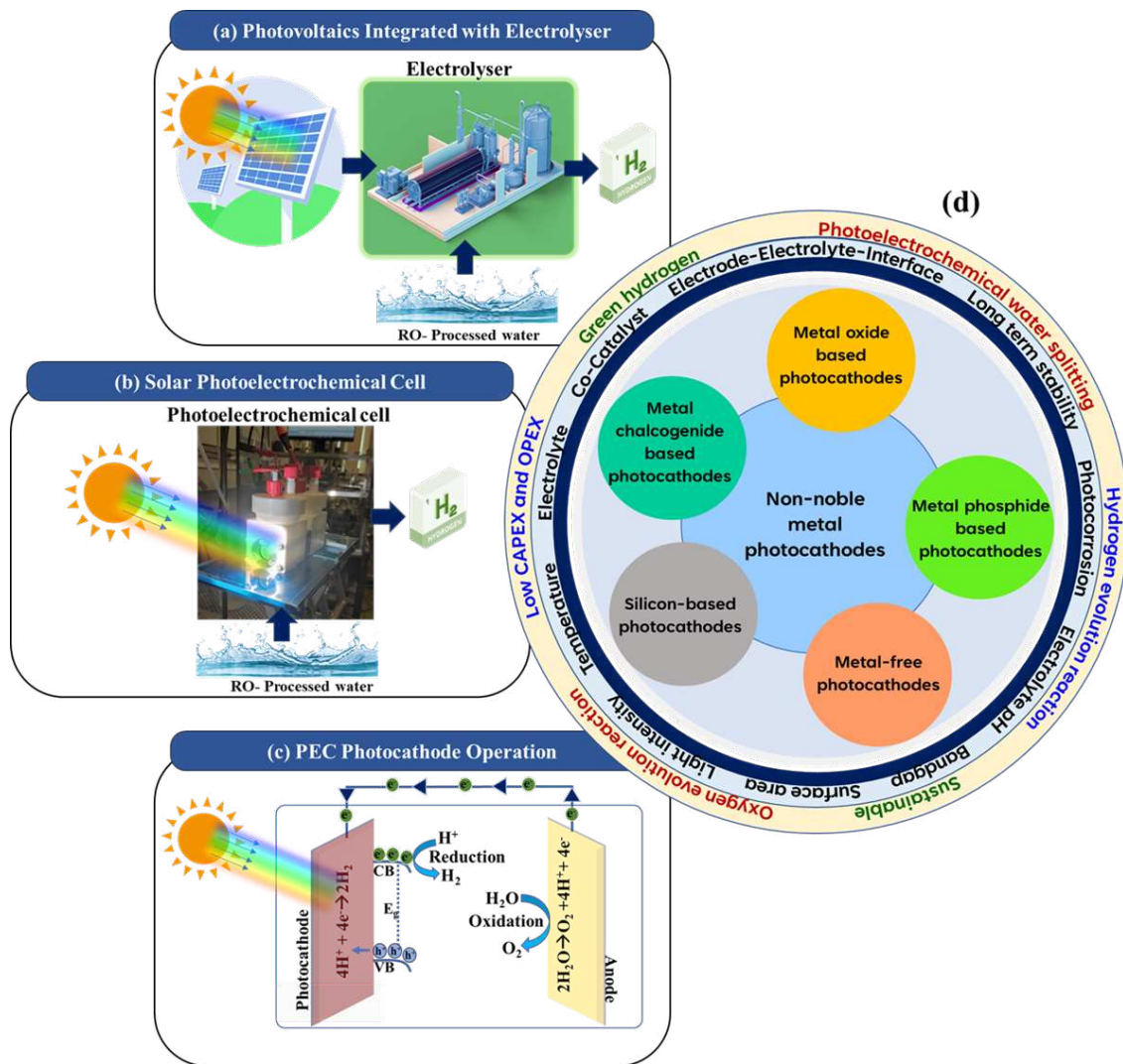
## 28 1. Introduction

29 In view of the current state-of-the-art, the most technologically mature route to produce  
30 solar green hydrogen is electrolysis powered by renewable electricity (photovoltaics (PV) or  
31 wind). As both electrolysis and PV are two market-ready technologies, the majority of the  
32 green hydrogen projects implement an integrated system of the two (**Figure 1 (a)**) [1].  
33 However, for green hydrogen to become a mainstream fuel and chemical raw material, the cost  
34 of hydrogen production has to be lowered significantly from the current >5\$ per kg to <2 \$ per  
35 kg [2]. In that perspective, direct photoelectrochemical (PEC) water splitting for hydrogen  
36 production is gaining increasing importance (**Figure 1 (b)**). Firstly, the direct conversion of  
37 solar energy and water into hydrogen and oxygen in PEC cells excludes extra PV systems. This  
38 process occurs at the semiconductor-electrolyte interface, where the semiconductor  
39 photoelectrodes absorb solar energy that leads to the excitation of electrons from the valence  
40 band to the conduction band generating electron-hole pairs. This creates a potential difference  
41 between the photoelectrodes and the electrolyte solution driving the redox reactions involved  
42 in water splitting. The electrons are then transported to the cathode, where they reduce protons  
43 ( $H^+$ ) in water to form hydrogen gas ( $H_2$ ). Simultaneously, the holes are transported to the anode,  
44 where they oxidize water molecules ( $H_2O$ ) to produce oxygen gas ( $O_2$ ). The use of earth-  
45 abundant non-noble metal based photoelectrodes is not only significantly cheaper but can also  
46 be advantageous from a stability and maintenance point of view. Although PEC water splitting  
47 could be at the basis of a new generation of green hydrogen production technologies [3, 4] its  
48 large-scale practical implementation (*i.e.*, going all the way up to megawatt (MW) scale) is  
49 generally considered to be hindered by mainly the poor durability of photoanodes, that catalyze  
50 the oxygen evolution reaction (OER) [5]. It should be noted though, that photocathodes, which  
51 catalyze the hydrogen evolution reaction (HER), also largely suffer from photocorrosion and  
52 poor reaction kinetics [6]. While much research has focused on improving the sluggish OER

53 kinetics by modifying photoanodes, less attention has been paid to photocathodes, even though  
54 they are an equally critical component in the full system. This review aims to highlight recent  
55 efforts (*i.e.*, limited to the past five years) to improve photocathodes utilizing earth-abundant  
56 materials for sustainable hydrogen production through PEC water splitting.

57 Photocathodes, by virtue of being *p*-type semiconductor materials, generate holes as the  
58 primary charge carriers under solar illumination in parallel to *n*-type photoanodes that aim at  
59 generating electrons. This brings about a potential difference that drives a photocurrent in the  
60 external circuit, thereby facilitating HER at the cathode and OER at the anode, respectively  
61 (**Figure 1 (c)**). To achieve efficient hydrogen production, it is crucial to have high HER  
62 efficiency and selectivity, which largely depends on the properties of the photocathode  
63 material, including its band structure, electronic conductivity, and surface characteristics [7-9]  
64 shown in **Figure 1 (c)** and **(d)**, what makes the fabrication of a photocathode challenging is the  
65 fact that the photocathode has to fulfil two primary requirements in its band structure. Firstly,  
66 the band gap of the photocathode must be sufficiently smaller than that of the photoanode so  
67 that the low energy part of the solar spectrum that transmits through the photoanode can still  
68 excite electrons in the photocathode (unless a reactor design is employed wherein both  
69 photoelectrodes are illuminated from both sides). Secondly, the alignment of the bands should  
70 be such that the Fermi-level of the photoanode is higher than the Fermi-level of the  
71 photocathode in order for the electrons to flow in the direction as shown. Thus, the development  
72 of photocathodes meeting these criteria to complement the photoanodes is inherently  
73 challenging from the perspective of fundamental material properties. When one also considers  
74 the surface catalytic activity (often imparted by a co-catalyst) and chemical stability required  
75 for the entire electrochemical process to take place, it becomes increasingly challenging. This  
76 work critically reviews recent advances in photocathode development for PEC water splitting  
77 without the use of noble metals. The focus of this review is exclusively on recent reports on

78 photocathodes that demonstrate promising performance even without the incorporation of  
 79 noble metals.



80

81 **Figure 1.** (a) Presently employed systems that combine electrolysers and photovoltaic technology to generate  
 82 green hydrogen from solar energy. (b) Photoelectrochemical cells operating directly under sunlight without  
 83 requiring photovoltaic panels. (c) Schematic showing the operation of a photocathode in a PEC cell. (d) Schematic  
 84 illustration of various non-noble metal-based photocathodes, categorized by their material composition. It also  
 85 showcases influencing experimental parameters and advantages of PEC water splitting in the inner and outer rings,  
 86 respectively.

87

88 Over the years, noble metal-based photocathodes, particularly those incorporating  
 89 platinum (Pt) [10] and Ruthenium (Ru) in the form of RuO<sub>x</sub> as co-catalyst [11], have been  
 90 extensively utilized in PEC hydrogen production and water splitting due to their excellent  
 91 catalytic activity and stability. For instance, a study conducted by Chen *et al.* demonstrated a  
 92 Pt-based photocathode with an earth-abundant semiconductor multi-junction composed of

93 Cu<sub>2</sub>O/ZnO/TiO<sub>2</sub> as the photo-absorber, achieving a high photocurrent density of 8.2 mA cm<sup>-2</sup>  
94 [12]. Another study by Tong *et al.* introduced a Pt<sub>1</sub>/N–Ni co-catalyst integrated into a Cu<sub>2</sub>O  
95 photocathode, which not only enhanced charge transfer kinetics at the electrode-electrolyte  
96 interface but also reduced the overpotential for hydrogen evolution. This resulted in an  
97 impressive photocurrent density of 11.9 mA cm<sup>-2</sup> at 0 V vs RHE [13]. These investigations  
98 demonstrate that co-catalysts effectively lower the required overpotential for hydrogen  
99 evolution, thereby improving the overall efficiency and stability of the photocathode.  
100 Additionally, various semiconductor photocathode designs such as those of copper including  
101 oxides [14], sulfides [15], and others [16] have employed thin gold (Au) films as back contacts  
102 between the semiconductor layer and the substrate. These gold films serve the dual purpose of  
103 reflecting transmitted light for enhanced energy utilization, and charge collection, facilitating  
104 efficient charge separation, improved charge transfer kinetics, and overall long-term stability.

105 The use of non-noble metal-based materials in the construction of photocathodes,  
106 serving different functions such as HER co-catalysts, protective layers, and hole selective  
107 layers, is essential for shifting from costly and scarce noble metal-containing photocathodes in  
108 water splitting and hydrogen production. Efficient and practical alternative designs and  
109 constructions of photocathodes should possess desirable characteristics such as high  
110 photocurrent density, low onset potential and overpotential, and good stability. However, the  
111 other side of the medal is that non-noble metal-based photocathodes can be vulnerable to  
112 corrosion and degradation [17, 18]. To address this challenge, researchers have explored  
113 various strategies, including co-catalyst impregnation [7], surface modification [19], doping  
114 [20], and heterojunction fabrication [21], to enhance the stability and catalytic activity of non-  
115 noble metal-based photocathodes.

## 116 2. Non-noble metal-based photocathode architectures

117           The increasing demand for sustainable hydrogen production has sparked a shift towards  
118 development of non-noble metal-based photocathodes rather than their noble metal  
119 counterparts. This transition is driven by the abundance, cost-effectiveness, tuneable  
120 properties, stability, and environmental advantages of non-noble metal materials, particularly  
121 as we envision large-scale hydrogen production applications. Recent advances in non-noble  
122 metal-based photocathodes for water splitting and hydrogen production can be categorized  
123 based on the primary semiconducting material, as depicted in graphical scheme **Figure 1 (d)**.  
124 **Table 1** provides a summary of all non-noble metal-based photocathode architecture covered  
125 in this review. Subsequent sections that correspond to the earlier scheme categorisation, will  
126 delve into the specific roles played by different non-noble metal materials integrated into these  
127 photocathodes and how they contribute to improved performance.

**Table 1.** An overview of all non-noble metal-based photocathode studies presented in this review.

Photocathode	Co-catalyst	Protective layer/ Hole transport layer	Method of preparation	Photocurrent density (mA/cm <sup>2</sup> )	Electrolyte	Stability	ABPE (%)	IPCE (%)	STH (%)	Light source used	Ref
<b>Metal oxide-based photocathodes</b>											
Multi-junction NiO <sub>x</sub> /Cu <sub>2</sub> O/AZO/MoO <sub>x</sub>	MoO <sub>x</sub>	AZO	Electrodeposition and RF magnetron sputtering	6.1 mA/cm <sup>2</sup> at 0 V vs RHE	1 M Na <sub>2</sub> SO <sub>4</sub> + 0.1 M KPi (pH 4.9 ± 0.1)	6 h at 0 V	1.75 % at 0.5 V vs RHE	60 % at 300 to 500 nm	-	1 Sun	[22]
TiO <sub>2</sub> /ZnO/Cu <sub>2</sub> O	-	TiO <sub>2</sub> /ZnO	Electrodeposition and dip coating	0.18 mA/cm <sup>2</sup> at -0.2 V vs Ag/AgCl	0.1M Na <sub>2</sub> SO <sub>4</sub> (pH 6.8)	20 min at -0.2 V vs Ag/AgCl	-	-	-	2 Sun	[21]
Cu <sub>2</sub> O/Ni-CuBTC	Ni-CuBTC	-	Electrodeposition and Hydrothermal	1.51 mA/cm <sup>2</sup> at 0 V vs RHE	0.5 M Na <sub>2</sub> SO <sub>4</sub> + 0.2 M PBS (pH 5)	20 min at 0.5 V vs RHE	-	-	-	1 Sun	[7]
Cu <sub>2</sub> O/MoS <sub>2</sub>	-	MoS <sub>2</sub>	Electrodeposition and spin coating	6.5 mA/cm <sup>2</sup> at -0.2 V vs RHE	0.5M Na <sub>2</sub> SO <sub>4</sub> (pH 6.7)	1 h at 0 V vs RHE	-	27 % at 400 nm, 0.2 V vs RHE	-	1 Sun	[23]
Cu/Al/Cu <sub>2</sub> O	-	Cu/Al	Thermal evaporation and electrodeposition	2.16 mA/cm <sup>2</sup> at 0 V vs RHE	0.1 M Na <sub>2</sub> SO <sub>4</sub> (pH 7)	2 h at 0 V vs RHE	-	-	-	1 Sun	[4]
Co <sub>3</sub> O <sub>4</sub>	-	-	Hydrothermal	1.15 mA/cm <sup>2</sup> at -0.4 V vs Ag/Agcl	0.1 M H <sub>2</sub> SO <sub>4</sub> (pH 1)	-	-	-	-	1 Sun	[24]
Co <sub>3</sub> O <sub>4</sub> /CuO	CuO	-	Electrodeposition	6.5 mA/cm <sup>2</sup> at -0.3 V vs SCE	0.5 M Na <sub>2</sub> SO <sub>4</sub>	30 min at 0 V	-	-	-	1 Sun	[25]



Photocathode	Co-catalyst	Protective layer/ Hole transport layer	Method of preparation	Photocurrent density (mA/cm <sup>2</sup> )	Electrolyte	Stability	ABPE (%)	IPCE (%)	STH (%)	Light source used	Ref
Fe-Ni co-doped <b>Co<sub>3</sub>O<sub>4</sub></b>	Co-doped Fe-Ni	-	Spray pyrolysis	13.6 mA/cm <sup>2</sup> at -1 V	0.3 M Na <sub>2</sub> SO <sub>4</sub>	1 h at -1 V	0.34 % at 0.28 V and 636 nm	42 % at 405 nm	11.37	1 Sun	[26]
Cu doped <b>NiO</b>	Cu	-	Sol-gel spin coating	6.97 mA/cm <sup>2</sup> at -1.5 V vs Ag/AgCl	0.1 M NaOH	-	-	-	-	30 w UV	[27]
Cu: NiO/ <b>CuBi<sub>2</sub>O<sub>4</sub></b>	-	Cu:NiO (HTL)	Spray pyrolysis and electron beam evaporation	2.83 mA/cm <sup>2</sup> at 0.6 V vs RHE	0.3 M K <sub>2</sub> SO <sub>4</sub> +0.2 M phosphate buffer (pH 6.65)	2 h at 0.6 V vs RHE	-	40 % at 400 nm (0.6 V)	-	1 Sun	[28]
<b>CuBi<sub>2</sub>O<sub>4</sub></b> /CuO	CuO	-	Electrodeposition	1.87 mA/cm <sup>2</sup> at 0.6 V vs SHE	0.1 M NaOH (pH 13)	6 h at 0.6 V vs SHE	-	16 % at 400 nm	-	1 Sun	[29]
<b>Metal Chalcogenide based photocathodes</b>											
<b>MoS<sub>2</sub>/Sb<sub>2</sub>S<sub>3</sub></b>	-	MoS <sub>2</sub>	Sublimation and Sputtering	10 mA/cm <sup>2</sup> at 0 V vs RHE	0.5 M Na <sub>2</sub> SO <sub>4</sub> (pH 6.5)	2 h at 0.2 V vs RHE	-	33 % at 800 nm	-	1 Sun	[30]
<b>MoS<sub>2</sub>/TiO<sub>2</sub></b> nanorods	-	-	Hydrothermal	5.2 mA/cm <sup>2</sup> at -0.4 V vs RHE	0.35 M Na <sub>2</sub> S + 0.25 M Na <sub>2</sub> SO <sub>3</sub>	5 cycles of 5h at -0.4 V vs RHE	-	-	-	1 Sun	[31]
<b>Cu<sub>2</sub>BaSnS<sub>4</sub></b>	-	-	Solgel spin coating	6.805 mA/cm <sup>2</sup> at 0 V vs RHE	0.5 M Na <sub>2</sub> S (pH 12.9)	-	10 % at 0.1 V	-	-	4.7 mW/cm <sup>2</sup> LED	[32]
Mo/ <b>Cu<sub>2</sub>BaSnS<sub>4</sub></b>	-	Mo	Sputtering and spin coating	4 mA/cm <sup>2</sup> at 0 V vs RHE	0.1 M Na <sub>2</sub> SO <sub>4</sub> (pH 7)	2 h at 0 V vs RHE	-	-	-	1 Sun	[33]
<b>Cu<sub>2</sub>S/ZnO</b> nanorods	-	ZnO	Hydrothermal	0.64 mA/cm <sup>2</sup> at 0 V vs RHE	1 M KCl (pH 5.97)	-	-	10.5 % at 450 nm	-	1 Sun	[34]

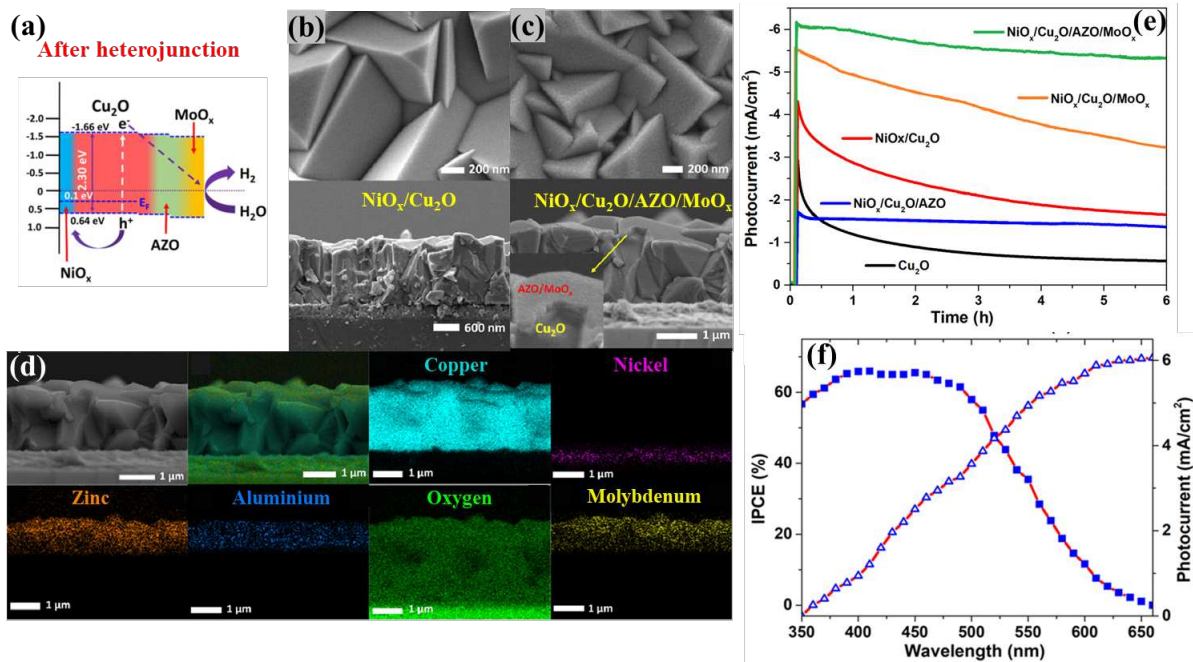
Photocathode	Co-catalyst	Protective layer/ Hole transport layer	Method of preparation	Photocurrent density (mA/cm <sup>2</sup> )	Electrolyte	Stability	ABPE (%)	IPCE (%)	STH (%)	Light source used	Ref
ZnO/Cu <sub>2</sub> S	-	ZnO	Electrodeposition, SILAR method	0.095 mA/cm <sup>2</sup> at 0 V vs Ag/AgCl	0.5 M Na <sub>2</sub> SO <sub>3</sub> (pH 10)	-	-	-	-	1 Sun	[35]
Dimensions dependent SnS nanocrystals	-	-	Hot and cold injection method	160 μA/cm <sup>2</sup> at -0.085 V vs RHE	0.5 M H <sub>2</sub> SO <sub>4</sub>	-	-	2.74 % at 600 nm	-	1 Sun	[36]
Bi <sub>2</sub> S <sub>3</sub> -Cu <sub>3</sub> BiS <sub>3</sub>	-	-	CBD and spin coating	7.8 mA/cm <sup>2</sup> at 0 V vs RHE	0.5 M Kpi (pH 7)	21 h at 0 v vs RHE	1.83 % at 0.45 V	20 % at 500 nm	2.33	1 Sun	[37]
CuInS <sub>2</sub> /MoS <sub>2</sub>	-	MoS <sub>2</sub>	Electrodeposition and sulfurization	8.74 mA/cm <sup>2</sup> at -0.1 V vs RHE	1M HClO <sub>4</sub> (pH 0.9-1.1)	-	-	Above 25 % at 400 to 700 nm	-	1 Sun	[38]
<b>Metal phosphide-based photocathodes</b>											
Co <sub>2</sub> P- Si inverted pyramids-	-	-	Electrochemical etching and drop casting	35.2 mA/cm <sup>2</sup> at 0 V vs RHE	0.5 M H <sub>2</sub> SO <sub>4</sub> (pH 0.3)	150 h at 0 V vs RHE	-	76 % at 600 nm	-	1 Sun	[39]
TiO <sub>2</sub> /GaP		TiO <sub>2</sub>	ALD		0.2 M PBS (pH 7)	-	-		-	405 nm LED 40-120 μW/cm <sup>2</sup>	[40]
<b>Silicon based photocathodes</b>											
p-Si-NiFe codoped In <sub>2</sub> S <sub>3</sub>	NiFe- In <sub>2</sub> S <sub>3</sub>	-	Hydrothermal	80.9 mA/cm <sup>2</sup> at -1.3 V vs RHE	0.5 M Na <sub>2</sub> SO <sub>3</sub>	2 h at 0 V vs RHE	-	-	-	1 Sun	[41]
MoS <sub>2</sub> /p-Si	MoS <sub>2</sub>	-	Etching and in situ assembly	28.2 mA/cm <sup>2</sup> at 0 V vs RHE	0.5 M H <sub>2</sub> SO <sub>4</sub> (pH 0.6)	60 h at 0 V vs RHE	-	-	-	1 Sun	[42]

Photocathode	Co-catalyst	Protective layer/ Hole transport layer	Method of preparation	Photocurrent density (mA/cm <sup>2</sup> )	Electrolyte	Stability	ABPE (%)	IPCE (%)	STH (%)	Light source used	Ref
3D textured graphene/p-Si	Graphene	-	CVD and EVA graphene transfer	32.5 mA/cm <sup>2</sup> at 0 V vs RHE	1 M HClO <sub>4</sub> (pH 0)	80 h at -0.3 V	-	-	-	1 Sun	[43]
<b>Metal-free photocathodes</b>											
g-C <sub>3</sub> N <sub>4</sub> /WS <sub>2</sub> @rGONR <sub>x</sub>	WS <sub>2</sub> @rGO NR <sub>x</sub>	-	Hydrothermal and thermal decomposition	47 mA/cm <sup>2</sup> at -0.7 V vs RHE	0.5 M H <sub>2</sub> SO <sub>4</sub> (pH 1)	1 h at -0.6 V vs RHE	-	-	-	1 Sun	[44]
Cu- Polymeric C <sub>3</sub> N <sub>4</sub>	Cu	-	Thermal decomposition	200 μA/cm <sup>2</sup> at -0.42 V vs RHE	0.2 M Na <sub>2</sub> SO <sub>4</sub> (pH 6.8)	1 h at -0.42 V vs RHE	-	-	-	1 Sun	[45]
Cu <sub>2</sub> O/g-C <sub>3</sub> N <sub>4</sub>	-	-	Electrodeposition and dip coating	2.5 mA/cm <sup>2</sup> at 0 V vs RHE	0.1M Na <sub>2</sub> SO <sub>4</sub> (pH 7)	-	-	-	-	1 Sun	[46]
NiWO <sub>4</sub> /g-C <sub>3</sub> N <sub>4</sub>	NiWO <sub>4</sub>	-	Thermal polycondensation and solvothermal	167.8 mA/cm <sup>2</sup> at 0.048 V vs RHE	0.5 M H <sub>2</sub> SO <sub>4</sub>	10 h at 0.048 V vs RHE	25.5 % at 0.048 V vs RHE	-	-	300 W Xenon lamp at a distance of 5 cm	[47]

**Note:** *ABPE*- Applied bias photon to conversion efficiency; *IPCE*- Incident photon to conversion efficiency; *STH*- Solar to hydrogen efficiency; *1 Sun* = 100 mW/cm<sup>2</sup>; *RHE*- Reversible hydrogen electrode; *SCE*- Saturated calomel electrode; *Ag/AgCl*- Silver-silver chloride reference electrode; *SHE*- Standard hydrogen electrode; *HTL*- Hole transport layer; *ALD*- Atomic layer deposition; *CBD*- Chemical bath deposition; *PBS* – Phosphate buffer solution;

## 113 2.1. Metal oxide-based photocathodes

114 Copper and its oxide compounds are commonly utilized as photocathodes in PEC cells  
115 for water splitting. Two main forms of copper oxide, cuprous oxide ( $\text{Cu}_2\text{O}$ ) and cupric oxide  
116 ( $\text{CuO}$ ), are frequently studied due to their p-type semiconductor properties.  $\text{Cu}_2\text{O}$  has  
117 particularly received extensive attention as a photocathode material, with a bandgap ranging  
118 from 2.1 to 2.6 eV [48], enabling effective absorption of a significant portion of the solar  
119 spectrum.  $\text{CuO}$ , on the other hand, possesses a direct bandgap of approximately 1.2 to 1.7 eV  
120 [49]. Both materials are prone to photo-corrosion, as the photo-generated electron-hole pairs  
121 often participate in reducing the copper oxides rather than driving the HER. To address the  
122 issue of photo-corrosion, Kalanur *et al.* [22] proposed a multijunction approach (**Figure 2 (a-**  
123 **f)**). Their study involved incorporating a layer of  $\text{Cu}_2\text{O}$  with a carefully designed hole  
124 extraction layer composed of a non-noble metal material,  $\text{NiO}_x$ . Furthermore, an aluminium-  
125 doped zinc oxide (AZO) layer was added as an electron tunnelling layer on top of the  $\text{Cu}_2\text{O}$   
126 (**Figure 2 (a, c)**). This encapsulation technique significantly inhibits photo-corrosion. To  
127 further enhance performance, a non-toxic  $\text{MoO}_x$  catalyst layer was integrated into the  
128 multijunction structure. The resulting  $\text{NiO}_x/\text{Cu}_2\text{O}/\text{AZO}/\text{MoO}_x$  photocathode demonstrated an  
129 impressive photocurrent density of  $6.1 \text{ mA cm}^{-2}$  at 0 V *vs* RHE and exhibited minimal decay  
130 in photocurrent during a 6-hour stability test (**Figure 2 (e)**) and showed maximum incident to  
131 photon conversion efficiency (IPCE) of about  $\sim 60\%$  until 500 nm (**Figure 2 (f)**). This  
132 approach represents a significant advancement in photocathode construction, as it enables non-  
133 noble metal materials to extract holes, facilitate electron tunnelling, and co-catalyse the HER,  
134 achieving performance comparable to noble metal-integrated counterparts.



135  
 136 **Figure 2.** (a) Formation of heterojunction along with the photogenerated carriers dynamics on  
 137 FTO/NiO<sub>x</sub>/Cu<sub>2</sub>O/MoO<sub>x</sub>. FESEM images of (b) NiO<sub>x</sub>/Cu<sub>2</sub>O and (c) NiO<sub>x</sub>/Cu<sub>2</sub>O/AZO/MoO<sub>x</sub> on FTO substrate. (d)  
 138 HAADF and Elemental mapping of all elements. (e) Long-term stability of all prepared electrodes. (f) IPCE plots  
 139 of NiO<sub>x</sub>/Cu<sub>2</sub>O/AZO/MoO<sub>x</sub>. Reproduced with permission [22], Copyright 2022, Elsevier.

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141 CO<sub>3</sub>O<sub>4</sub> (cobaltic oxide) is another semiconducting material that has received  
 142 significant attention in recent research due to its favourable electronic and optical properties,  
 143 as well as its long-term chemical stability. With an energy gap ranging from 1.5 to 2.6 eV [24],  
 144 CO<sub>3</sub>O<sub>4</sub> can efficiently absorb visible light, making it a promising candidate for hydrogen  
 145 production via water splitting. However, there are challenges associated with using CO<sub>3</sub>O<sub>4</sub> as  
 146 a photocathode, including its relatively high overpotentials ranging from 400 mV to 600 mV  
 147 in alkaline solutions (i.e., KOH). To overcome these limitations using sustainable non-noble  
 148 metal materials, researchers like Abdelmoneim *et al.* have explored doping techniques [26].  
 149 They successfully co-doped CO<sub>3</sub>O<sub>4</sub> with nickel (Ni) and iron (Fe), resulting in a 6% (Fe, Ni)  
 150 CO<sub>3</sub>O<sub>4</sub> electrode. This modified electrode exhibited a significant cathodic photocurrent density  
 151 of 13.6 mA cm<sup>-2</sup> at -1V vs RHE and a solar-to-hydrogen conversion efficiency of  
 152 approximately 11.3% [26]. These findings offer promising possibilities for further

153 advancements in the utilization of  $\text{CO}_3\text{O}_4$  as a photocathode material, with the potential for  
154 even better performance without the need for noble metal incorporation.

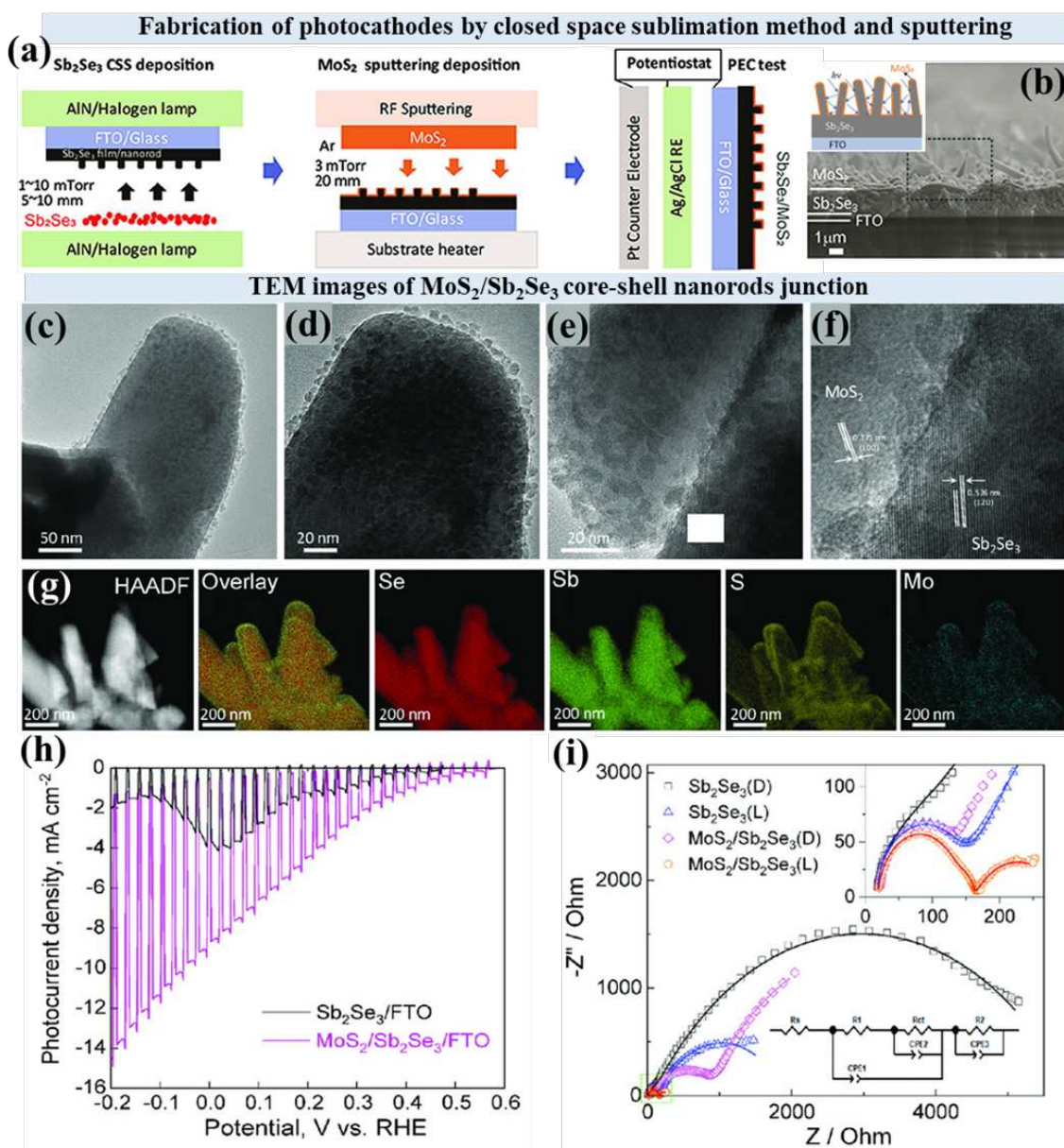
155 Another metal oxide of significant interest is NiO. It is commonly utilized as a hole-  
156 selective layer in many photocathodes due to its suitable bandgap and band alignment [50].  
157 However, its performance in hydrogen reduction during water splitting has been less  
158 impressive compared to other metal oxide photocathodes due to relatively high rates of charge  
159 recombination. To tackle this challenge, Sahoo *et al.* implemented a doping strategy by  
160 introducing monovalent metals, specifically copper, into NiO. This doping resulted in  
161 improved physical properties and yielded a modest cathodic photocurrent density of  $6.97 \text{ mA}$   
162  $\text{cm}^{-2}$  at  $0 \text{ V vs RHE}$ . The enhancement was attributed to a reduction in charge transfer resistance  
163 observed in electrochemical impedance spectroscopy (EIS) measurements, indicating  
164 enhanced separation of photogenerated charge carriers after copper doping [27]. Another study  
165 by Song *et al.* investigated the use of such copper-doped nickel oxide (Cu:NiO) layer as a hole-  
166 selective back contact material for the primary semiconductor  $\text{CuBi}_2\text{O}_4$ . The combination of  
167 the thin Cu:NiO layer with  $\text{CuBi}_2\text{O}_4$  exhibited a synergistic effect, significantly improving the  
168 cathodic photocurrent density. Specifically, they achieved a cathodic photocurrent density of  
169  $2.83 \text{ mA cm}^{-2}$  at  $0.6 \text{ V vs RHE}$  [28]. This improvement is noteworthy considering that  $\text{CuBi}_2\text{O}_4$   
170 typically demonstrates low charge carrier mobility, which hinders efficient extraction and  
171 transport of photoexcited charges [29].

## 172 **2.2. Metal chalcogenide-based photocathodes**

173 Metal chalcogenides have attracted significant attention as photocathode materials due  
174 to their distinctive electronic and optical properties. These materials consist of one or more  
175 electropositive elements combined with at least one chalcogen anion, such as Sulfur (S),  
176 Selenium (Se), or Tellurium (Te). The composition of metal chalcogenides results in unique  
177 energy bands that make them suitable for various applications. By adjusting the metal and

178 chalcogenide components, it is possible to tune the bandgaps according to specific  
179 requirements. Moreover, many metal chalcogenides have direct bandgaps, enabling efficient  
180 absorption of a wide range of wavelengths and promoting high photocurrent generation while  
181 minimizing energy losses. However, several challenges exist in utilizing metal chalcogenides  
182 for PEC HER, including instability, slow reaction kinetics, charge recombination, and  
183 performance degradation caused by surface oxidation and corrosion during harsh water-  
184 splitting processes [51]. Researchers have been working to address these limitations. For  
185 example, Guo *et al.* developed high-quality films of  $\text{Sb}_2\text{Se}_3$  with surface morphologies  
186 composed of nanorod (NR) arrays using a cost-effective and scalable close-spaced sublimation  
187 technique (**Figure 3(a, b)**). They further improved the performance by sputtering a non-  
188 precious and scalable crystalline molybdenum sulfide ( $\text{MoS}_2$ ) film as a cocatalyst and  
189 protective layer on the  $\text{Sb}_2\text{Se}_3$  NR arrays (**Figure 3 (c-g)**). The resulting core-shell structured  
190  $\text{MoS}_2/\text{Sb}_2\text{Se}_3$  NR PEC devices achieved an impressive photocurrent density of  $10 \text{ mA cm}^{-2}$  at  
191  $0 \text{ V vs RHE}$  under simulated solar light [30]. The enhanced performance (**Figure 3 (h)**) of the  
192 photocathode was attributed to the effective charge transfer facilitated by the reduced charge  
193 transfer resistance also observed in the EIS results (**Figure 3 (i)**). This was achieved through  
194 the synergistic effect of the photocathode's structure (**Figure 3 (a, b)**). The role of  $\text{MoS}_2$  as a  
195 cocatalyst should not be underestimated, as it has demonstrated remarkable performance, even  
196 surpassing that of Pt in photocatalytic applications [52]. Depending on its phase, whether in  
197 the 2H semiconducting phase or the 1T metallic phase, or a combination of both,  $\text{MoS}_2$  can be  
198 incorporated to fulfil specific requirements. It can act as a semiconductor, forming junctions  
199 with other semiconductors if desired, or function similarly to a metallic cocatalyst like Pt. This  
200 versatility was also utilized by Hu *et al.*, who hydrothermally loaded a mixture of 1T and 2H  
201 phase  $\text{MoS}_2$  onto a hydrogenated anatase/rutile hetero-phase  $\text{TiO}_2$  to create a new  $\text{MoS}_2/\text{TiO}_2$   
202 photocathode. The resulting cathodic photocurrent density of  $\text{MoS}_2/\text{TiO}_2$  reached  $5.2 \text{ mA cm}^{-2}$ ,

203 which was approximately 2.7 times higher than that of a Pt-loaded TiO<sub>2</sub> photocathode  
 204 achieving 1.9 mA cm<sup>-2</sup> in similar conditions [31].



205  
 206 **Figure 3.** (a) Fabrication process of core-shell MoS<sub>2</sub>/Sb<sub>2</sub>Se<sub>3</sub> nanorods (NR) array electrodes by closed space  
 207 sublimation (CSS) method and sputtering. (b) Cross-sectional SEM image of MoS<sub>2</sub>/Sb<sub>2</sub>Se<sub>3</sub> photocathodes. (c-f)  
 208 HRTEM images of prepared core-shell MoS<sub>2</sub>/Sb<sub>2</sub>Se<sub>3</sub> NR interface. (g) Elemental mapping of prepared  
 209 photocathodes. (h) Linear sweep voltammetry of bare Sb<sub>2</sub>Se<sub>3</sub> and 30 nm MoS<sub>2</sub>/Sb<sub>2</sub>Se<sub>3</sub> photocathodes under AM  
 210 1.5G illumination. (i) Nyquist plots of Sb<sub>2</sub>Se<sub>3</sub> and core-shell MoS<sub>2</sub>/Sb<sub>2</sub>Se<sub>3</sub> NR electrodes at 0V vs RHE under  
 211 dark and illumination conditions. Reproduced with permission [30]. Copyright 2020, John Wiley & Sons.

212  
 213 The chalcogenide compound Cu<sub>2</sub>BaSn(S,Se)<sub>4</sub>, also known as CBTSSe, has gained  
 214 significant attention as a photocathode for water splitting, similar to other chalcogenides like



215  $\text{Cu}_2\text{ZnSn}(\text{S},\text{Se})_4$  (CZTSSe) [53, 54]. These materials have attracted interest because they utilize  
216 abundant elements such as copper, zinc, and tin in their structures, making them cost-effective  
217 and potentially suitable for large-scale hydrogen production. CBTSSe photocathodes have a  
218 desirable direct band gap ranging from 1.5 to 2.0 eV, which makes them well-suited for HER,  
219 especially when compared to CZTSSe photocathodes that have inherent anti-site defects [32,  
220 55]. When combined with sustainable cocatalysts or modified with abundant elements, they  
221 have the potential to enable economically viable hydrogen production through water reduction.  
222 A study conducted by Xie *et al.* demonstrated promising results using molybdenum (Mo)-  
223 coated quartz substrates to apply a precursor solution, followed by annealing with sublimated  
224 sulfur to create Mo/CBTS photocathodes. They achieved a cathodic photocurrent of  $4 \text{ mA cm}^{-2}$   
225 at 0 V vs RHE. The excellent performance of the photocathodes was attributed to the coated  
226 Mo and the large grain sizes of the resulting structures, which reduce charge recombination as  
227 grain boundaries and defects would otherwise promote recombination. These findings offer a  
228 pathway to unlock the full potential of these photocathodes by integrating other abundant and  
229 efficient HER materials, like  $\text{MoS}_x$  to achieve the maximum theoretical photocurrents for this  
230 material [33].

231 Several other metal chalcogenides have demonstrated potential as photocathodes for  
232 water splitting without needing noble metal incorporation. These include  $\text{Cu}_2\text{S}$  [34, 35], and  
233 ternary chalcogenides such as  $\text{Cu}_3\text{BiS}_3$  [37] and  $\text{CuInS}_2$  [38]. When combined with non-noble  
234 metal materials, such as cocatalysts like CoP [56] or semiconducting protective layers like ZnO  
235 [34, 35] that serve as sites for electron mobility after junction formation, improved  
236 photocathodes can be developed.

237

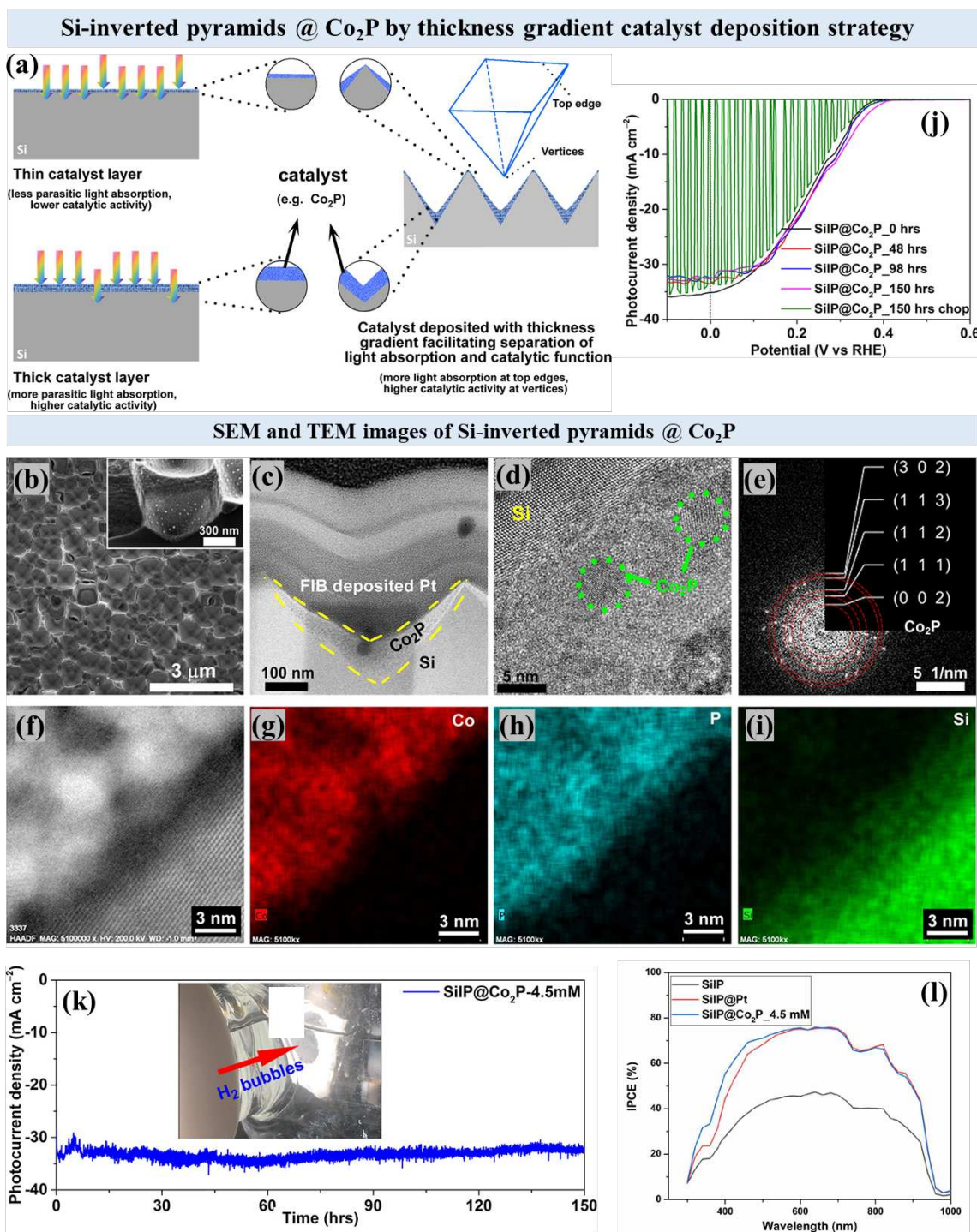
### 238 ***2.3. Metal phosphide based photocathodes***

239 Cobalt phosphide with its different phases, CoP and Co<sub>2</sub>P, serves a dual role as a  
240 cocatalyst for the HER and as a protective layer for unstable primary semiconductors, shielding  
241 them from corrosion when in direct contact with an electrolyte solution [57]. They exhibit high  
242 catalytic activity comparable to that of platinum (Pt). In a study conducted by Thalluri *et al.*,  
243 Co<sub>2</sub>P was employed to significantly enhance the photocurrent production of p-type silicon (p-  
244 Si), resulting in an impressive photocurrent density of 35.2 mA cm<sup>-2</sup> at 0 V vs RHE[39].

245 This performance rivalled that of other silicon photocathodes incorporating junctions for  
246 passivation and Pt as the HER catalyst. The enhancement was achieved through a series of  
247 steps and the advantage of thickness gradient in the inverted pyramid configuration is given in  
248 **Figure 4 (a)**. First, galvanostatic electrochemical etching was performed on the p-Si surface  
249 (**Figure 4 (b)** and inset) to create a pyramid texture. Then, Co<sub>2</sub>P was gradient deposited on the  
250 textured surface (**Figure 4 (c, d)**), effectively reducing parasitic energy losses by separating  
251 the catalytic sites from the light absorption sites through the controlled distribution of Co<sub>2</sub>P  
252 loading. This configuration exhibited remarkable stability, maintaining its photocurrent density  
253 even with prolonged operation (**Figure 4 (k)**). When Pt was used as a replacement for the  
254 optimized Co<sub>2</sub>P loading under the same conditions, its performance was found to be  
255 significantly inferior (**Figure 4 (l)**) [38].

256

257



258

259 **Figure 4.** (a) Influence of catalyst layer thickness on light absorption and advantages of thickness gradient  
 260 catalytic layer in the inverted pyramid configuration. (b) SEM image of Si-inverted pyramid @  $\text{Co}_2\text{P}$  4.5mM  
 261 photocathodes. (c) Cross-sectional TEM image showing  $\text{Co}_2\text{P}$  in Si-IP. (d) HRTEM images showing the  
 262 crystalline structure of  $\text{Co}_2\text{P}$  (e) FFT-ED pattern of  $\text{Co}_2\text{P}$  layer on Si-IP configuration. (f-i) HAADF and elemental  
 263 maps of Co, P, Si. (j) LSV taken before and after long-term stability test with varied time intervals. (k) Long-term  
 264 stability (I-T curve) of prepared  $\text{Co}_2\text{P}$  @Si-IP photocathodes measured at 0 V vs RHE under AM 1.5G  
 265 illumination. (l) IPCE curve of all prepared photocathodes measured at -1 V in 0.5M  $\text{H}_2\text{SO}_4$  electrolyte.  
 266 Reproduced with permission [39]. Copyright 2019, American Chemical Society.

267

268 Other metal phosphides, such as GaP [40] and GaInP<sub>2</sub> [58], have demonstrated potential  
269 in generating high photocurrents when combined with non-noble metal materials and modified  
270 accordingly. In the case of GaInP<sub>2</sub>, when coupled with thin films of MoS<sub>2</sub> that serve as both a  
271 catalyst and a protective layer, it achieves performance comparable to that of a platinum.

#### 272 **2.4. Silicon-based photocathodes**

273 Silicon-based photocathodes have consistently attracted significant attention for their  
274 potential in water reduction to produce environmentally friendly hydrogen. Silicon (Si) is a  
275 cost-effective and readily available semiconductor material with desirable electronic  
276 properties. Among silicon-based photocathodes, p-type silicon has been the focus of most  
277 research, despite its inherent sluggish kinetics for the HER [41]. Recent studies for this material  
278 have aimed to develop photocathode designs using abundant materials that can rival or surpass  
279 the performance of noble metals. One such study conducted by Lin *et al.* employed a facile  
280 assembly process to create an amorphous thin film of MoS<sub>x</sub> on p-Si, resulting in a remarkable  
281 cathodic photocurrent density of 28.2 mA cm<sup>-2</sup> at 0 V vs RHE, accompanied by a high Faradaic  
282 efficiency of approximately 98% [42]. Similar trends are observed in other studies that utilize  
283 non-noble metal-based materials to form efficient heterojunctions with p-Si. For instance, Ku  
284 *et al.* achieved a powerful Schottky junction by integrating 3D textured graphene onto p-Si.  
285 This integration enhanced electrochemical activity, stability, and charge separation efficiency,  
286 leading to an impressive cathodic photocurrent density of 32.5 mA cm<sup>-2</sup> at 0 V vs RHE [43].  
287 These results indicate the tremendous potential of silicon-based photocathodes for water  
288 splitting and hydrogen production. The theoretical limit of their photocurrent has not yet been  
289 reached, definitely leaving room for further improvement.

#### 290 **2.5. Metal-free photocathodes**

291 Metal-free photocathodes composed of carbon-based semiconductor materials, such as  
292 graphene oxide [44] and graphitic carbon nitride (g-C<sub>3</sub>N<sub>4</sub>) [45, 59], have been investigated as

293 highly active and cost-effective alternatives. These photocathodes are synthesized using  
294 various methods, including thermal treatment, hydrothermal synthesis, and solvothermal  
295 synthesis. Similar to other photocathode designs that utilize p-n junctions by integrating p-type  
296 and n-type semiconductors for charge separation and as passivation layers for stability, these  
297 carbon-based materials also play a similar role due to their favourable electrical and optical  
298 properties. In a study by Ma *et al.*, three-dimensional Cu<sub>2</sub>O foams, which outperform Cu<sub>2</sub>O  
299 films, are coated with g-C<sub>3</sub>N<sub>4</sub> to enhance stability and photocurrent generation. The resulting  
300 composite photocathode achieves a cathodic photocurrent density of 2.5 mA cm<sup>-2</sup> at 0 V vs  
301 RHE [46], representing a modest improvement brought by better electron mediation within the  
302 photocathode.

### 303 **3. Conclusion and outlook**

304 A diverse range of non-noble metal-based materials has emerged as promising  
305 photocathodes for water splitting, exhibiting remarkable performance in HER. Metal  
306 phosphides, such as CoP, have demonstrated exceptional catalytic activity for HER while  
307 simultaneously serving as passivation layers, shielding vulnerable primary semiconductors  
308 from corrosion. These materials have achieved competitive photocurrent densities, rivalling or  
309 even surpassing those of Pt-based photocathodes. Carbon-based semiconductors, including  
310 graphene oxide and graphitic carbon nitride, hold immense potential due to their exceptional  
311 stability, however, further research is warranted to fully harness their capabilities. Similarly,  
312 significant advancements have been made in the utilization of metal chalcogenides. When  
313 combined with non-noble metal materials or modified with abundant elements, these materials  
314 have exhibited potential for efficient and cost-effective hydrogen production. Silicon-based  
315 photocathodes stand out as particularly promising material technologies for clean energy  
316 production through PEC water splitting due to their exceptional efficiency in converting  
317 sunlight illumination into electric current, resistance to corrosion and degradation over

318 extended operation periods, and relatively low production costs. Non-noble plasmonic metal-  
319 based photocatalysts, such as copper (Cu), aluminium (Al), and nickel (Ni), exhibit plasmon  
320 resonance effects, a phenomenon where collective oscillations of electrons are induced by  
321 incident light, leading to enhanced light absorption and scattering, ultimately resulting in  
322 localized electromagnetic field enhancement [60]. Researchers have made significant strides  
323 too in the development and optimization of these materials through meticulous synthesis  
324 methods tailored to improve surface modifications, ultimately enabling hybridization with non-  
325 noble metal semiconductors for HER water splitting.

326         The outlook for non-noble metal-based photocathodes for water splitting is decidedly  
327 optimistic. The rapid advancements in non-noble metal-based photocathodes, some with  
328 performances already surpassing those of their noble metal counterparts, mark a pivotal  
329 moment in the evolution of PEC water splitting technology. These promising alternatives have  
330 the potential to propel PEC hydrogen production towards greater scalability and widespread  
331 adoption. Ongoing research endeavours continue to aim for optimization of the performance,  
332 stability, and scalability of these photocathodes through various discussed strategies, including  
333 surface modification, heterojunction formation, and nano-structuring. By leveraging these  
334 abundant and sustainable materials, non-noble metal-based photocathodes will play a pivotal  
335 role in realizing a clean and renewable energy future.

### 336 **CRedit authorship contribution statement**

337 **Antony Charles Minja:** Conceptualization, Writing – original draft, Writing- review &  
338 editing, **Karthick Raj AG:** Conceptualization, Writing – original draft, Writing – review &  
339 editing, **Arno Raes:** Writing review & editing, **Rituraj Borah:** Writing- review & editing,  
340 **Sammy W. Verbruggen:** Conceptualization, Funding acquisition, Supervision, Writing-  
341 review & editing.

342 **Declaration of competing interest**

343 The authors declare no competing financial interests.

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471 This approach effectively showcased a stable and efficient photocathode based on  
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