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1 **Hierarchical zeolite single crystal reactor for excellent catalytic**
2 **efficiency**

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1 **SUMMARY**

2 As a size and shape selective catalyst, zeolites are widely used in petroleum and fine
3 chemicals processing. However, their small micropores severely hinder molecular
4 diffusion and are sensitive to coke formation. Hierarchically porous zeolite single
5 crystals with fully interconnected, ordered and tunable multimodal porosity at macro-,
6 meso- and micro-length scale like in leaves offer the ideal solution. However, their
7 synthesis remains highly challenging. Here we report a versatile confined zeolite
8 crystallization process to achieve these superior properties. Such zeolite single
9 crystals lead to significantly improved mass transport properties by shortening the
10 diffusion length while maintaining shape selective properties, endowing them with a
11 high efficiency of zeolite crystals, enhanced catalytic activities and life time, highly
12 reduced coke formation and reduced deactivation rate in bulky-molecule reactions and
13 methanol-to-olefins process. Their industrial utilization can lead to the design of
14 innovative and intensified reactors and processes with highly enhanced efficiency and
15 minimum energy consumption.

16

17 **KEYWORDS:** hierarchical zeolites, zeolite single crystals, ordered structure,
18 interconnected pores, improved diffusion rate, highly reduced coking, catalytic
19 efficiency, catalytic cracking, MTO process

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1 INTRODUCTION

2 Zeolites are one of the most important catalytic materials used today. Each zeolite
3 crystal can be considered as a catalytic reactor.¹⁻⁴ However, the slow transport of the
4 reactants and products within their intra-crystalline small microporous channels
5 greatly hinders their efficiency in catalysis and separation. It is highly desirable to
6 reduce the diffusion limitations within each zeolite crystal.⁵ The introduction of
7 additional interconnected intra-crystalline hierarchical mesopores or/and macropores
8 is thus an efficient strategy towards this.⁶⁻¹¹ In comparison with conventional
9 demetalation methods which are based on random removal of Al or Si atoms from
10 zeolite frameworks through acid or alkali leaching¹²⁻¹⁴ or chemical treatment,¹⁵
11 templating methods¹⁶⁻²⁸ allow precise control of the density and the size of additional
12 mesopores or macropores, generate much less defects in the zeolite framework and
13 can avoid any negative impact on crystallinity and active sites of the zeolite crystals.
14 Tsapatsis et al. realized the synthesis of size-tunable zeolite nanocrystals (MFI,¹⁷⁻¹⁹
15 BEA²⁰ and FAU²⁰) with various intercrystalline mesoporosities using mesoporous
16 carbon templates by steam-assisted crystallization¹⁷⁻¹⁹ or hydrothermal synthesis²⁰ for
17 improved catalytic performance in ethanol dehydration.¹⁹ However, the synthesis of
18 zeolite single crystals with intracrystalline hierarchically porous structure still remain
19 a great challenge. Schwieger et al. very nicely synthesized microporous MFI-type
20 (Silicalite-1,²¹ ZSM-5,^{22,23} TS-1²⁴) zeolite single crystals by steam-assisted
21 crystallization of mesoporous silica spheres. The embedded intracrystalline
22 macropores in ZSM-5 zeolite and TS-1 zeolite can significantly improve the catalyst
23 performance in the methanol to olefins (MTO) reaction^{22,23} and the liquid phase
24 epoxidation of 2-octene.²⁴

25 The interconnectivity between the intrinsic microporosity of zeolites and
26 additional meso- or/and macroporosity is essential to improve their mass transport
27 properties.⁵ This leads to a better accessibility to the active sites located within the
28 micropore system of the zeolites.²⁹ Many advanced techniques are developed to probe
29 the interconnectivity of the pores.³⁰ Pérez-Ramírez et al. contributed greatly to get
30 insight into the interconnectivity of the hierarchical pores,³¹ including the gas
31 adsorption studies,³² mercury porosimetry,³³ gravimetric uptake of bulking alkanes,³⁴
32 infrared spectroscopy of substituted alkyipyridines with different size³⁵ and positron
33 annihilation lifetime spectroscopy.^{36,37} In the same way, advanced imaging techniques
34 based on electron tomography and rotation electron diffraction,^{21,28} continuous-flow
35 variable-temperature hyperpolarized ¹²⁹Xe nuclear magnetic resonance(NMR)^{38,39} and
36 pulsed field gradient (PFG) NMR^{40,41} have also proven to be powerful tools for
37 assessing the pore interconnectivity in hierarchically structured zeolites.

38 More precise control over the location and interconnectivity of the additional
39 porosity could offer potential gains from the structural modification.⁴² To address this,
40 a cationic amphiphilic copolymer was designed by Liu et al.²⁷ to synthesize ZSM-5
41 zeolite single crystalline sheets with abundant and highly interconnected
42 b-axis-aligned mesoporous channels. This approach resulted in much higher catalytic
43 activities for bulky molecules conversion than bulk ZSM-5 and ZSM-5 with randomly

1 oriented mesopores while the stability and shape selectivity need further improved.²⁷
2 Many classes of organisms both in plants and animals contain hierarchical networks
3 of interconnected pores to maximize mass transport and rates of reactions for the
4 highest efficiency and the lowest energy consumption. It is thus highly desirable to
5 construct such hierarchical networks of interconnected pores within each of zeolite
6 crystals. However, hierarchically structuring tunable mesoporosity and macroporosity
7 simultaneously with excellent interconnectivity, desired location and high ordering
8 within each zeolite single crystal to maximize the benefits of the porous hierarchy in
9 catalytic reactions still remains highly challenging.^{10,42}

10 We report synthesis of ZSM-5 zeolite single crystals with a fully interconnected
11 and highly ordered intra-crystalline macro-meso-microporous hierarchy in all
12 directions like in leaves through a versatile templating strategy via an in-situ
13 bottom-up confined zeolite crystallization process (Figure 1). The unique hierarchical
14 structure of our single zeolite crystals with high thermal and hydrothermal stability
15 and excellent mechanic strength can maximize the intracrystalline diffusion rate of
16 reactants and products and the accessibility to active sites, for example, the effective
17 diffusion rate of molecules in our zeolite single crystals reaches 10 and 5 folds higher
18 than that of benchmark commercial microsized mesoporous ZSM-5 and the
19 state-of-the-art nanosized macroporous ZSM-5, respectively. The efficiency of each
20 of our zeolite single crystals in bulky 1,3,5-triisopropylbenzene (TIPB) cracking
21 reaction can reach a high value of 0.84, which is doubled compared to that of
22 benchmark commercial ZSM-5 catalyst and 20% higher than that of the
23 state-of-the-art nanosized ZSM-5 catalyst. All these excellent properties lead to a
24 significant reduction in coking and deactivation rate, outstanding catalytic activity and
25 selectivity and highly improved catalyst lifetime in bulky molecules reactions and
26 methanol-to-olefins (MTO) process. The average coking rate in MTO reaction using
27 our zeolite single crystals can reach 8.9 and 7.2 times lower and their life time can be
28 extended 13 and 3.5 times longer than that for benchmark commercial microsized
29 mesoporous ZSM-5 and the state-of-the-art nanosized macroporous ZSM-5,
30 respectively. It is envisioned that the catalytic reactor can be made solely by zeolite
31 single crystal. The excellent performance of zeolite single crystals can be fully
32 exploited in the catalytic reactions without any negative effect of binder and additives
33 with much highly enhanced efficiency, high catalytic activity, selectivity and life time,
34 minimum energy, time and raw material consumption. Most importantly, the
35 up-scalable and versatile synthesis strategy has been used to synthesize single crystals
36 of any kind of zeolite structures and zeotype materials and can be extended to
37 fabricate single crystalline solids of any chemical compositions such as transition
38 metal oxides (TiO₂, ZnO, ...), phosphate compounds, silicate compounds,
39 perovskites,... These new hierarchically meso-macroporous materials with single
40 crystalline nature will open a large panel of applications not only in adsorption,
41 catalysis and separation, but also in energy storage and conversion and create new
42 research areas.

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4 RESULTS AND DISCUSSION

5 Hierarchical Zeolite Single Crystals.

6 Hierarchically ordered macro-mesoporous carbon (OMMC, Figures 1B and S1) with
7 an inverse opal structure is used as the meso-macroporous template.^{43,44} The key
8 advantage is that such a template provides a confined and tailorable space for the
9 formation of well-shaped hierarchically porous zeolite single crystals (Figures 1C and
10 1D). The synthesis of such uniform zeolite single crystals involves three steps: 1)
11 infiltration of zeolite precursor slurry into OMMC template (Step 1 in Figure 1), 2)
12 steam-assisted crystallization (SAC) of precursors confined in OMMC template (Step
13 2 in Figure 1), and 3) template removal (Step 3 in Figure 1). Each zeolite single
14 crystal is made up of an assembly of tightly interconnected small zeolite spheres in a
15 close-packed FCC arrangement (Figure 1E), which inherits the highly ordered FCC
16 structure of the original polystyrene opalline crystal structure for the OMMC template
17 fabrication (Figure S2). The FCC arrangement of these tightly interconnected small
18 zeolite spheres results in the formation of tetrahedral (T_d) and octahedral (O_h) voids
19 by 4 and 6 spheres from two layers (Figures 1F and S3-S5, Video S1), respectively.
20 Such periodically arranged T_d and O_h voids within one zeolite single crystal, in
21 addition to the intrinsic microporosity of zeolite, provide ordered, fully interconnected
22 meso- and macroporous structure (Figure 1G and Video S1), precisely replicating the
23 OMMC template (Figure S1).

24 As proof of concept, hierarchically ordered and highly interconnected
25 macro-meso-microporous ZSM-5 single crystals with a tunable Si/Al molar ratio from
26 30 to ∞ and tunable mesopore and macropore size have been synthesized. We denote
27 the as-synthesized samples as OMMS-ZSM-5(x,y); OMMS represents ordered
28 macro-mesoporous zeolite single crystals, whereas x and y indicate the macropore size
29 (nm) of the OMMC templates and the obtained Si/Al molar ratio, respectively.

30 Scanning electron microscope (SEM) images show that after 10 h crystallization,
31 all the crystals of OMMS-ZSM-5(400,30) used as a representative sample for
32 illustration exhibit a truncated hexagonal prism morphology (Figures 2A-2D).
33 Although the size of the OMMC template is tens of microns, the obtained zeolite
34 single crystals are well shaped with a similar crystal size of 4-5 μm (Figure 2A). The
35 final size and the uniformity of the zeolite single crystals are highly dependent on the
36 mass ratio of SiO_2 and OMMC template ($m\text{SiO}_2/m\text{OMMC}$) and the crystallization
37 time (Figures S6-S7). When $m\text{SiO}_2/m\text{OMMC}$ mass ratio is low, the macropores of the
38 OMMC template can only partly be filled by the dry gel, only several zeolite single
39 crystals with different crystal size are obtained (Figure S6) while at high
40 $m\text{SiO}_2/m\text{OMMC}$ ratio, all the porosities of the OMMC template can be fully filled by
41 the dry gel, the uniform zeolite single crystals are formed. With increasing the
42 crystallization time to 10h, zeolite single crystals reach their maximum size of 4-5 μm
43 (Figure S7). The highly ordered arrangement of uniformly sized and tightly

1 interconnected small zeolite spheres can be clearly visualized from three different
2 directions in the representative SEM images (Figures 2B-2D) and the corresponding
3 schematic illustrations (Figures 2E-2G) of one individual zeolite crystal. Typical
4 high-angle annular dark field scanning transmission electron microscope
5 (HAADF-STEM) images reveal that tightly packed and inter-grown spheres are
6 imprinted in a highly ordered arrangement within one entire zeolite crystal of
7 OMMS-ZSM-5(400,30), indicating a positive replication of the OMMC template. The
8 diameter of the small zeolite spheres is ~400 nm (Figure 2H), identical to the
9 macropore size of the OMMC template and slightly smaller than the size of the PS
10 spheres (~420 nm) used for the OMMC template fabrication (Figure S2). The
11 corresponding selected area electron diffraction (SAED) pattern of one zeolite crystal
12 (Figure 2H inset) contains discrete diffraction spots and can be indexed according to
13 the MFI type zeolite along the [010] zone axis, providing the clear evidence of the
14 single-crystalline nature of OMMS-ZSM-5(400,30). By indexing the diffraction
15 pattern, the crystal facets can be readily identified. The largest facets, oriented
16 perpendicular to the electron beam, are (010)-type, while the smaller side facets are
17 (101) and (001)-type. The dominance of the (010) surface facets can be explained by
18 the MFI-type zeolite crystal preferential growth direction, being in accordance with
19 theoretical calculations predicting the lowest surface energy for the (010) facet.²⁷ A
20 typical joint between two spheres can be seen in Figure 2I. SAED patterns taken from
21 three different regions of the joint (Figure 2I zone 1 to zone 3) show that the two
22 spheres, including the half sphere on the left (Figure 2J), the joint area (Figure 2K)
23 and the sphere on the right (Figure 2L), all have the same [001] orientation. This
24 unequivocally confirms the continuous crystalline phase in the same axis. The high
25 resolution transmission electron microscopy (HRTEM) image (Figure 2M and inset)
26 of the area indicated by the green rectangle in Figure 2I further reveals a continuous,
27 uniform and dislocation free lattice arrangement at the joint over the entire image
28 region. No (sub)grain boundaries or interfaces are observed. This proves that all the
29 spheres organized in an FCC arrangement are not independent but crystallized in the
30 same orientation and intergrown to form the entire zeolite single crystal. Besides, no
31 planar defects such as stacking faults, twin formation or dislocations are observed,
32 confirming the pristine structure throughout the zeolite crystal.

33 X-ray diffraction (XRD) pattern of OMMS-ZSM-5(400,30) is compared with
34 standard PDF card (Figure 2N), confirming the formation of a pure MFI zeolite phase
35 with excellent crystallinity. The intrinsic microporosity of zeolite single crystals is
36 determined using Ar adsorption while N₂ adsorption is adopted to investigate the
37 mesoporosity. OMMS-ZSM-5(400,30) shows type I isotherms with a micropore size
38 distribution centered at 5.3 Å, characteristic of ZSM-5 zeolite with 10-membered ring
39 channels (Figure 2O and inset).⁴⁵ The micropore surface area and volume of
40 OMMS-ZSM-5(400,30) are 299 (m²g⁻¹) and 0.14 (cm³g⁻¹), respectively (Table S1),
41 indicating that the formation of a hierarchically macro-mesoporous system does not
42 influence its inherent microporous structure. The existence of mesopores in
43 OMMS-ZSM-5(400,30) is confirmed by N₂ adsorption-desorption experiments. The
44 steep adsorption of N₂ in the 0.6 < p/p₀ < 1.0 region corresponds to the capillary

1 condensation in the mesopores (Figure S8A), with the mesopore size distribution
2 centered at 35 nm. The mesoporous surface area and total volume are 166 (m^2g^{-1}) and
3 $0.27 (\text{m}^3\text{g}^{-1})$ (Table S1), respectively. The presence of meso- and macropores in
4 OMMS-ZSM-5(400,30) is fully confirmed by means of mercury intrusion
5 porosimetry measurement showing (inset of Figure 2P) a narrow mesopores
6 distribution centered at ~ 35 nm, same as the value obtained by the N_2 adsorption, and
7 a broader macroporous distribution centered at ~ 100 nm. These interconnected and
8 periodically ordered mesopores and macropores stem from the tetrahedral and
9 octahedral voids within OMMS-ZSM-5 single crystals, respectively (Figure 1G). The
10 presence of the highly ordered mesopores of 35 nm and macropores of 100 nm is also
11 evidenced by the theoretical calculation and TEM and SEM observations (Figures S4
12 and S5), showing the consistence between the results obtained from N_2 adsorption,
13 mercury intrusion, theoretical calculations and TEM and SEM observations. The ^{27}Al
14 MAS NMR spectrum (Figure S9A) indicates that aluminum atoms solely exist in the
15 tetrahedral position ($\delta = 55$ ppm) and no extra-framework aluminum species ($\delta = 0$
16 ppm) are found. The ^{29}Si MAS NMR spectrum (Figure S9D) shows that the
17 framework primarily consists of crosslinked Q^4 silica units [$\delta = -114$ ppm, $\text{Si}(\text{OSi})_4$]
18 and Q^3 silica units [$\delta = -106$ ppm, $\text{Si}(\text{OSi})_3(\text{OH})$ and/or $\text{Si}(\text{OSi})_3(\text{OAl})$] without Q^2
19 units, indicative of complete condensation of the framework.

20 HAADF-STEM (Figure 3A) and 3D electron tomographic reconstruction (Figure
21 3B and Video S2) of OMMS-ZSM-5(400,30) demonstrate again that each zeolite
22 single crystal is constructed by densely stacked and tightly interconnected small
23 zeolite spheres of ~ 400 nm diameter. For a better visualization of the arrangement of
24 the intergrown small zeolite spheres, slices along three different axes are shown in
25 Video S2. For the slices along the Y-axis, two neighboring layers in
26 OMMS-ZSM-5(400,30) (Figures 3E-3F) correspond to the (010) planes in an FCC
27 arrangement (Fig. 3c). Similarly, for the slices along the Z-axis, two neighboring
28 layers in OMMS-ZSM-5(400,30) (Figures 3G-3H) correspond to the (-101) planes in
29 an FCC arrangement (Figure 3D). These results confirm that small zeolite spheres are
30 densely stacked in a close-packed FCC arrangement within an individual zeolite
31 single crystal, consistent with the inverse opal structure of the carbon template (Figure
32 S1). It is important to note that these small uniform zeolite spheres are all tightly
33 interconnected by the joints (Figures 3C-3H and Video S2), which are formed due to
34 the simultaneous formation of small zeolite spheres through the windows connecting
35 macropores of the OMMC template (Figure S1). All the above results clearly
36 demonstrate that highly crystalline zeolite single crystals with a hierarchically porous
37 network with fully interconnected, ordered and multimodal porosity at macro-, meso-
38 and micro-length scale like in leaves are obtained.

39 The diameter of the spheres in OMMS-ZSM-5 single crystals can be tuned from
40 200 to 600 nm by using OMMCs with different macropore sizes (Figure S1), resulting
41 in tunable tetrahedral and octahedral voids and thus, tunable intra-crystalline
42 mesopore and macropore sizes in each zeolite single crystal (Figures S10-S12 and
43 Table S1). The Si/Al ratio of OMMS-ZSM-5 can also easily be tuned from 30 to ∞
44 (Figures S13-S15). The excellent mechanical, thermal and hydrothermal stability

1 (Figure S16) of our OMMS-ZSM-5 single crystals have been evidenced through a
2 series of high pressure and high temperature treatment with/without vapor. The poor
3 (hydro)thermal stability of the Nano-ZSM-5 (Figure S17) compared to those of
4 zeolite single crystals with ordered intra-crystalline macro-meso-microporous
5 hierarchy further demonstrates the superiority of zeolite single crystals. Such high
6 stability of zeolites under very severe conditions is of great importance for their
7 catalytic applications.

9 **Catalytic Performance in Bulky Molecules Cracking and MTO.**

10 The significant superiority of such unique hierarchical porous structure of our zeolite
11 single crystals is first revealed in the catalytic cracking of bulky
12 1,3,5-triisopropylbenzene (1,3,5-TIPB) (Table S2). The benchmark commercial
13 microsized mesoporous ZSM-5 (C-ZSM-5) and the state-of-the-art nanosized
14 macroporous ZSM-5 (Nano-ZSM-5) have been used as reference catalysts. The
15 detailed adsorption-desorption and SEM characterization (Figures S18-S19 and Table
16 S1) shows clearly that the benchmark C-ZSM-5 has mesopores at 2 nm and the
17 state-of-the-art Nano-ZSM-5 has macropores at 51 nm. These two samples containing
18 hierarchically multiple porosity (micro-mesopores for the benchmark C-ZSM-5 and
19 micro-macropores for the state-of-the-art Nano-ZSM-5) are thus excellent references
20 for comparison with our micro-meso-macroporous zeolite single crystals. The
21 benchmark C-ZSM-5 and the state-of-the-art Nano-ZSM-5 give very low (10.65%)
22 and medium (48.40%) conversion, respectively. In contrast, OMMS-ZSM-5(400,30),
23 used as a representative sample, is the most active, with a conversion of 55.06%.
24 Considering their similar Si/Al ratio (20-30) and acidities (Table S3), the improved
25 catalytic activities in the conversion of bulky molecules over OMMS-ZSM-5(400,30)
26 are directly related to the leaf-like open and interconnected intracrystalline ordered
27 hierarchically micro-meso-macroporous architecture.

28 The performance boost is further evidenced in the MTO process. It is well
29 known that the delayed transport of the reactants and products in zeolite channels
30 usually accelerates the accumulation of coke, which in turn blocks the micropores and
31 negatively affects not only the lifetime, the zeolite utilization efficiency, but also the
32 selectivity of the zeolite catalyst.⁴⁶⁻⁴⁹ Thus, microporous zeolites are very sensitive to
33 deactivation due to coke formation. For a better comparison, all the catalytic tests are
34 made under harsh conditions with a high weight hourly space velocity (WHSV) of 3.6
35 h⁻¹ (See Experimental procedures). It is clearly seen (Figure 4A, Table 1) that
36 C-ZSM-5 suffers from a rapid deactivation after only ~2.4 h on stream, much earlier
37 than Nano-ZSM-5 (9.0 h). The life time extension of OMMS-ZSM-5(400,30) is 5 and
38 1.4 times longer than C-ZSM-5 and Nano-ZSM-5, respectively (Figure 4A and Table
39 1). The selectivity of ethylene and propylene (Figures 4A, S20-S21) is found to be
40 ~45% over C-ZSM-5 and from 45% to 40% over Nano-ZSM-5 with time on stream.
41 The highest selectivity (~50%) to ethylene and propylene is observed for
42 OMMS-ZSM-5(400,30). More importantly, such high selectivity to light olefins
43 remains almost constant during the whole catalytic testing. The average coking rate is
44 only 3.92 mg g⁻¹ h⁻¹ for OMMS-ZSM-5(400,30), much lower than 8.88 mg g⁻¹ h⁻¹ and

1 7.15 mg g⁻¹ h⁻¹ for C-ZSM-5 and Nano-ZSM-5, respectively (Table 1). The above
2 results show that not only our OMMS- ZSM-5(400,30) sample gives the highest
3 selectivity to ethylene and propylene in MTO reaction, but also its average
4 deactivation rate is 3.2 and 2.3 times lower than that of C-ZSM-5 and Nano-ZSM-5
5 catalysts, respectively.

6
7 More importantly, the catalytic performances of OMMS-ZSM-5 in MTO process
8 can be significantly improved by varying the Si/Al ratio and, in particular,
9 intra-crystalline mesopore and macropore size. A clear correlation between the Si/Al
10 ratio and the deactivation behaviour is evidenced in Figure 4B and Table 1. A higher
11 Si/Al molar ratio in the zeolite catalyst results in a lower deactivation rate and a
12 reduced coke formation rate of the catalysts, leading to a longer life time in the MTO
13 reaction (Figure 4B and Table 1). The correlation between the Si/Al ratio and catalyst
14 lifetime appears to be linear in the tested OMMS-ZSM-5s with the sphere size of 400
15 nm. By increasing Si/Al ratio from 30 to 100, the life time of OMMS-ZSM-5(400,100)
16 (Si/Al = 100) (20.4 h) is nearly two folds longer than that of OMMS-ZSM-5(400,30)
17 (Si/Al = 30). The Si/Al molar ratio of our ZSM-5 single crystals has only a minor
18 effect on their final catalytic selectivity of ethylene and propylene, with all values
19 staying between 45% and 50%. The interconnection and the size of their
20 intra-crystalline micro-meso-macropores are therefore crucial factors in determining
21 their catalytic lifetime (Figure 4B and Table 1). Among all the samples,
22 OMMS-ZSM-5(600,100) with the largest mesopore and macropore sizes exhibits the
23 lowest coke formation rate (1.01 mg g⁻¹h⁻¹), being 8.9 and 7.2 times lower than that of
24 C-ZSM-5 and Nano-ZSM-5 catalysts and the longest life time (31 h), being 13 and
25 3.5 times longer than that of C-ZSM-5 and Nano-ZSM-5 catalysts, respectively. As
26 the benchmark C-ZSM-5 and the state-of-the-art Nano-ZSM-5 contain only
27 micro-mesoporosity and micro-macroporosity, respectively, we attribute the above
28 excellent catalytic performance to the improved diffusion behavior of the embedded
29 larger intracrystalline pore size (Table S1) and to the completely open hierarchically
30 interconnected and ordered micro-meso-macroporous structure of the ZSM-5 single
31 crystals. As a result, the products can easily escape from the microporous channels
32 and side reactions are significantly avoided, leading to a high propylene and ethylene
33 selectivity, a much lower coke formation and deactivation rate and a substantial
34 improvement in catalytic lifetime.^{50,51} A linear correlation between deactivation rate
35 and average coke formation rate in MTO reaction is established among all tested
36 ZSM-5 catalysts with different physico-chemical characteristics, irrespective of their
37 microsize, nanosize, or interconnected hierarchical structure, or even different Si/Al
38 molar ratio (Figure 4B). This further confirms that the resistance of zeolites to
39 deactivation is mostly correlated with the suppression of coke formation.
40 Consequently, accelerating the transport of reactants and products is of crucial
41 importance in designing long-lived catalysts.

42 43 **Diffusion Behavior and Catalytic Efficiency.**

1 Generally, it is believed that both introducing an additional macro/mesopore system
 2 within an individual zeolite crystal and decreasing the crystal size can improve the
 3 accessibility of reactant molecules to the active sites of the zeolites. However, our
 4 findings demonstrate that an open, interconnected and intra-crystalline hierarchically
 5 ordered micro-meso-macroporous structure like in leaves is much more favorable. To
 6 illustrate the differences between our zeolite single crystals with interconnected
 7 micro-meso-macroporosity and the benchmark C-ZSM-5 with micro-mesoporosity
 8 and Nanosized ZSM-5 with micro-macroporosity in diffusion and catalytic efficiency,
 9 a kinetic study over 1,3,5-TIPB cracking reaction is performed. By Thiele modulus
 10 analysis method (Equations 1-3 in Experimental procedures and Table S2), the
 11 utilization efficiency and effective diffusion length/rate in C-ZSM-5, Nano-ZSM-5
 12 and OMMS-ZSM-5(400,30) are calculated. The variation of the effectiveness factor η
 13 related to the observed reaction rate as a function of the Thiele modulus ϕ ⁴⁸ is
 14 presented in Figure 4C and the η and ϕ values of studied zeolites are presented in
 15 Table S2. ϕ values drawn from C-ZSM-5, Nano-ZSM-5 and OMMS-ZSM-5(400,30)
 16 are significantly different (Table S2). Assuming that the diffusivity in all MFI
 17 microporous networks is the same,^{34,52} the obtained characteristic diffusion length L
 18 of Nano-ZSM-5 and OMMS-ZSM-5(400,30) are found to be shortened to 50% and
 19 30%, respectively, compared to that of C-ZSM-5. The effective diffusion rate (D_{eff}/L^2
 20 derived from $\frac{L_1}{L_2} = \frac{\phi_1}{\phi_2} \sqrt{\frac{k_2 D_{\text{eff},1}}{k_1 D_{\text{eff},2}}}$) of Nano-ZSM-5 is enhanced by two folds compared
 21 to C-ZSM-5 while that of OMMS-ZSM-5 (400,30) reaches 10 and 5 folds higher than
 22 that of C-ZSM-5 and Nano-ZSM-5, respectively, due to the effectively shortened
 23 diffusion length. The utilization efficiency η of C-ZSM-5 gives a generally observed
 24 value of 0.4 due to severe internal restricted diffusion because of large size of
 25 1,3,5-TIPB molecule. That of Nano-ZSM-5 increases to 0.68 due to the large
 26 reduction in crystal size to nanometer scale, while OMMS-ZSM-5(400,30) reaches
 27 the highest and a high value η of 0.84. From Figure 4C, the OMMS-ZSM-5(400,30) is
 28 located in the full use regime while C-ZSM-5 in the diffusion limited regime, showing
 29 poor utilization efficiency of C-ZSM-5 catalyst in 1,3,5-TIPB cracking reaction and
 30 explaining its low catalytic activity and short life time. This is in excellent agreement
 31 with the MTO reaction results. By varying mesopore and macropore size within each
 32 of our ZSM-5 single crystals, the effective diffusion rate and catalyst utilization
 33 efficiency and performance can be further improved (Table 1 and Figure 4A). The
 34 continuous-flow variable-temperature hyperpolarized ¹²⁹Xe NMR (Figure S22) and
 35 the pulsed field gradient nuclear magnetic resonance (PFG NMR) techniques (Figure
 36 S23) comprehensively confirm the high quality, the excellent accessibility of the
 37 hierarchical pore network structures and interconnectivity in OMMS-ZSM-5.

38 To further compare the differences between the diffusion behavior and the
 39 accessibility to acid sites at dynamic condition of hierarchically porous zeolite single
 40 crystals, the benchmark C-ZSM-5 and the nanosized zeolite, an intelligent gravimetric
 41 analysis (IGA) on the diffusion of 1,3,5-trimethylbenzene (1,3,5-TMB) is performed
 42 under inert conditions (Figures 4D and S24, Tables 1 and S4 and Equation 4) in
 43 Experimental procedures). It is believed that only a few 1,3,5-TMB molecules (kinetic

1 diameter = 0.78 nm) can penetrate the micropores (= 0.55 nm) while most of them
2 will only be adsorbed on to the external surface of microporous zeolites. In
3 comparison, the interparticular macropores of Nano-ZSM-5 and the ordered
4 macro-mesoporous networks of OMMS-ZSM-5(400,30) offer sufficient space for
5 more 1,3,5-TMB molecules. The relative diffusion rate of OMMS-ZSM-5(400,30)
6 sample, which is calculated by linear fits of the normalized uptake profile (Q_t/Q_0), is
7 1.8 and 2.8 times higher than those in Nano-ZSM-5 and C-ZSM-5. The adsorption
8 amount of 1,3,5-TMB at dynamic conditions at different diffusion time was measured
9 for OMMS-ZSM-5(400,30), OMMS-ZSM-5(600,100), C-ZSM-5 and Nano-ZSM-5
10 and is given in Figure S23 and Table S4. It is clearly seen that at dynamic condition,
11 OMMS-ZSM-5 zeolite single crystals have much higher accessibility to acid sites
12 than that of C-ZSM-5 and Nano-ZSM-5. This observation is directly linked to the
13 excellent connectivity between micropores and meso-macropores of OMMS-ZSM-5
14 zeolite single crystals. The highest adsorption amounts of 1,3,5-TMB and the highest
15 diffusivity observed for OMMS-ZSM-5 (600, 100) explain its best catalytic
16 performance. All the results obtained by the laser hyperpolarized ^{129}Xe NMR, the
17 PFG ^{129}Xe NMR, kinetic study of 1,3,5-TIPB cracking reaction, diffusion and
18 dynamic accessibility of 1,3,5-TMB by IGA all in excellent consistence explains the
19 excellent catalytic performance of OMMS-ZSM-5 zeolite single crystal. The highly
20 ordered hierarchically micro-meso-macroporous structure of ZSM-5 single crystals is
21 thus a decisive factor in improving the mass transfer property. By varying mesopore
22 and macropore size within each of our ZSM-5 single crystals, the relative diffusion
23 rate of OMMS-ZSM-5(600,100) with the largest mesopore and macropore size is
24 found to be 7 and 4.5 times higher than that of C-ZSM-5 and Nano-ZSM-5. The
25 above results show that the highly ordered and interconnected hierarchically
26 macro-mesoporous structure inside our zeolite single crystal is much more effective
27 than macroporous nanocrystals. Such unique hierarchically porous architecture can
28 maximize the intracrystalline diffusion rate of reactants and products by reducing
29 effective diffusion length, leading to a much higher zeolite utilization efficiency, a
30 significant reduction in coking and deactivation rate and improved catalyst
31 lifetime.^{53,54}

32 **Conclusions**

33 ZSM-5 single crystals with a highly interconnected and ordered intracrystalline
34 macro-mesoporous system and variable Si/Al ratios and tunable mesopore and
35 macropore size have been successfully synthesized by a bottom-up confined
36 crystallization process. The hierarchical porous system in the single crystals results in
37 a significantly improved diffusion performance for various reactants compared to the
38 benchmark commercial microsized micro-mesoporous and micro-macroporous
39 nanosized zeolite counterparts, leading to excellent catalytic performance in bulky
40 molecules reaction and MTO process. Using OMMS-ZSM-5(400, 30) as example, not
41 only the catalytic activity is much higher in bulky molecule cracking reaction, the
42 effective diffusion rate of molecules reaches 10 and 5 folds higher than that of
43 benchmark commercial micro-mesoporous C-ZSM-5 and the state-of-the-art
44

1 micro-macroporous Nano-ZSM-5, respectively. Its utilization efficiency in
2 1,3,5-TIPB cracking reaction can reach a very high value of 0.84, which is doubled
3 compared to that of C-ZSM-5 catalyst and 20% higher than that of Nano-ZSM-5
4 catalyst. The average coking rate in MTO reaction is 2.2 and 1.7 times lower than that
5 for C-ZSM-5 and Nano-ZSM-5, respectively. Most importantly, the catalytic
6 performances of our zeolite single crystals can be significantly improved by varying
7 the Si/Al ratio and, in particular, by adjusting the intra-crystalline mesopore and
8 macropore size. The relative diffusion rate of OMMS-ZSM-5(600, 100) with the
9 largest mesopore and macropore size (Table 1) can reach even 7 and 4.5 times higher
10 than that of C-ZSM-5 and Nano-ZSM-5, respectively, its average coking rate in MTO
11 reaction can attain 8.9 and 7.2 times lower than that for C-ZSM-5 and C-ZSM-5,
12 respectively. Compared to C-ZSM-5 and Nano-ZSM-5 catalysts, the life time of this
13 sample can be extended 13 and 3.5 times longer, respectively.

14 Our synthesis route to zeolite single crystals is not only simple and readily
15 up-scalable, but also highly versatile and has already been extended to the successful
16 synthesis of other zeolite structures and zeotype materials, including Beta, TS-1,
17 MOR, Y and SAPO-34. The utilization of such zeolite single crystals with excellent
18 properties to a large series of industrial catalytic reactions can lead to the design of
19 innovative and intensified reactors and processes with much highly enhanced
20 efficiency, high catalytic activity, selectivity and life time, minimum energy, time and
21 raw material consumption. The zeolite single crystal reactor concept can be exploited.
22 Most importantly, our synthesis strategy can be extended to fabricate single
23 crystalline solids of any chemical compositions such as transition metal oxides (TiO₂,
24 ZnO, ...), phosphate compounds, silicate compounds,..., perovskites,... The key
25 parameters to the construction of hierarchical porous structure within each single
26 crystal by our innovative confined crystallization method are the control of the water
27 content in the dried precursor gel and the occupancy rate and the crystallization rate of
28 dried precursor gel confined in the hierarchical porous template. The utilization of
29 such single crystalline solids in a various reactions and procedures can lead to a
30 revolution in industrial processes.

31

1 **EXPERIMENTAL PROCEDURES**

2 **Resource Availability**

3 *Lead Contact*

4 Further information and requests for resources and reagents should be directed to and
5 will be fulfilled by the lead contact, Bao-Lian Su (bao-lian.su@unamur.be).

7 *Materials Availability*

8 OMMS-ZSM-5s generated in this study will be made available on request, but we
9 may require a payment and/or a completed Materials Transfer Agreement if there is
10 potential for commercial application.

12 *Data and Code Availability*

13 The data supporting the findings of this study are available in the article and
14 Supplemental Information, or from the lead contact upon request.

16 **Hierarchical zeolite single crystals preparation**

17 *Synthesis of uniform polystyrene (PS) spheres with tunable diameter*

18 The uniform polystyrene (PS) spheres with a diameter of 220 nm were synthesized by
19 emulsion polymerization. 0.6 g Sodium laurylsulfonate (emulsifying agent) was
20 dissolved in 360 g deionized H₂O followed by addition of 47 g styrene. After adding
21 0.54 g potassium persulfate, the reaction was performed at 75 °C for 4.5 h under an
22 argon atmosphere. Polystyrene spheres with monodispersed sizes of about 220 nm
23 were obtained.

24 The uniform polystyrene (PS) spheres with a diameter of 420 nm were
25 synthesized by soap-free polymerization. 47 g styrene were added in 400 g deionized
26 H₂O followed by addition of 0.43 g potassium persulfate. The reaction was performed
27 at 80 °C for 5 h under an argon atmosphere. Polystyrene spheres with monodispersed
28 sizes of about 420 nm were obtained.

29 The uniform polystyrene (PS) spheres with a diameter of 620 nm were
30 synthesized by soap-free polymerization. 49 g styrene were added in 430 g deionized
31 H₂O followed by addition of 0.51 g potassium persulfate. The reaction was performed
32 at 70 °C for 11 h under an argon atmosphere. Polystyrene spheres with
33 monodispersed sizes of about 620 nm were obtained.

35 *Synthesis of hierarchically ordered macro-mesoporous carbon (OMMC) with tunable 36 diameter*

37 In a typical procedure, the mass composition of the initial reaction mixtures is
38 100PS/15SiO₂/15C₁₂H₂₂O₁₁/1.5H₂SO₄, polystyrene spheres (as-synthesized) were first
39 blended with silica sol (Ludox AS-40 colloidal silica) under magnetic stirring for 1 h.
40 The obtained colloidal suspension was mixed with sucrose at room temperature for 10
41 min. Sulfuric acid was then slowly added under stirring for another 10 min to obtain a
42 stable dispersion. The as-prepared dispersion was directly dried in an oven at 110 °C
43 for 6 h, then at 160°C for 6 h, followed by heating to 700 °C with a rate of 2 °C /min
44 under pure nitrogen flow (1 L/min). The temperature was kept at 700 °C for 4 h to

1 decompose the polymer spheres and carbonize sucrose. The obtained carbon/silica
2 composite was cooled in pure nitrogen and immersed in a 10 wt % hydrofluoric acid
3 aqueous solution to remove silica, followed by washing with deionized water and
4 drying at 60 °C to yield hierarchically ordered macro-mesoporous carbon (OMMC).
5 The size of the macropores in OMMC can be tuned by varying the size of the
6 corresponding polystyrene spheres (220 nm, 420 nm, 620 nm). OMMC templates
7 with a macropore size of ~200 nm (OMMC-200), ~400 nm (OMMC-400) and
8 ~600nm (OMMC-600) were obtained, respectively.

9
10 *Synthesis of ordered and interconnected hierarchically macro-meso-microporous*
11 *ZSM-5 single crystals (OMMS-ZSM-5s) with tunable meso-macroporosity and*
12 *tunable Si/Al molar ratio:*

13 Bottom-up confined crystallization process was used for the confined synthesis of
14 hierarchical ordered macro-mesoporous ZSM-5 single crystals within the OMMC
15 templates. In a typical synthesis, OMMC with different macropore size was
16 impregnated with an excess solution of tetrapropylammonium hydroxide (TPAOH),
17 tetraethyl orthosilicate (TEOS), aluminum isopropoxide and water. The molar
18 composition of the initial reaction mixture is 0.36TPAOH /SiO₂/xAl₂O₃/19.2 H₂O
19 (x=0.067, 0.040, 0.020). After slow evaporation of water from the mixture of the
20 precursors confined in OMMC template at room temperature, the mixture was
21 transferred to a Teflon-lined stainless-steel autoclave equipped with a porous metallic
22 network suspended at the middle of the autoclave for holding the mixture. Sufficient
23 water was placed at the bottom of autoclave to produce saturated steam. The
24 steam-assisted crystallization (SAC) process was performed at 180 °C for 10 h.
25 Thereafter, the samples were taken out of the autoclave and washed by filtration with
26 abundant deionized water. Finally, the product was dried at 60°C and further calcined
27 at 550 °C for 7 h to remove the carbon template. The as-synthesized samples are
28 denoted by OMMS-ZSM-5(x,y) (OMMS represents ordered macro-mesoporous
29 single-crystalline), where x and y represent the macropore size of OMMC templates
30 and Si/Al molar ratio. In this work, OMMS-ZSM-5s with controllable mesopore and
31 macropore size and tunable Si/Al ratio [OMMS-ZSM-5(200,100),
32 OMMS-ZSM-5(400,100), OMMS-ZSM-5(600,100), OMMS-ZSM-5(400,30) and
33 OMMS-ZSM-5(400,50)] were obtained.

34 Commercial microsized ZSM-5 (C-ZSM-5) and nanosized ZSM-5
35 (Nano-ZSM-5) from FUYU New Materials Technology Co., Ltd. were used as
36 reference samples with Si/Al ratios of 23 and 21, respectively.

37 38 **Catalyst Characterization**

39 XRD patterns were recorded on a Bruker D8 Advance diffractometer using CuK α
40 monochromatized radiation ($\lambda = 1.5418 \text{ \AA}$). The measurements were taken at 45 kV
41 and 40 mA in a continuous mode and a 2 θ range from 5° to 60° with a step increase of
42 0.02°.

43 Scanning electron microscope (SEM) images were obtained on a Hitachi S4800
44 field-emission SEM operated at 5 kV and 10 μ A. Transmission electron microscopy

1 (TEM), high resolution transmission electron microscopy (HR-TEM), selected area
2 electron diffraction (SAED) and high angle annular dark field scanning transmission
3 electron microscope (HAADF-STEM) studies were performed on a Thermo Fisher
4 Titan Themis 60-300 'cubed' microscope fitted with double aberration-correctors for
5 both TEM and STEM, operated at 120 kV. Three-dimensional (3D) reconstruction
6 was performed through ASTRA Tomography Toolbox.

7 The chemical composition of the samples was determined by inductively coupled
8 plasma (ICP) optical emission spectroscopy using a PerkinElmer Optima 4300DV.
9 The wavelength range was 165~782 nm and resolution was 0.006 nm (at 200 nm).

10 Ar and N₂ adsorption-desorption isotherms were recorded using a Micromeritics
11 ASAP 2020 gas sorptometer after the samples were degassed at 300 °C under vacuum
12 for 12h. The micropore surface area was determined from Ar adsorption isotherm
13 using t-plot method. By using the model based on non-local density functional theory
14 (NLDFT), micropore volume and micropore size were derived from the adsorption
15 branches of Ar isotherms with relative pressure p/p_0 of <0.01. The mesopore surface
16 area was determined from N₂ adsorption using BET method. By using
17 Barret-Joyner-Halenda (BJH) model, mesopore volume and mesopore size were
18 determined by the adsorption branches of N₂ isotherms. Total pore volumes were
19 estimated from the adsorbed amount at a relative pressure p/p_0 of 0.99.

20 Mercury intrusion porosimetry was performed with a Micromeritics Autopore IV
21 9500 operated in the pressure range from vacuum to 207 MPa. Samples were
22 degassed in situ prior to measurement. The pore size distribution was determined by
23 application of the Washburn equation. The macropore volume was determined by
24 the volume of Hg intruded into pores of >50nm diameter.

25 The NMR spectra were recorded at room temperature, using a Varian VNMRS
26 spectrometer operating at 9.4 T (²⁷Al freq. = 79.46 MHz; ²⁹Si freq = 79.46 MHz). The
27 probe used was a Varian/Chemagnetics HX 4 mm CPMAS. The samples were packed
28 in a standard 4 mm rotor and spun at 10 kHz. The number of transients range between
29 ~200 and 11000 for the ²⁹Si spectra, and between 2000 and 3500 for the ²⁷Al spectra.
30 For the ²⁷Al spectra, the parameters were: spectral width ~104 kHz, relaxation delay
31 100 ms, excitation pulse 3 μs, acquisition time 5 ms. For ²⁹Si spectra, the parameters
32 were: spectral width ~104 kHz, relaxation delay 6 ms, excitation pulse 3 μs,
33 acquisition time 5 ms.

34 Laser hyperpolarized ¹²⁹Xe NMR experiments were carried out at 110.6 MHz on
35 a Varian Infinity-plus 400 spectrometer using a 7.5 mm probe. Before each
36 experiment, samples (60–80 mesh) were dehydrated at 673 K under vacuum (<10⁻⁵
37 Torr) for 24 h. The optical polarization of xenon was achieved with a homemade
38 apparatus with the optical pumping cell in the fringe field of the spectrometer magnet
39 and a 60 W diode laser array (Coherent FAP-System). A flow of gas mixture (1% Xe
40 – 1% N₂ – 98% He) was delivered at the rate of 100–150 mL min⁻¹ to the sample in
41 the detection region via plastic tubing. Variable-temperature NMR measurements
42 were performed in the range of 153–273 K. All one-dimensional spectra were
43 acquired with 3.0 μs $\pi/2$ pulse, 100–200 scans, and 2 s recycle delay. The chemical
44 shifts were referenced to the signal of xenon gas. Although the line of the xenon gas is

1 temperature dependent, generally chemical shifts vary no more than 1 ppm in the
2 temperature range of the experiments.

3 Prior to PFG NMR diffusion measurements, the sample was dehydrated on a
4 vacuum system at 673 K for 16 h. Then the samples were transferred into the NMR
5 tube with pressure valve in glove box. After degassing overnight at 393 K, Xe was
6 quantitatively introduced into NMR tube on a homemade uptake apparatus. After Xe
7 adsorption, the tube was sealed and equilibrated at room temperature. The loading of
8 adsorbed xenon was calculated by the ideal gas equation. All data were acquired at
9 equilibrium condition. ^{129}Xe PFG NMR experiments were conducted on a 600 MHz
10 Bruker Advance III spectrometer equipped with a 5 mm Diff50 diffusion probe
11 delivering a maximum gradient of 1800 G cm^{-1} in the Z-direction. A bipolar-gradient
12 stimulated echo sequence (STEBP, 13-interval pulse sequence) was applied in
13 diffusion measurements in order to eliminate distortions of the PFG NMR results by
14 internal magnetic field inhomogeneities (*i.e.* internal magnetic field gradients)
15 induced by susceptibility variations in heterogeneous samples.

16 The acidity of the samples was determined by NH_3 temperature programmed
17 desorption (TPD) using a Micromeritics ASAP 2920. For measurement, 50 mg
18 sample powder was introduced into a quartz reactor and degassed under vacuum at
19 $550 \text{ }^\circ\text{C}$. After cooling to room temperature, NH_3 gas was adsorbed for 1 h. This was
20 followed by evacuation at $120 \text{ }^\circ\text{C}$ to remove free and weakly adsorbed NH_3 .
21 Desorption profile was then measured with evacuation at the temperature gradient of
22 $10 \text{ }^\circ\text{C min}^{-1}$, using a thermal conductivity detector (TCD).

23 24 **Catalytic Stability Test**

25 For mechanical stability test, the samples were pressed under pressure at 5 MPa, 10
26 MPa and 15 MPa. The resultant samples were characterized by SEM and XRD.

27 For thermal stability test, the samples were treated in a muffle furnace at two
28 different temperatures: 800 and 1000 $^\circ\text{C}$ for 1h. The resultant samples were
29 characterized by SEM, XRD and Ar gas adsorption.

30 For hydrothermal stability test, the samples were pressed into moulds and placed
31 into high temperature hydrothermal ageing tube under N_2 flow. Once the temperature
32 reached to $780 \text{ }^\circ\text{C}$, N_2 flow was shut down. Then the samples were treated in saturated
33 steam at 780°C for 2 h. The resultant samples were characterized by XRD and Ar gas
34 adsorption.

35 36 **Catalytic Activity Test**

37 Before the catalytic activity evaluation, all the samples were converted into H^+ form.
38 As a typical run, the sample was ion-exchanged with NH_4NO_3 (0.6 M) aqueous
39 solution at 80°C for 3h, followed by calcination at $550 \text{ }^\circ\text{C}$ for 6 h. This procedure was
40 repeated three times. The methanol to olefines (MTO) reaction was performed in a
41 stainless steel micro-catalytic-reactor. The catalyst sample (0.19 g) was first diluted
42 with 1.81 g quartz sand into a well-distributed mixture and placed in
43 micro-catalytic-reactor, was then preheated at $500 \text{ }^\circ\text{C}$ for 1 h under N_2 flow (50
44 ml/min) beforehand. After settling the temperature at $480 \text{ }^\circ\text{C}$, methanol was uniformly

1 pumped into a preheated device (180 °C) to transform into vapor phase and then
 2 mixed with N₂ flow (90 ml/min) into a homogeneous phase. For a better and quick
 3 comparison, the tests under harsh conditions with a high weight hourly space velocity
 4 (WHSV) of methanol feedstock was used. The WHSV was settled at 3.6 h⁻¹. The
 5 products were collected (time interval was 0.6 hour⁻¹ and analyzed by an on-line gas
 6 chromatograph (Agilent 6820) equipped with a flame ionization detector. For the
 7 conversion calculations, dimethyl ether (DME) was considered as a reactant rather
 8 than a product.

9 **Kinetic Evaluation**

11 Bulky 1,3,5-triisopropylbenzene (1,3,5-TIPB) cracking reaction was employed to
 12 quantitatively determine effective diffusivity and reactivity of different MFI zeolite
 13 catalysts. The reaction was taken in a specially-made stainless steel catalytic
 14 microreactor. 0.12g catalyst sample was dispersed in 4.68g quartz sand. Then the
 15 mixture was preheated at 500 °C for 1 h under N₂ flow before reaction. 1,3,5-TIPB
 16 was uniformly pumped into the preheated device to transform the liquid into vapor
 17 which was then mixed with N₂ as the carrier gas. The contact time was adjusted by
 18 changing the flow rate of the gas mixture. The catalytic activities of the zeolite
 19 samples were evaluated at different temperatures. The products were analyzed by an
 20 on-line gas chromatograph (Agilent 6820) equipped with a flame ionization detector.
 21 Before the catalytic test, the elimination of external diffusion was confirmed by
 22 establishing the conversion-contact time curves over different amounts of catalysts.

23 The rate constants of different zeolites in 1,3,5-TIPB cracking reaction at
 24 different temperatures and the reaction activation energy with/without diffusion limit
 25 (i.e. apparent/intrinsic activation energy, $E_{app,a}/E_{int,a}$) in zeolite catalysts have been
 26 calculated by fitting with Arrhenius equation. Thiele modulus φ is defined as
 27 Equation 1 and can be obtained by Equation 2. Effectiveness factor η was calculated
 28 by Equation 3 for zeolite with geometry of flat plate (thickness $2L$).

$$29 \quad \varphi = \sqrt{\frac{r_{intrinsic}}{r_{diffusion}}} = L \sqrt{\frac{K}{D_{Aeff}}} \quad \text{(Equation 1)}$$

30 where φ is the thiele modulus, $r_{intrinsic}$ is the max reaction rate, $r_{diffusion}$ is the max
 31 diffusion rate, L is the distance from the center of the catalyst pellet to the surface, K
 32 is the reaction rate constant, $D_{A,eff}$ is the effective diffusivity in the zeolite pores.

$$33 \quad \frac{E_{app,a}}{E_{int,a}} = \frac{1}{2} + \varphi \frac{1 - \tanh^2 \varphi}{2 \tanh \varphi} \quad \text{(Equation 2)}$$

34 where $E_{app,a}$ is the reaction activation energy with diffusion limit (i.e. apparent
 35 activation energy), $E_{int,a}$ is the reaction activation energy without diffusion limit (i.e.
 36 intrinsic activation energy), φ is the thiele modulus.

$$37 \quad \eta = \frac{\tanh \varphi}{\varphi} \quad \text{(Equation 3)}$$

38 where η is the effectiveness factor, φ is the thiele modulus.

40 **Diffusion and Dynamic accessibility Test**

41 1,3,5-trimethylbenzene diffusion measurement in ZSM-5 zeolites was performed

1 using a computer-controlled intelligent gravimetric analyzer (IGA, Hiden Analytical
 2 Ltd., Warrington, UK). An ultrahigh vacuum system was employed by which
 3 adsorption isotherms and corresponding kinetics can be accurately acquired. A
 4 sensitive microbalance (resolution of 0.2 μg) was mounted in an enclosure with fitted
 5 thermostat to remove thermal coefficients of the weighing system and thus provide
 6 high stability and accuracy. The zeolite sample (100 ± 1 mg) was degassed under
 7 vacuum ($<10^{-5}$ Pa) at 450 $^{\circ}\text{C}$ for 10 h prior to the adsorption measurement. The
 8 system temperature was then set at 50 $^{\circ}\text{C}$, regulated within 0.1 $^{\circ}\text{C}$ by a water bath.
 9 The relative pressure was rapidly raised to 0.099 and then kept constant during the
 10 tests. For each step, the amount of adsorbate introduced in the system was kept small
 11 enough to keep the adsorption process isothermal. The weight increase of sorbents
 12 was automatically measured by a microbalance and continuously recorded as a
 13 function of time under the settled gas vapor pressure.

14 Diffusion coefficient D was obtained based on Fick's Law. Eqn (4) gives the
 15 fitting formula of diffusion coefficient D for zeolite with geometry of flat plate
 16 (thickness $h = 2L$) where Q_0 is the initial adsorbed quantity, Q_t is the intermediate
 17 adsorbed quantity, Q_{∞} is the equilibrium adsorbed quantity and D is diffusion
 18 coefficient.

$$\frac{Q_t - Q_0}{Q_{\infty} - Q_0} = 1 - \sum_{n=0}^{\infty} \frac{8}{[(2n+1)\pi]^2} \exp\left[\frac{-D(2n+1)^2\pi^2 t}{h^2}\right] \approx \frac{8}{\sqrt{\pi^2}} \left(\frac{D}{h^2}\right)^{1/2} \sqrt{t}$$

19 (when t is small) (Equation 4)
 20 where Q_0 is the initial adsorbed quantity, Q_t is the intermediate adsorbed quantity, Q_{∞}
 21 is the equilibrium adsorbed quantity, n is the natural numbers, D is the diffusion
 22 coefficient for zeolite with geometry of flat plate (thickness $h = 2L$), t is the adsorbed
 23 time, h is the thickness of zeolite.

24

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41

1 AUTHOR CONTRIBUTIONS

2 L.-H. C. and B.-L. S. conceived the project, L.-H. C., Z.-K. X., G. V. T., and B.-L. S.
3 supervised the study, M.-H. S. designed and performed experiments, characterized
4 samples, analyzed data and wrote the draft of the manuscript. J. Z. performed the
5 catalysis reactions, analyzed the catalysis data and performed kinetic evaluation. Z.-Y.
6 H. and S. T., performed transmission electron microscopy and tomography analysis.
7 L.-Y. L. and Y.-D. W. performed the diffusion test and analyzed the diffusion data. T.
8 H. provided advice for the writing of this manuscript. L.-H. C. wrote the manuscript.
9 L.-H. C., G. V. T., Z.-K. X., T. H. and B.-L. S. revised the manuscript and B.-L. S.
10 finalized the manuscript.

11 DECLARATION OF INTERESTS

12 The authors declare no competing interests.

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1 **Figure 1. Synthesis and structure of OMMS-ZSM-5**

2 (A-E) Schematic illustration of the synthesis route of OMMS-ZSM-5.

3 (F) The enlarged image of the selected yellow cubic area in (E), confirming the
4 *abcabc*..... stacking in FCC arrangement. The units in layer *a* are shaded white, those
5 in layer *b* are green and those in layer *c* are pink. Such FCC arrangement contains two
6 voids: the tetrahedral void (T_d , blue) and the octahedral void (O_h , red).

7
8 **Figure 2. Characterization of OMMS-ZSM-5**

9 (A-D) SEM images of OMMS-ZSM-5(400,30).

10 (E-G) Schematics of an individual crystal from three directions corresponding to B-D.

11 (H) HAADF-STEM image and ED pattern (inset) of an individual zeolite.

12 (I) TEM image of enlarged area in (H).

13 (J-L) SAED patterns of zones 1(J), 2 (K) and 3 (L) in (I).

14 (M) HRTEM images of green area in (I) with enlarged intergrowth region inset.

15 (N-P) XRD pattern (N), Argon adsorption-desorption isotherms (O) and
16 micropore-size distribution (inset) and mercury intrusion (P) of
17 OMMS-ZSM-5(400,30).

18
19 **Figure 3. STEM tomography of OMMS-ZSM-5**

20 (A and B) HAADF-STEM image (A) and corresponding 3D tomographic
21 reconstruction (B) of an individual OMMS-ZSM-5(400,30) crystal.

22 (C and D) Schematic illustration of the FCC unit along the *Y* (C) and *Z* (D) axis.

23 (E and F) HAADF-STEM images of two adjacent orthoslices obtained from the 3D
24 reconstruction: A layer (E) and B layer (F) (*XZ* plane, perpendicular to the *Y* axis).

25 (G and H) HAADF-STEM images of two adjacent orthoslices obtained from the 3D
26 reconstruction: A layer (G) and B layer (H) (*XY* plane, perpendicular to the *Z* axis).

27
28 **Figure 4. Catalytic performance, effectiveness factor and diffusion effectiveness
29 of OMMS-ZSM-5**

30 (A) Catalytic performances over various catalysts in the MTO reaction. Circles denote
31 CH_3OH and DME conversion, whereas bars represent product selectivities.

32 (B) The correlation between the average coking rate and the deactivation rate of
33 different catalysts.

34 (C) Effectiveness factor η as a function of thiele modulus ϕ of different catalysts in
35 the cracking of 1,3,5-triisopropylbenzene.

36 (D) Normalized uptake (Q_t/Q_0) profiles of 1,3,5-trimethylbenzene over different
37 catalysts.

1 **Table 1. Catalytic activities for MTO reactions over various catalysts**

Catalyst	Si/Al ^{a)} [mol mol ⁻¹]	Life time ^{b)} [h ⁻¹]	Average deactivation rate ^{c)} [% h ⁻¹]	Coke amount [%]	Average Coke rate in MTO ^{d)} [mg g ⁻¹ h ⁻¹]	Relative diffusion rate ^{e)} [s ^{1/2}]
C-ZSM-5	23	2.4	25.0	3.73	8.88	0.0116
Nano-ZSM-5	21	9.0	18.0	9.01	7.15	0.0181
OMMS-ZSM-5 (400,30)	30	12.5	7.90	7.06	3.92	0.0330
OMMS-ZSM-5 (400,50)	50	18.5	4.31	6.62	2.29	0.0767
OMMS-ZSM-5 (400,100)	100	20.4	2.73	5.15	1.36	0.0477
OMMS-ZSM-5 (200,100)	100	14.1	6.20	4.72	2.21	0.0421
OMMS-ZSM-5 (600,100)	100	31.0	2.91	4.59	1.01	0.0805

2 ^{a)} determined by the elemental analysis (ICP).

3 ^{b)} the time for which the conversion of methanol exceeds 95%;

4 ^{c)} representing the percent loss of conversion from 95% to 50% per hour;

5 ^{d)} calculated by the mass loss per gram of catalyst over life time, the mass loss was accounted between
6 300□ and 700□;

7 ^{e)} the slope of the fitted line in normalized uptake profile, in which $t^{1/2}$ of the drops lower than $2 s^{1/2}$
8 (Equation 4 in Experimental procedures).

9

10

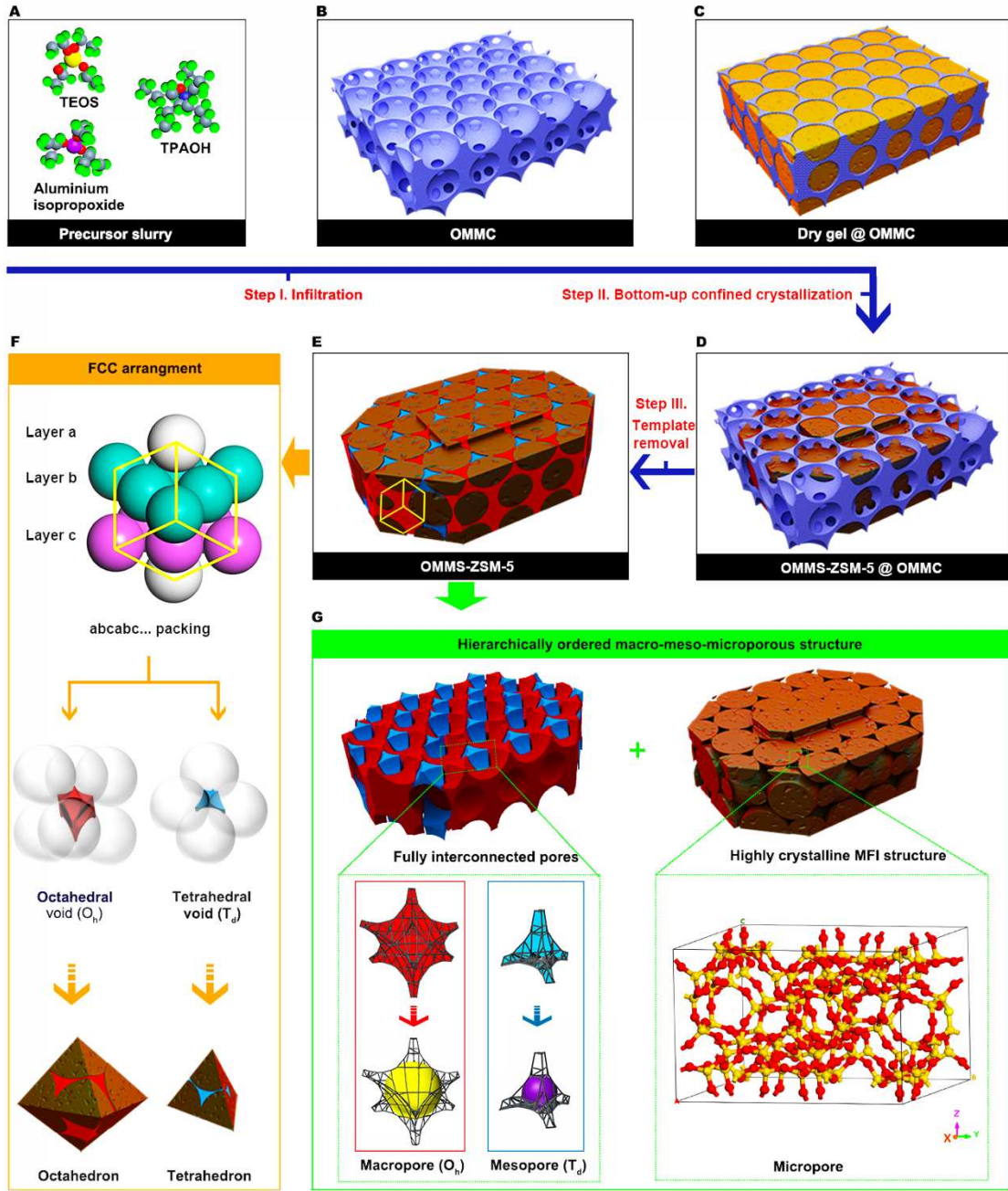
1 **Video S1. Zeolite spherical units that are densely stacked in a close-packed FCC**
2 **arrangement in an individual zeolite single crystal.**

3

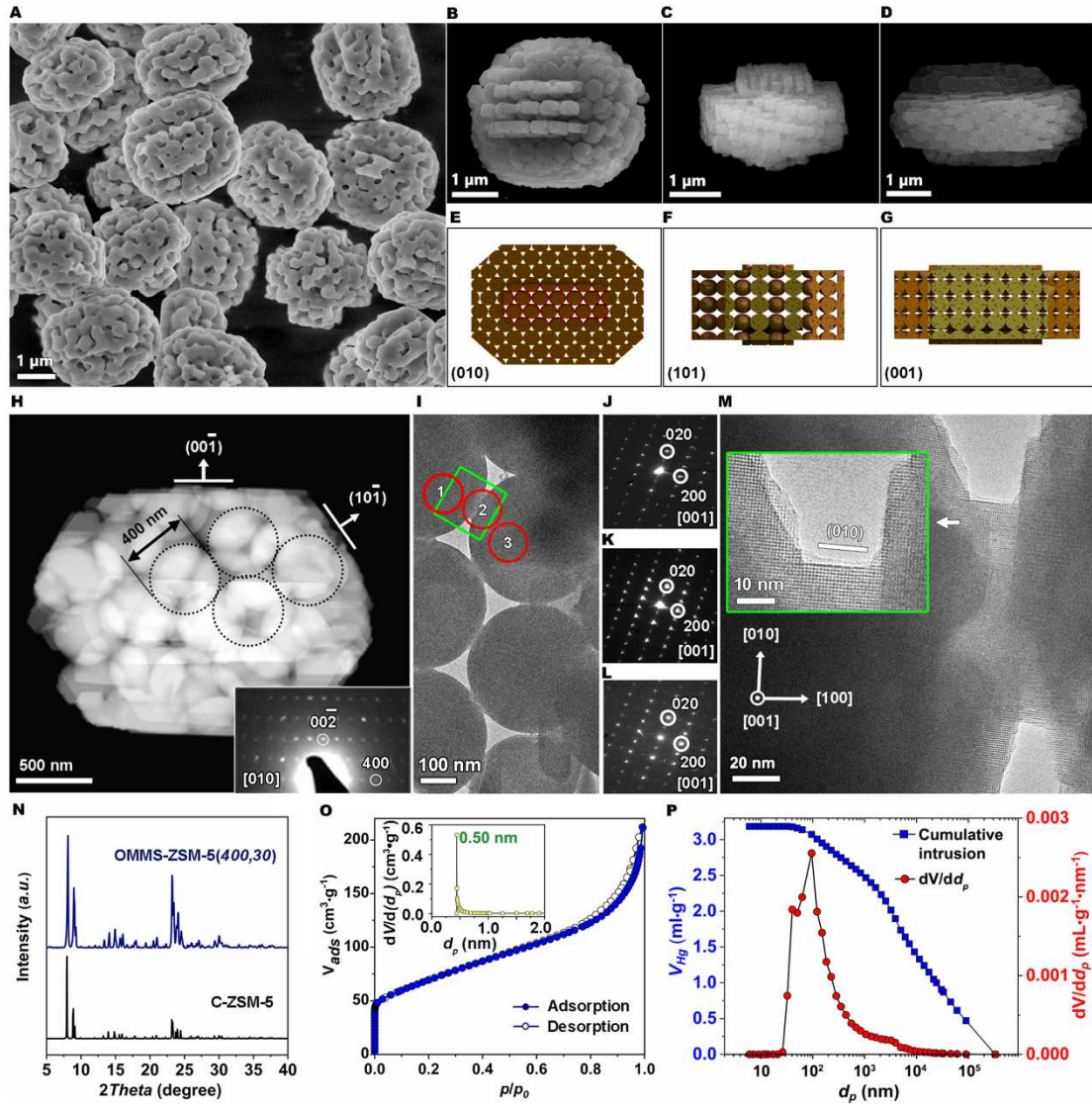
4 **Video S2. 3D electron tomographic reconstruction of OMMS-ZSM-5(400,30).**

5

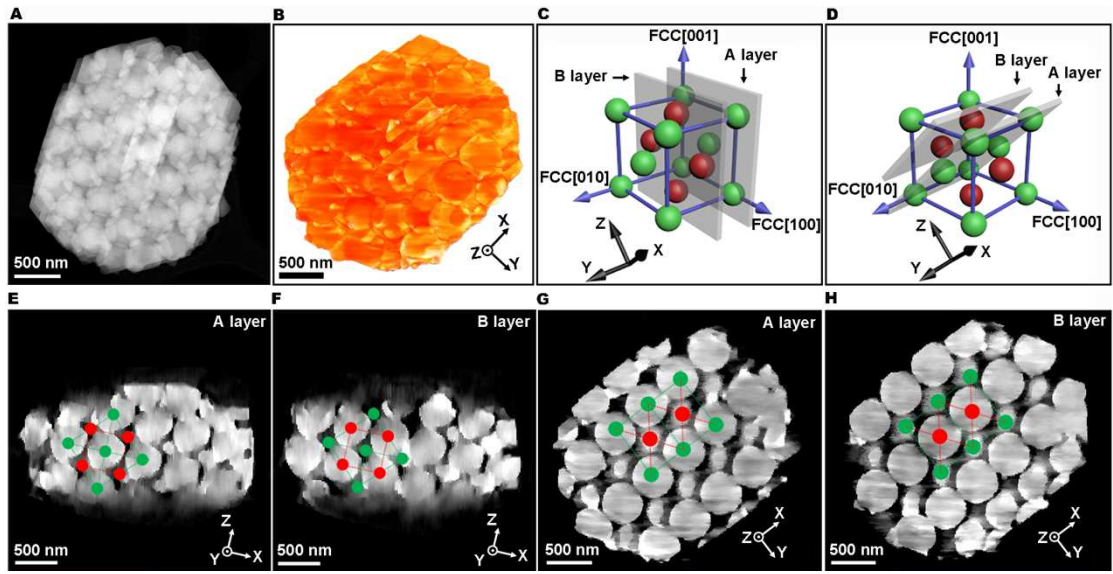
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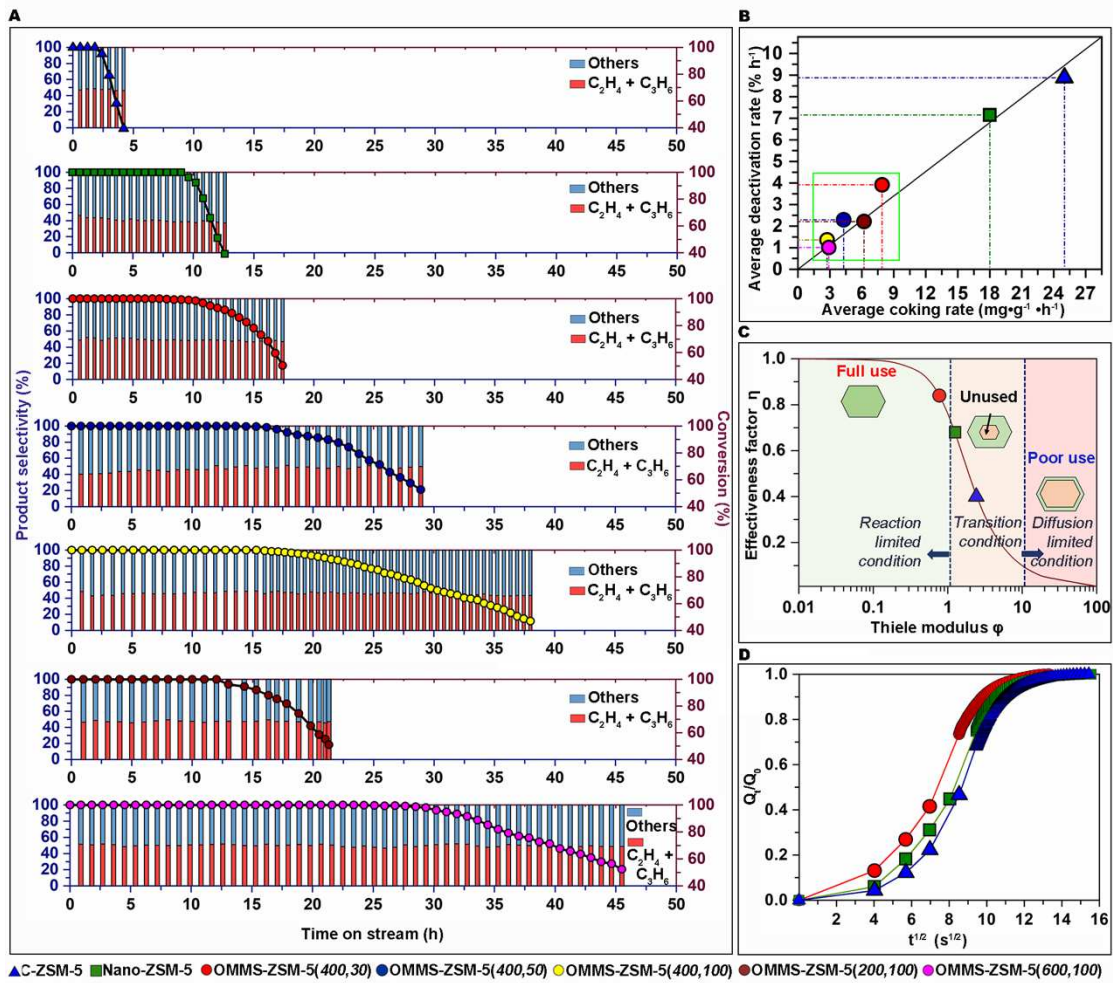
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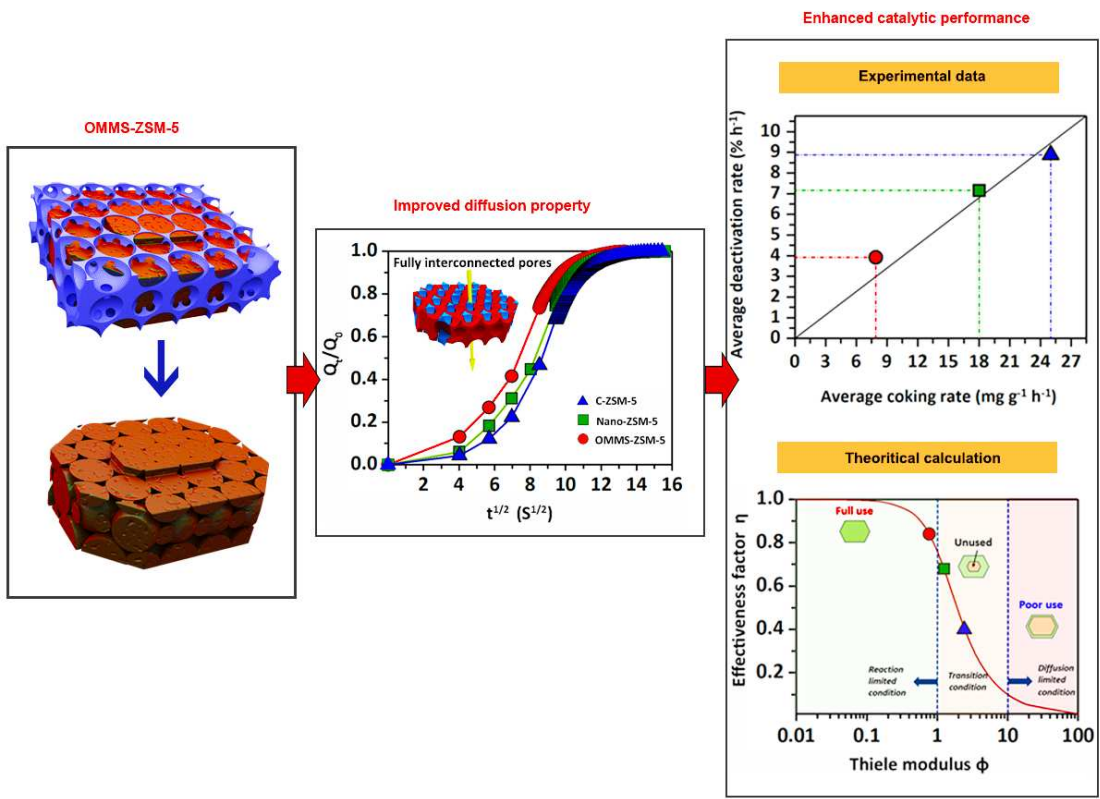


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